



# High-Z Detectors

Abdul K. Rumaiz

National Synchrotron Light Source II – Brookhaven National Laboratory

EDIT school 2023



# Outline

- X-ray detector: Quick Recap
- Germanium for Synchrotron Science
  - ✓ Overview/background
  - ✓ Detector System
    - Sensor
    - ASIC (MARS)
  - ✓ Ge-strip Detectors for synchrotron science
  - ✓ New Ge Detectors: Ge drift detectors, GAIA, Gepta-EX...
- Amorphous Selenium for Medical Imaging
  - ✓ Overview/background
  - ✓ Preliminary results
    - High spatial resolution detector with CMOS readout
  - ✓ Proposed research
    - SWAD with High-k blocking layer
- Large band gap semiconductor (Diamond, CZT for Gamma rays...)

Periodic table of the elements

Legend:

- Alkali metals
- Alkaline-earth metals
- Transition metals
- Other metals
- Other nonmetals
- Halogens
- Noble gases
- Rare-earth elements (21, 39, 57–71) and lanthanoid elements (57–71 only)
- Actinoid elements

1	2											13	14	15	16	17	18	
1	H											He						
2	3	4											5	6	7	8	9	10
	Li	Be											B	C	N	O	F	Ne
3	11	12	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18
	Na	Mg											Al	Si	P	S	Cl	Ar
4	19	20	21	22	23	24	25	26	27	28	29	30	31	32	33	34	35	36
	K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
5	37	38	39	40	41	42	43	44	45	46	47	48	49	50	51	52	53	54
	Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe
6	55	56	57	72	73	74	75	76	77	78	79	80	81	82	83	84	85	86
	Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn
7	87	88	89	104	105	106	107	108	109	110	111	112	113	114	115	116	117	118
	Fr	Ra	Ac	Rf	Db	Sg	Bh	Hs	Mt	Ds	Rg	Cn	Nh	Fl	Mc	Lv	Ts	Og

lanthanoid series	6	58	59	60	61	62	63	64	65	66	67	68	69	70	71
		Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu
actinoid series	7	90	91	92	93	94	95	96	97	98	99	100	101	102	103
		Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lr

\*Numbering system adopted by the International Union of Pure and Applied Chemistry (IUPAC). © Encyclopædia Britannica, Inc.

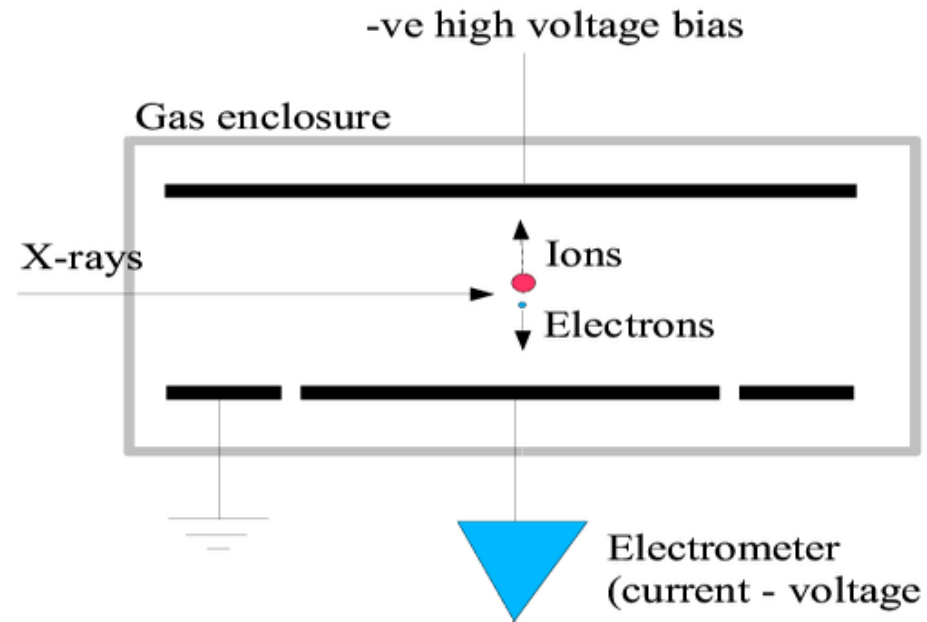


# X-ray Detectors

- Only discuss electronic hard x-ray detectors.
- All use same basic principle:
  - X-ray is absorbed by atoms within detector active volume
  - Atom is ionized
  - Electrons and ions are collected by imposing high electric field across collection electrodes.
- Differences come from detector material and readout electronics.

