



# GeSn nano-structures for HEP applications

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# Advantages of GeSn for HEP detectors

- **Strain engineering**

- **Absorption** in NIR to MWIR vs Sn content
- Band gap with **indirect to direct transition** for Sn of around 7% @ 2  $\mu\text{m}$  emission in bulk
- **High mobility** due to smaller effective mass

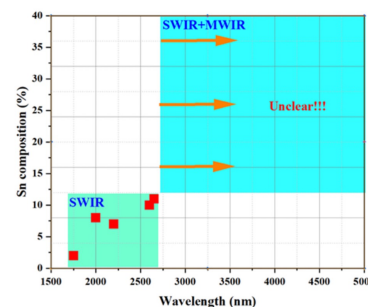
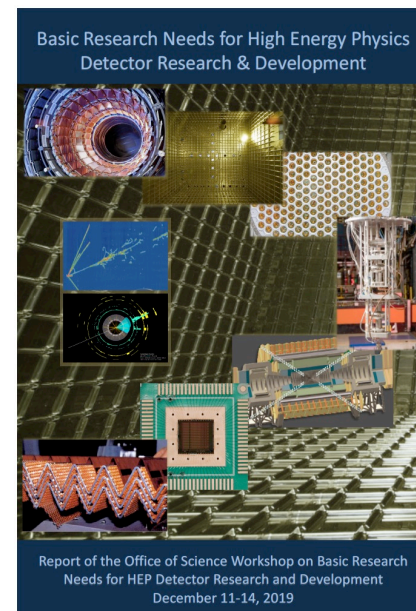


Figure 23. Sn content vs. cut-off wavelength of the GeSn PIN detector.

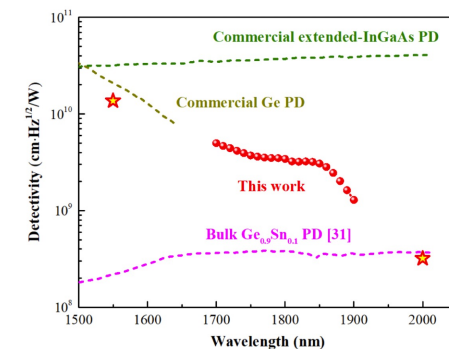
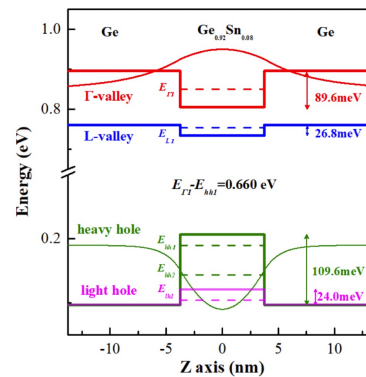
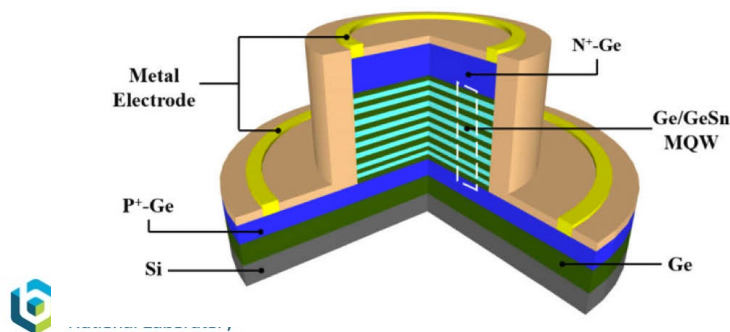
- **Applications to HEP Detectors**

- Extend detectors response beyond 1.5  $\mu\text{m}$  for Ge and 1  $\mu\text{m}$  for Si
- **Wavelength tuning** covering NIR to LWIR, i.e. 1 to 5  $\mu\text{m}$  possible
- **Multispectral devices demonstrated** i.e. working between 0.8 and 2  $\mu\text{m}$
- **Reaching direct band gap** beyond Si/Ge or III-V devices that are difficult to integrate on Silicon
- Monolithic integration with **metal-oxide-semiconductors (CMOS) technology**, i.e. lower manufacturing cost and easy adaptation