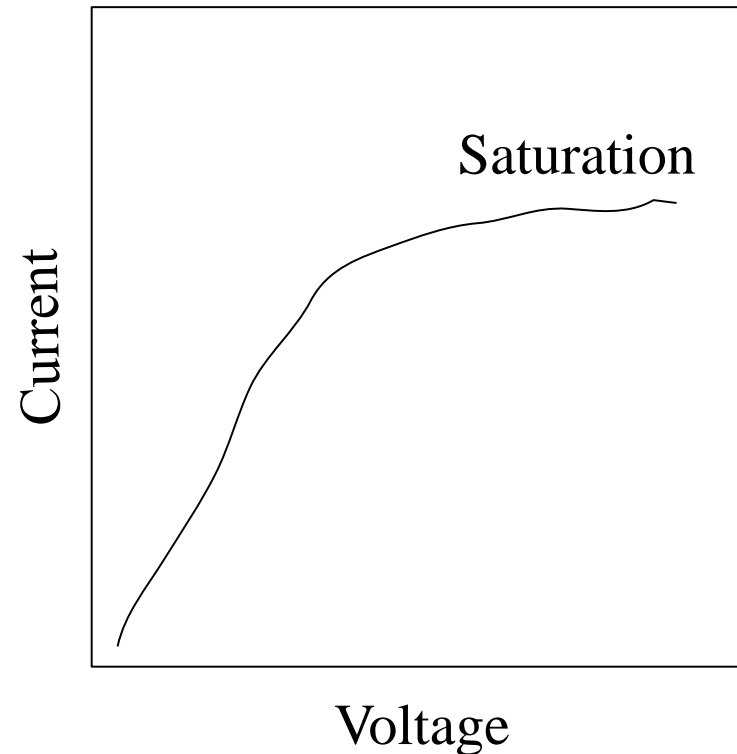
# Ionization chamber

- Photon is absorbed by gas atom
- Photoelectron and Auger electrons emitted (ionization)
- These electrons cause more ionization
- High voltage bias across plates causes electrons and ions to drift in opposite directions.
- Charges collected results in current flow which is proportional to incident x-ray intensity.



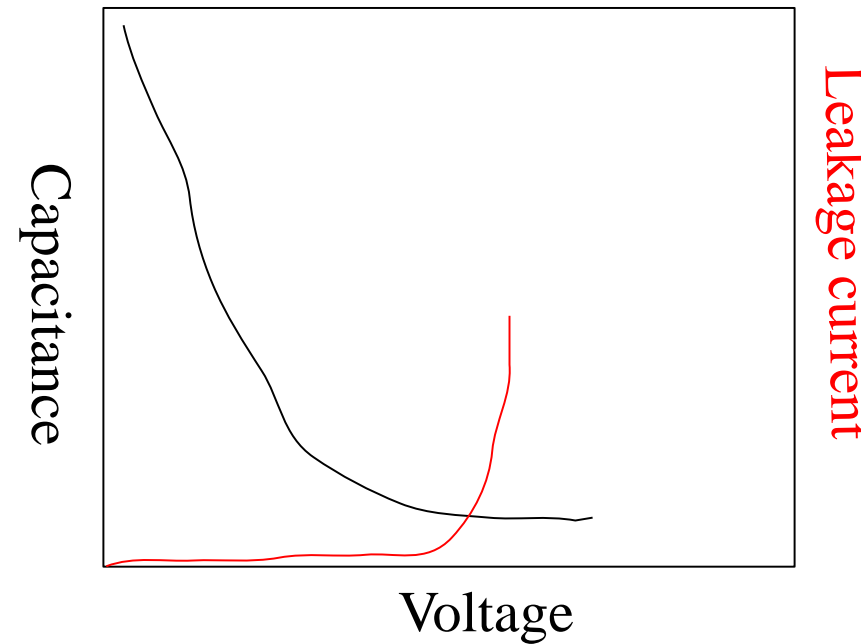
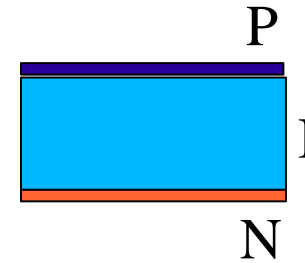
# Ion Chamber: Operating voltage

- Electrons and ions like to recombine
- Bias voltage pulls them apart before they can recombine.
- Measured current increases with bias voltage until all charges are collected. This is called 'saturation'.
- Ion chambers should ALWAYS be operated in the saturation region.
- Signal currents are small: care is needed to achieve good signal / noise performance



# Silicon photodiodes: solid state ion chamber

- PIN or high-resistivity PN photodiodes
  - 'solid-state ion chamber'
  - modest area
  - low leakage & capacitance
  - Usually used fully depleted
- Can be used either in integrating or photon-counting mode.



# Detector types

- Integrating
  - Ionization chambers
  - Silicon diodes
    - PIN diodes
    - PN diodes on high-resistivity silicon
    - PIPS detectors
  - CCD detectors
- Photon counting
  - HPGe detector
  - Silicon diode
  - Silicon drift detector
  - Gas proportional counter
  - Avalanche photodiode

## Which one to use when?

- Integrating detectors simple to operate when only interest is total intensity
  - No energy resolution
- Make use of proportionality between absorbed energy and generated charge
- Photon counting
  - Need to inspect every charge packet individually
- Amount of charge generated is small
  - $Q = E_{ph} / w * 1.6e-19$
  - $w = 3.6eV$  for silicon
  - One 8 keV photon generates 2200 electron-hole pairs in silicon
- Statistical fluctuations in number of charges determines ultimate resolution
  - Distribution is non-Poissonian: Fano factor

# Detector performance Metrics

What is an ideal detector?

- $10^9$  pixels
- $1\mu\text{m}$  spatial resolution
- 1 eV energy resolution
- 1 fs time resolution
- Count rates up to  $10^9/\text{pixels}$
- Efficient from few eV to 100 KeV



# Detector performance Metrics

What is an ideal detector?

- $10^9$  pixels
- $1\ \mu\text{m}$  spatial resolution
- 1 eV energy resolution
- 1 fs time resolution
- Count rates up to  $10^9/\text{pixels}$
- Efficient from few eV to 100 KeV

# Detector performance Metrics: Spatial Resolution

Ideally electrons and holes simply follow the field lines and end up on a certain electrode. So spatial resolution is the pixel pitch

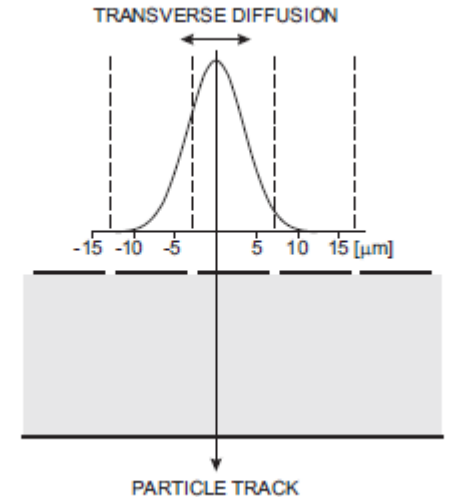
Thermal diffusion drifts the charge with a rms width  $\sigma_y = \sqrt{2Dt}$

Diffusion constant is related to mobility by Einstein relations  $D = \frac{k_B T}{e} \mu$

Using the average field approximation  $\bar{E} = V/d$  and drift velocity  $v_d = \mu \bar{E}$

$$\sigma_y = \sqrt{2 \frac{k_b T}{e} \frac{d^2}{V}}$$

For a sensor 300  $\mu\text{m}$  thick,  $T = 300\text{K}$  and  $V = 100\text{V}$ , transverse diffusion is about  $7\mu\text{m}$



**Spread of charge not a bad thing! Sub-pixel resolution possible by interpolation**

# Detector performance Metrics: Energy Resolution

Mean ionization energy exceeds band gap because conservation of momentum requires excitation of lattice vibration.

Radiation interactions in semiconductor produces two things: Ionization and formation of electric charge pairs and lattice excitation (phonon production)

Assume that in the course of energy deposition

$N_x$  excitations produce  $N_P$  phonons and

$N_i$  ionization interactions form  $N_Q$  charge pairs.

On the average, the sum of the energies going into excitation and ionization is equal to the energy deposited by the incident radiation

$$E_0 = E_i N_i + E_x N_x$$

Assuming Gaussian statistics the variance in number of excitation/ionizations:  $\sigma_x = \sqrt{N_x}$  and  $\sigma_i = \sqrt{N_i}$

Now for a single event (single pulse) the deposited energy is constant. Hence a fluctuation in excitation must be balanced by an equivalent fluctuation in ionization.