# Recent progress

- Devices based on GeSn include **detectors, modulators and lasers**
- For higher Sn content, large lattice mismatch resulting in **misfit and threading dislocations** degrades device performance; i.e. leakage current, recombination centers etc.
- Using **nanostuctures**
  - helps to **avoid strain relaxation**, similar to III-V alloys
  - Tune device parameters: absorption, mobility, energy band
- **Demonstrated example:**  $\text{Ge}_{0.92}\text{Sn}_{0.08}/\text{Ge}$  MQW *p-i-n* photo-detectors with high detectivity and low dark current based on the quantum confined Stark (QCSE) effect after *Zhou et al., Optics Express 28, 23 (2020)*



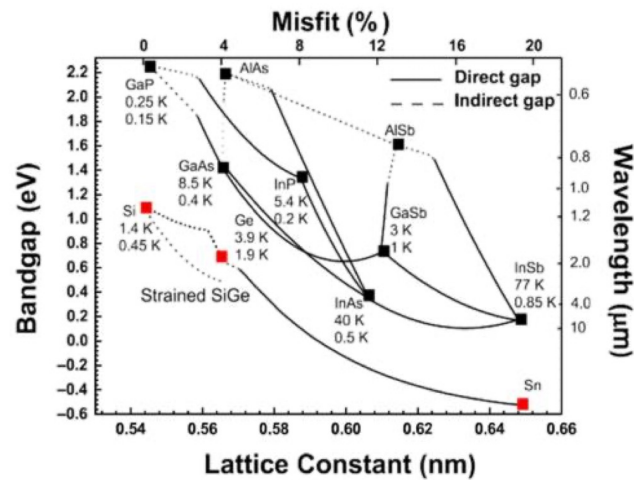
# Growth considerations and challenges for GeSn

- **Lattice mismatch 16%**
- **Low Sn solubility** in  $\text{Ge}_{1-x}\text{Sn}_x$  alloys
  - GeSn is eutectic alloy, thermal equilibrium solid solubility of Sn in the Ge matrix is as low as 1 at.% below 500°C . Additionally, the eutectic temperature of Ge – Sn binary alloys is as low as 231.1°C.
- **Sn segregation** during the growth and post-processing is one of the problems associated with the formation of homogeneous  $\text{Ge}_{1-x}\text{Sn}_x$  layers.

It strongly depends on substrate temperature



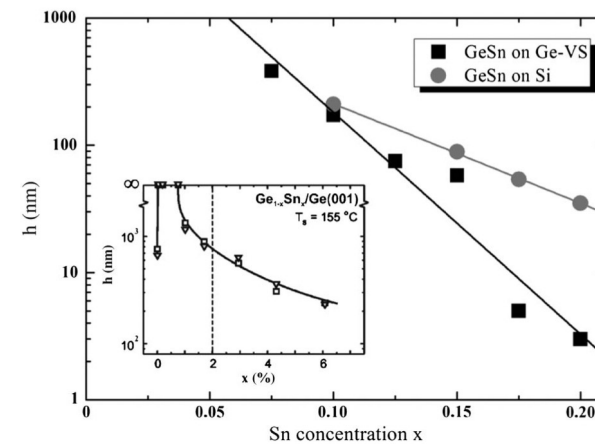
# Lattice mismatch



Radamson et al., Nanomaterials (2022)

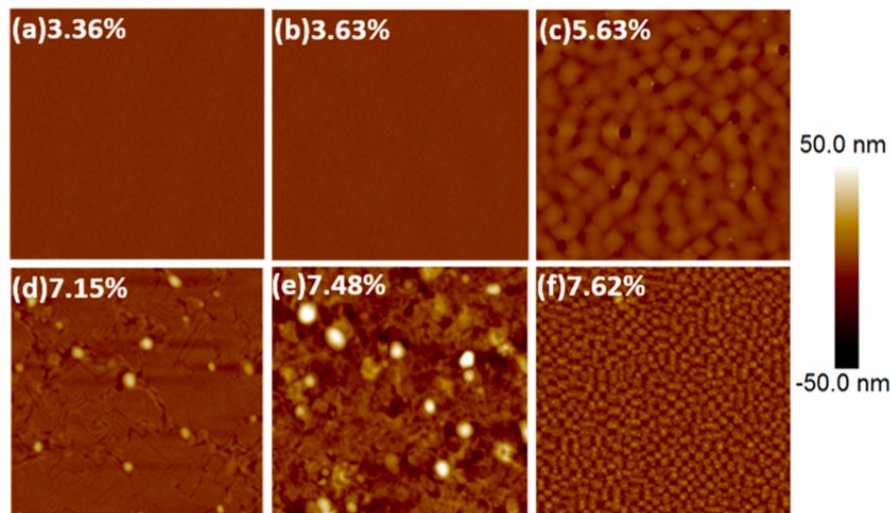
- For Sn content of 7% it is at 0.9%
  - Within moderate lattice mismatch with layer by layer growth for MBE
  - Critical thickness around 500 nm