$$E_x \Delta N_x + E_i \Delta N_i = 0$$

If for an event more energy goes to charge formation means less energy for excitation. Averaging over several events this implies that the variance in energy for the two process must be equal:

$$E_i \sigma_i = E_x \sigma_x$$

$$\sigma_i = \frac{E_x}{E_i} \sqrt{N_x}$$

# Detector performance Metrics: Energy Resolution

From the total energy  $E_i N_i + E_x N_x = E_0$

$$N_x = \frac{E_0 - E_i N_i}{E_x}$$

yielding

$$\sigma_i = \frac{E_x}{E_i} \sqrt{\frac{E_0}{E_x} - \frac{E_i}{E_x} N_i}$$

Since each ionization leads to charge pair that is part of the signal

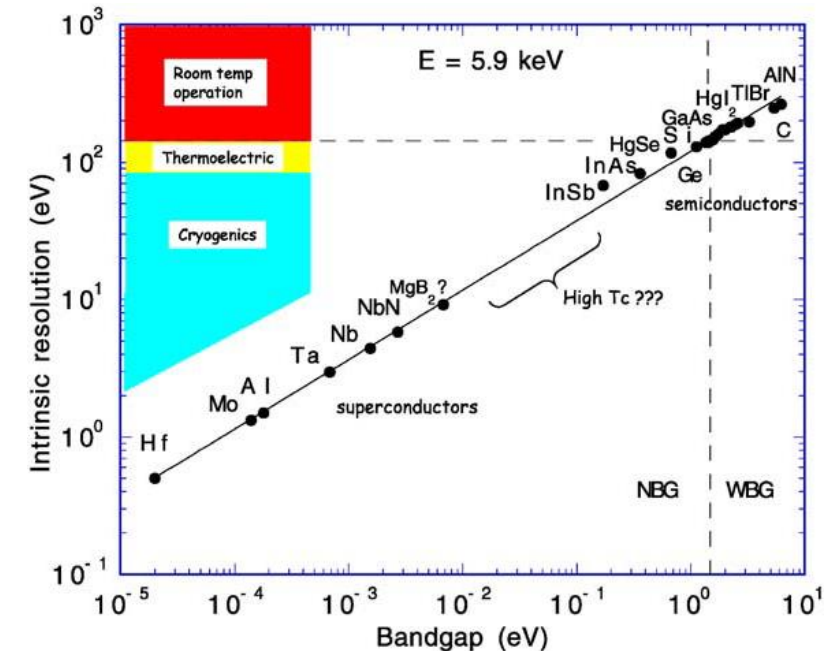
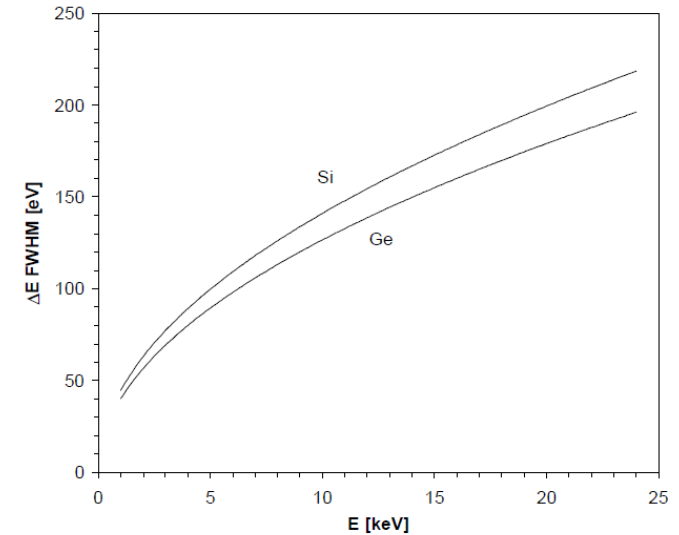
$$N_i = N_Q = \frac{E_0}{E} \quad \longrightarrow \quad \sigma_i = \sqrt{\frac{E_0}{E}} \cdot \sqrt{\frac{E_x}{E_i} \left( \frac{E}{E_i} - 1 \right)}$$

Since  $\sigma_i$  is the variance in signal charge  $Q$  and the number of charge pairs is  $N_Q = E_0/E$

$$\sigma_Q = \sqrt{F N_Q}$$

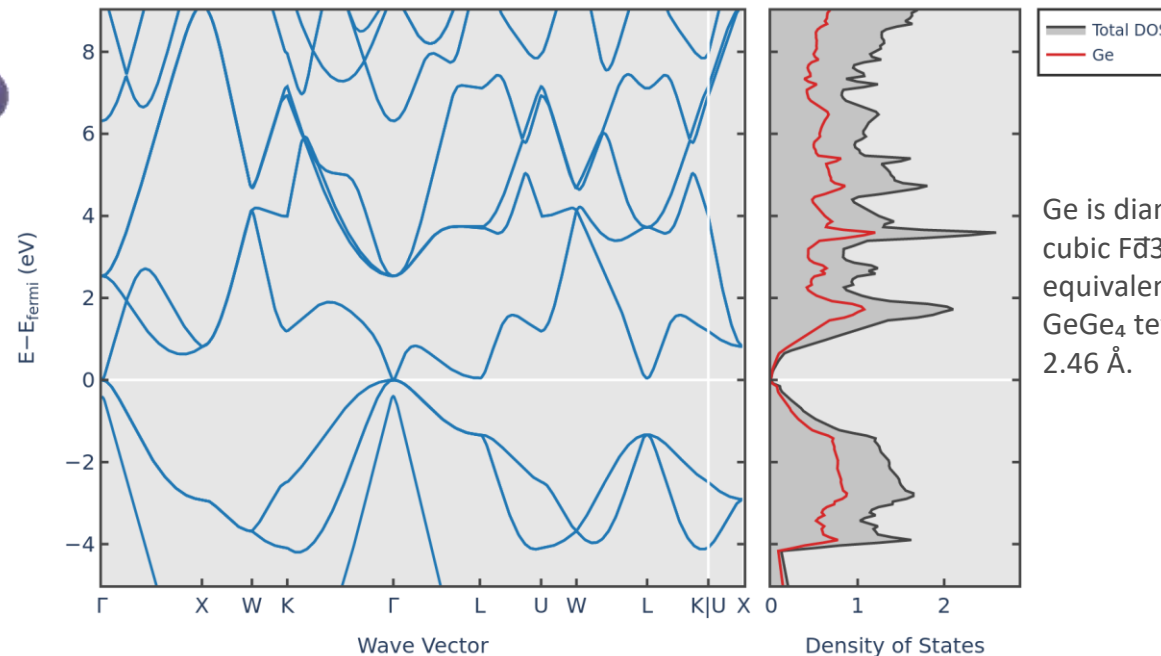
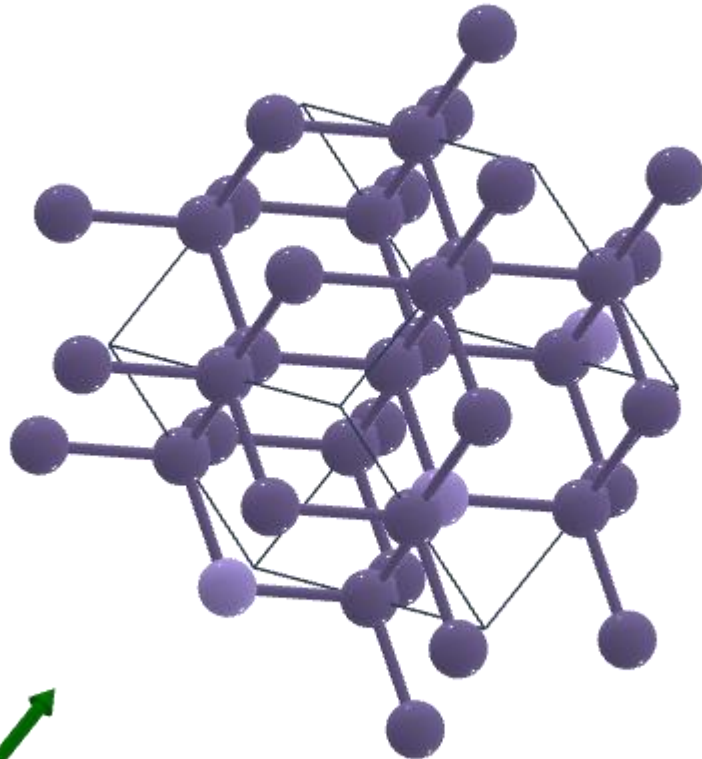
In Silicon  
 $E_x = 0.037 \text{ eV}$   
 $E_i = E_g = 1.1 \text{ eV}$   
 $E = 3.6 \text{ eV}$

Intrinsic Resolution of Si and Ge Detectors



# Si Vs Ge

In the following we compare material properties of silicon and germanium, and discuss how these material properties affect the electrical and optical performance



Ge is diamond structured and crystallizes in the cubic  $F\bar{d}3m$  space group. Ge is bonded to four equivalent Ge atoms to form corner-sharing  $\text{GeGe}_4$  tetrahedra. All Ge-Ge bond lengths are 2.46 Å.





# Properties of Si, Ge, and GaAs

Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.