- Lattice mismatch for Ge and Sn is 16%
  - Critical thickness below 1 ML



Arguirov et al., Photon Res 1,2 (2013)

# Sn segregation

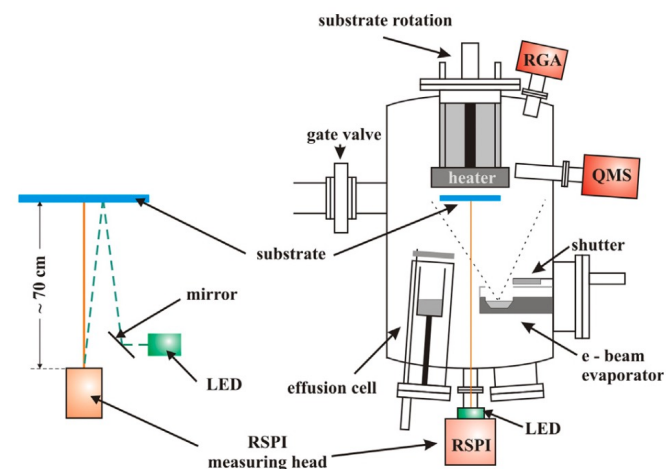


The Sn segregation can be clearly observed when the Sn concentration is above 7%. In FIG. 1. (d) and (e), Sn droplets are observed and become big with increasing Sn concentration. However, when the Sn concentration reaches 7.62%, the Sn droplets become much less, while the pyramids appear again

- Substrate temperature vs Sn-droplets formation
- In this study, GeSn thin films on Ge (001) with various Sn concentrations from 3.36 to 7.62% were grown by MBE.
- It is estimated that 7.62% Sn concentration could be the limitation at the growth temperature of 200 °C

# Solutions to MBE growth of GeSn

- To address segregation and low solubility use **Ultra-low temperature molecular beam epitaxy (MBE)**
  - Reported growth temperatures giving best results are at 160°C
  - At 200°C, surface segregation observed
- Growth on Si wafer with **Ge-VS** (virtual substrate)
- Reported **pseudo-morphic growth** for layer thicknesses between 260 to 30 nm
- No PL reported from these films

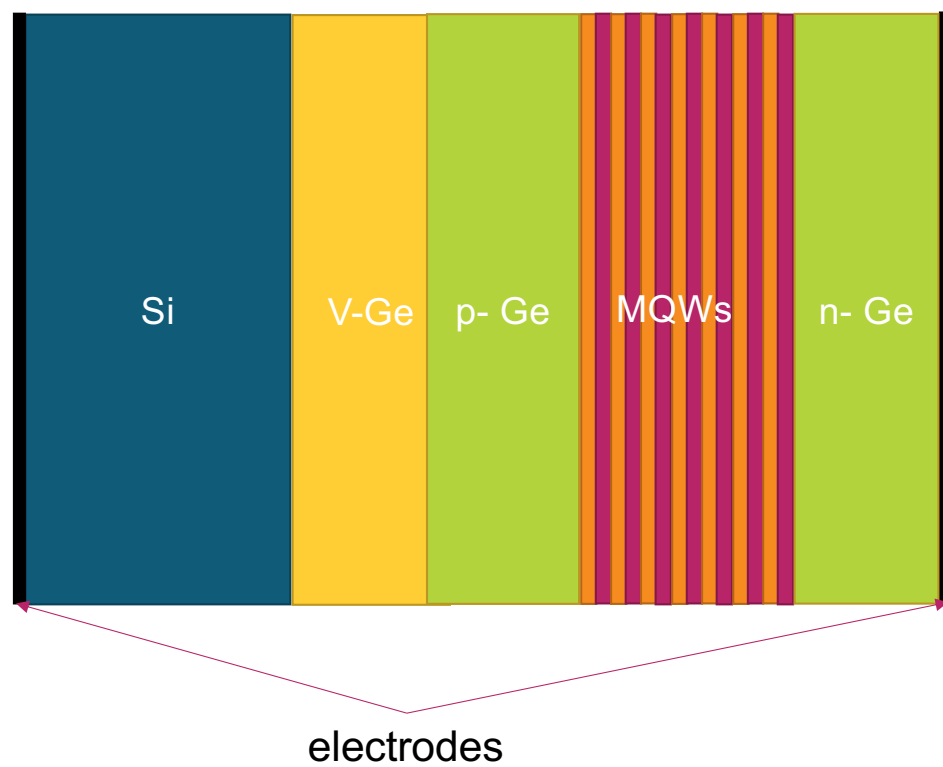


Growth chamber design

**Table 2**  
Values of angular displacement for GeSn layers with different Sn contents and the corresponding tetragonal strain.