Properties	Si	Ge	GaAs
Atoms/cm <sup>3</sup>	$5.02 \times 10^{22}$	$4.42 \times 10^{22}$	$4.42 \times 10^{22}$
Atomic weight	28.09	72.6	144.63
Breakdown field (V/cm)	$\sim 3 \times 10^5$	$\sim 1 \times 10^5$	$\sim 4 \times 10^5$
Crystal structure	Diamond	Diamond	Zincblende
Density (g/cm <sup>3</sup> )	2.329	5.326	5.317
Dielectric constant	11.9	16.0	13.1
Effective density of states in conduction band, $N_c$ (cm <sup>-3</sup> )	$2.86 \times 10^{19}$	$1.04 \times 10^{19}$	$4.7 \times 10^{17}$
Effective density of states in valance band, $N_v$ (cm <sup>-3</sup> )	$1.04 \times 10^{19}$	$6.0 \times 10^{18}$	$7.0 \times 10^{18}$
Optical phonon energy (eV)	0.063	0.037	0.035
Effective mass (conductivity)			
Electrons ( $m_n/m_0$ )	0.26	0.082	0.067
Holes ( $m_p/m_0$ )	0.69	0.28	0.57
Electron affinity, $\chi$ (V)	4.05	4.0	4.07
Energy gap (eV)	1.12	0.67	1.42
Intrinsic carrier concentration (cm <sup>-3</sup> )	$1.45 \times 10^{10}$	$2.4 \times 10^{13}$	$1.8 \times 10^6$
Intrinsic resistivity ( $\Omega$ -cm)	$2.3 \times 10^5$	47	$10^9$
Lattice constant (Å)	5.431	5.646	5.653
Melting point (°C)	1415	937	1240
Minority carrier lifetime (s)	$2.5 \times 10^{-3}$	$10^{-3}$	$\sim 10^{-8}$
Mobility (cm <sup>2</sup> /V·s)			
Electron ( $\mu_n$ )	1,500	3900	8,500
Holes ( $\mu_p$ )	450	1900	450
Thermal diffusivity (cm <sup>2</sup> /s)	0.9	0.36	0.24
Thermal conductivity (W/cm·°C)	1.5	0.6	0.46

Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.

Properties	Si	Ge	GaAs
Atoms/cm <sup>3</sup>	5.02 × 10 <sup>22</sup>	4.42 × 10 <sup>22</sup>	4.42 × 10 <sup>22</sup>
Atomic weight	28.09	72.6	144.63
Breakdown field (V/cm)	~ 3 × 10 <sup>5</sup>	~ 1 × 10 <sup>5</sup>	~ 4 × 10 <sup>5</sup>

Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.		
Properties	Si	Ge
Energy gap (eV)	1.12	0.67

Effective mass (conductivity)			
Electrons (m <sub>n</sub> /m <sub>0</sub> )		0.82	0.067
Holes (m <sub>p</sub> /m <sub>0</sub> )		0.28	0.57
Electron affinity, χ (V)		0	4.07
Energy gap (eV)		1.12	1.42
Intrinsic carrier concentration (cm <sup>-3</sup> )		10 <sup>13</sup>	1.8 × 10 <sup>6</sup>
Intrinsic resistivity (Ω-cm)		7	10 <sup>3</sup>
Lattice constant (Å)		3.56	5.653
Melting point (°C)	1415	937	1240
Minority carrier lifetime (s)	2.5 × 10 <sup>-3</sup>	10 <sup>-3</sup>	~ 10 <sup>-8</sup>
Mobility (cm <sup>2</sup> /V·s)			
Electron (μ <sub>n</sub> )	1,500	3900	8,500
Holes (μ <sub>p</sub> )	450	1900	450
Thermal diffusivity (cm <sup>2</sup> /s)	0.9	0.36	0.24
Thermal conductivity (W/cm <sup>2</sup> ·°C)	1.5	0.6	0.46