Sample	Sn content RBS	$a_o$ [nm] (Vegard's law)	$f$ [%] lattice mismatch with Ge	$\Delta\theta_{theo}$ [deg] pseudomorph	$\Delta\theta_{exp}$ [deg]	Tetragonal strain [%]
C	0.047	0.56966	0.69	-0.34	-0.30	1.0
D	0.06	0.57074	0.88	-0.44	-0.39	1.4
E	0.08	0.5724	1.18	-0.58	-0.53	1.9
F	0.11	0.5749	1.62	-0.80	-0.77	2.7
G	0.125	0.57614	1.84	-0.91	-0.93	3.2

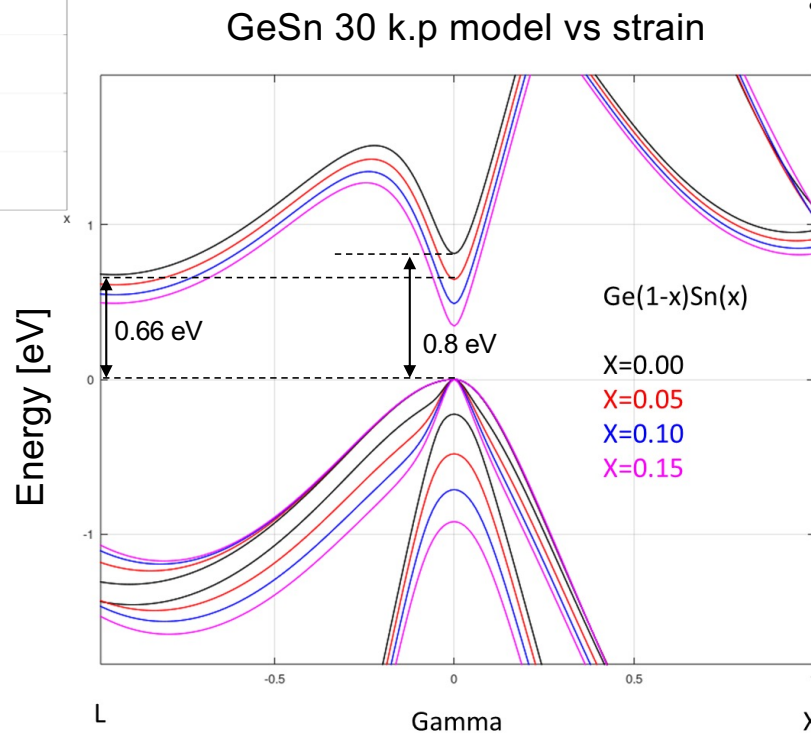
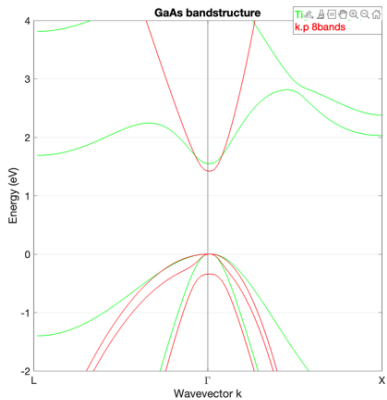
# Device concept