$$\lambda_{cutoff} \approx \frac{1.24}{E_g}$$

# Si Vs Ge

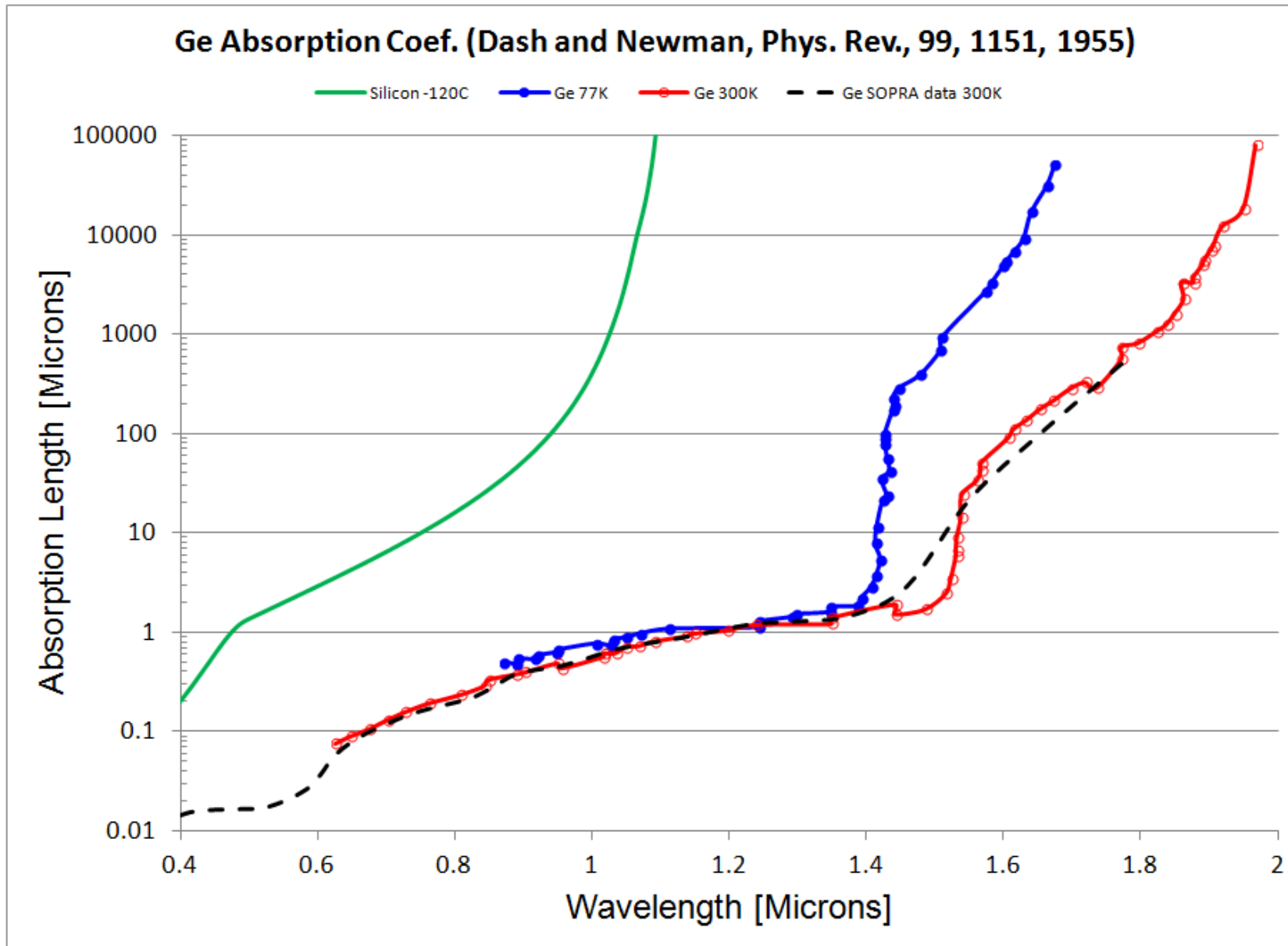


Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.																			
Properties	Si	Ge	GaAs																
Atoms/cm <sup>3</sup>	5.02 × 10 <sup>22</sup>	4.42 × 10 <sup>22</sup>	4.42 × 10 <sup>22</sup>																
Atomic weight	28.09	72.6	144.63																
Breakdown field (V/cm)	~ 3 × 10 <sup>5</sup>	~ 1 × 10 <sup>5</sup>	~ 4 × 10 <sup>5</sup>																
<table border="1"> <thead> <tr> <th colspan="4">Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.</th> </tr> <tr> <th>Properties</th> <th>Si</th> <th>Ge</th> <th>GaAs</th> </tr> </thead> <tbody> <tr> <td>Atomic weight</td> <td>Z = 14 28.09</td> <td>Z = 32 72.6</td> <td></td> </tr> <tr> <td>Density (g/cm<sup>3</sup>)</td> <td>2.329</td> <td>5.326</td> <td></td> </tr> </tbody> </table>				Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.				Properties	Si	Ge	GaAs	Atomic weight	Z = 14 28.09	Z = 32 72.6		Density (g/cm <sup>3</sup> )	2.329	5.326	
Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.																			
Properties	Si	Ge	GaAs																
Atomic weight	Z = 14 28.09	Z = 32 72.6																	
Density (g/cm <sup>3</sup> )	2.329	5.326																	
Holes (m <sub>p</sub> /m <sub>0</sub> )	0.69	0.28	0.57																
Electron																			
Energy																			
Intrinsic																			
Intrinsic																			
Lattice																			
Melting																			
Minority																			
<p style="text-align: center; color: blue; font-weight: bold;">           Ge is 2.3x denser than silicon            Good for <math>\gamma</math> and x-ray detection            Practical issue: wafers are heavy         </p>																			
Mobility (cm <sup>2</sup> /V·s)																			
Electron ( $\mu_n$ )	1,500	3900	8,500																
Holes ( $\mu_p$ )	450	1900	450																
Thermal diffusivity (cm <sup>2</sup> /s)	0.9	0.36	0.24																
Thermal conductivity (W/cm·°C)	1.5	0.6	0.46																