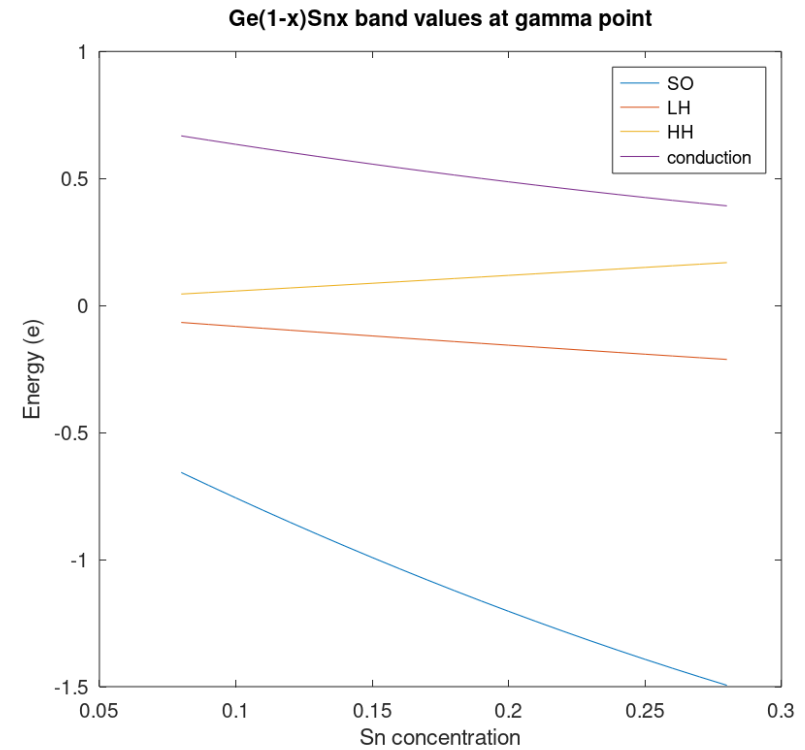
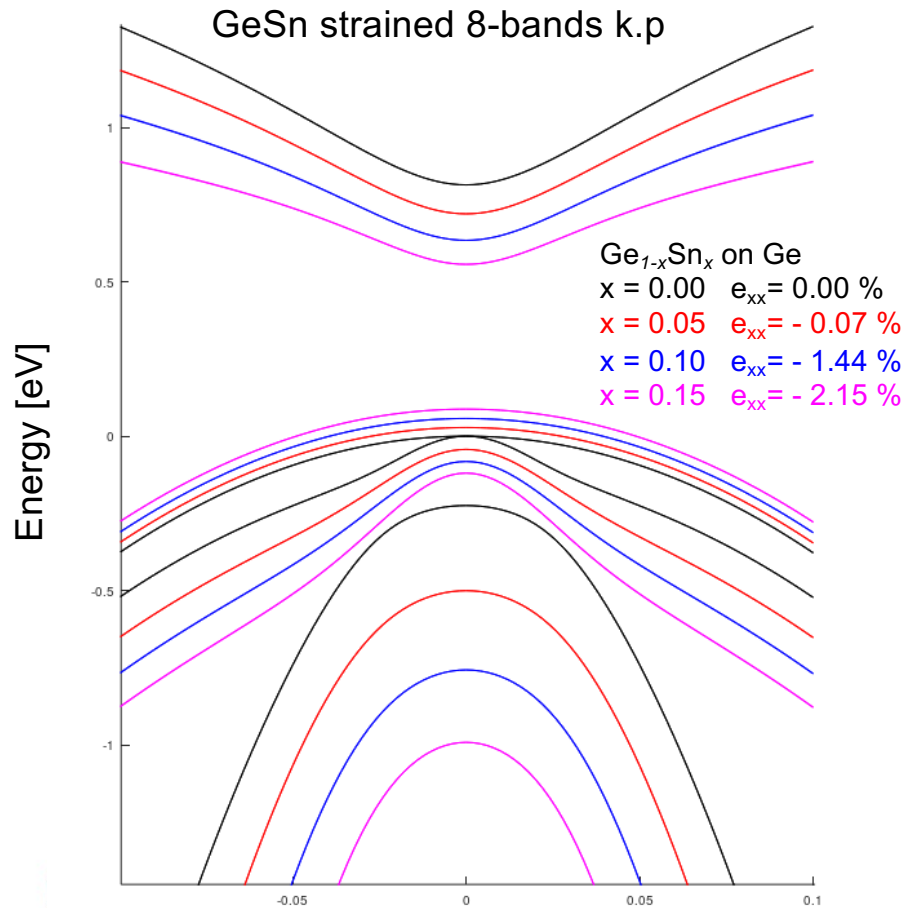
# Band structure modelling results

## Tight binding vs 8 k.p

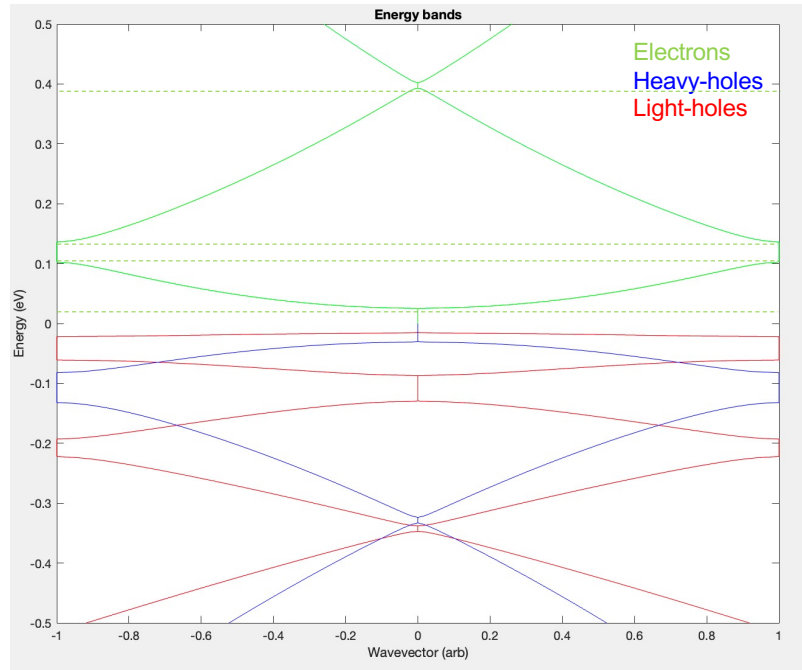


- Band modelling using k.p perturbation method
- 8 k.p vs 30 k.p
  - 8 k.p model sufficient around Gamma point
  - 30 k.p reproduces bands better for higher wave-vectors
- Consistent with Tight binding model
  - Some of the parameters are based on 'educated guesses' as data still lacking
  - Opportunity to fill these gap within this research

# Band structure modelling GeSn including strain

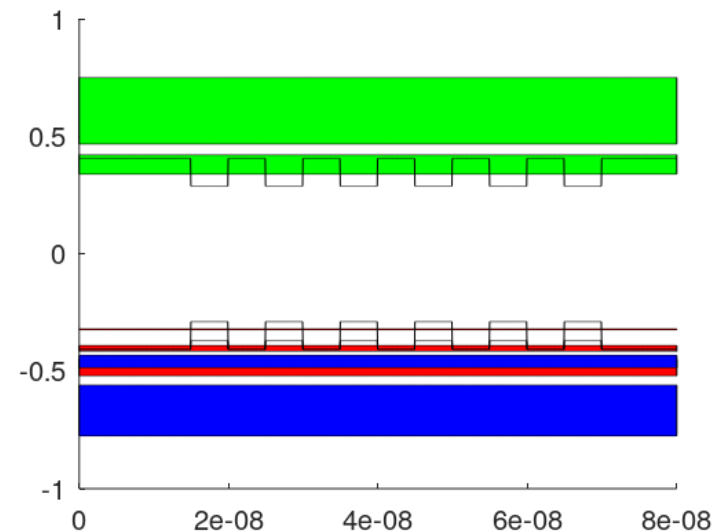


# Heterostructures modelling



Bands dispersion in MQWs

- Using modified Kronig-Penney model
- assuming QWs made of Ge and  $\text{Ge}_{0.9}\text{Sn}_{0.1}$  with superlattice period of  $a+b = (7.5 + 2.5) \text{ nm}$



# Brookhaven National Laboratory-IO fabrication capabilities



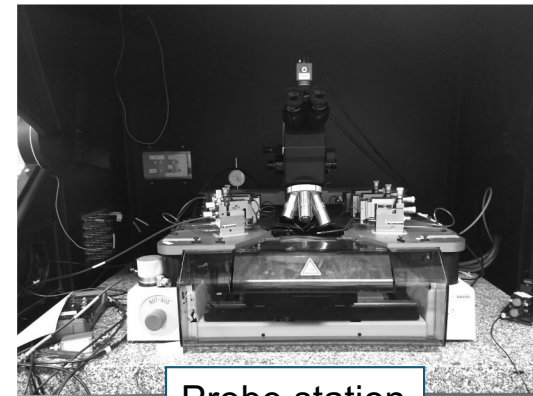
Sputtering, e-beam, thermal



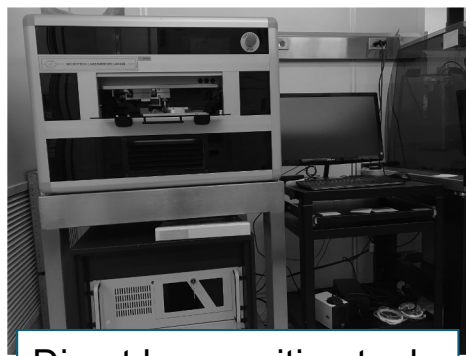
Thermal evaporation



SEM-EDX



Probe station



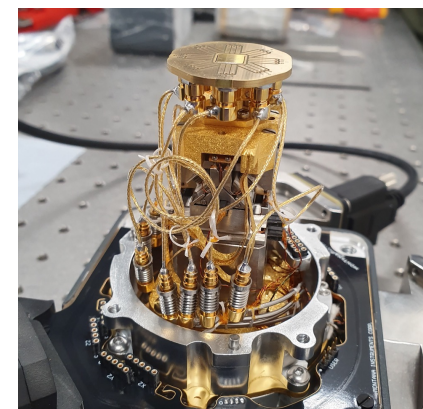
Direct laser writing tool



Augmented by  
access to CFN  
shared facilities



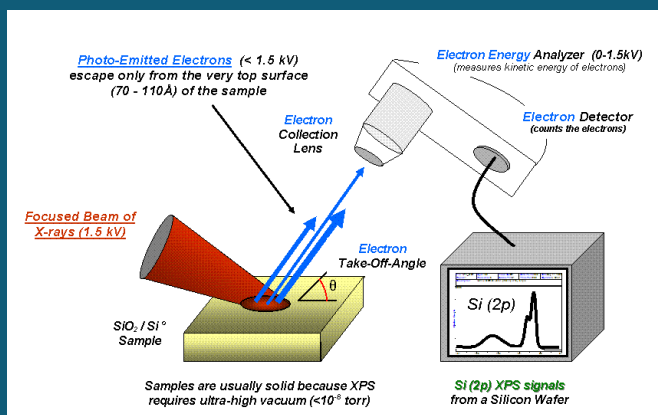
Clean room



Cryogenic probe station

# Brookhaven National Laboratory NSLS-II characterization

## Hard X-ray PhotoElectron Spectroscopy

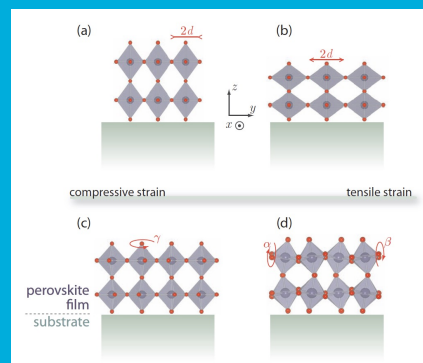


## Variable Kinetic Energy HAXPES

- Tune the HAXPES depth sensitivity (photoelectron kinetic energy) by tuning the X-ray excitation energy using a synchrotron X-ray beamline
- The X24A endstations form a unique measurement suite for surface to near bulk HAXPES by spanning X-ray excitation energy from 0.2 to 5 keV

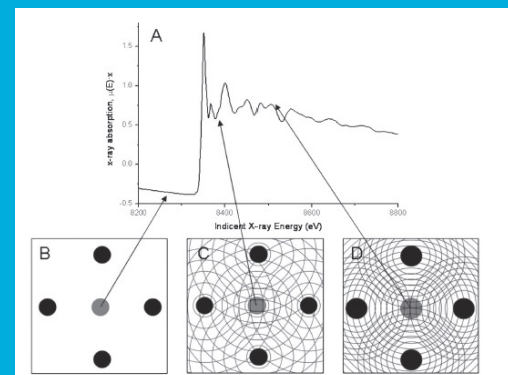
Example: Si electron binding energy 2p ~99 eV and 1s ~1840 eV, surface to bulk sensitivity (Lab source has fixed energy Al K $\alpha$ )

## Extended X-ray Absorption Fine Structure



Strain can result in changes in bond lengths or bond angles. This can be measured using EXAFS

J. M. Rondinelli and N. A. Spaldin, Adv. Mater. 23, 3363 (2011)





## Future work outlook

- Modelling of device performance using band-structure mapping and derived parameters i.e. gain, absorption (short term)
- Device growth and development (short term)
- Device integration (long term)