# Si Vs Ge

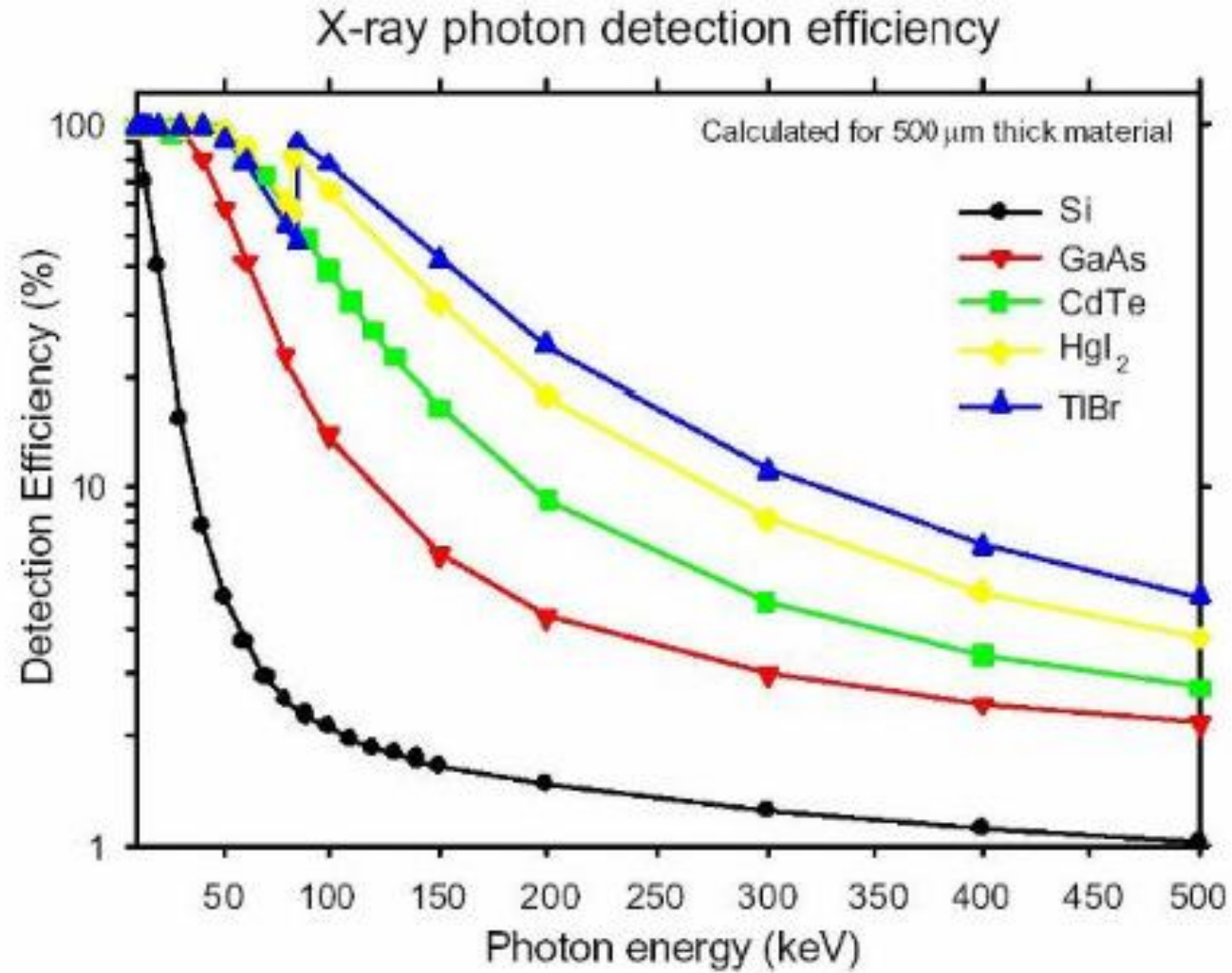




Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.

Properties	Si	Ge	GaAs
Atoms/cm <sup>3</sup>	5.02 × 10 <sup>22</sup>	4.42 × 10 <sup>22</sup>	4.42 × 10 <sup>22</sup>
Atomic weight	28.09	72.6	144.63
Breakdown field (V/cm)	~ 3 × 10 <sup>5</sup>	~ 1 × 10 <sup>5</sup>	~ 4 × 10 <sup>5</sup>

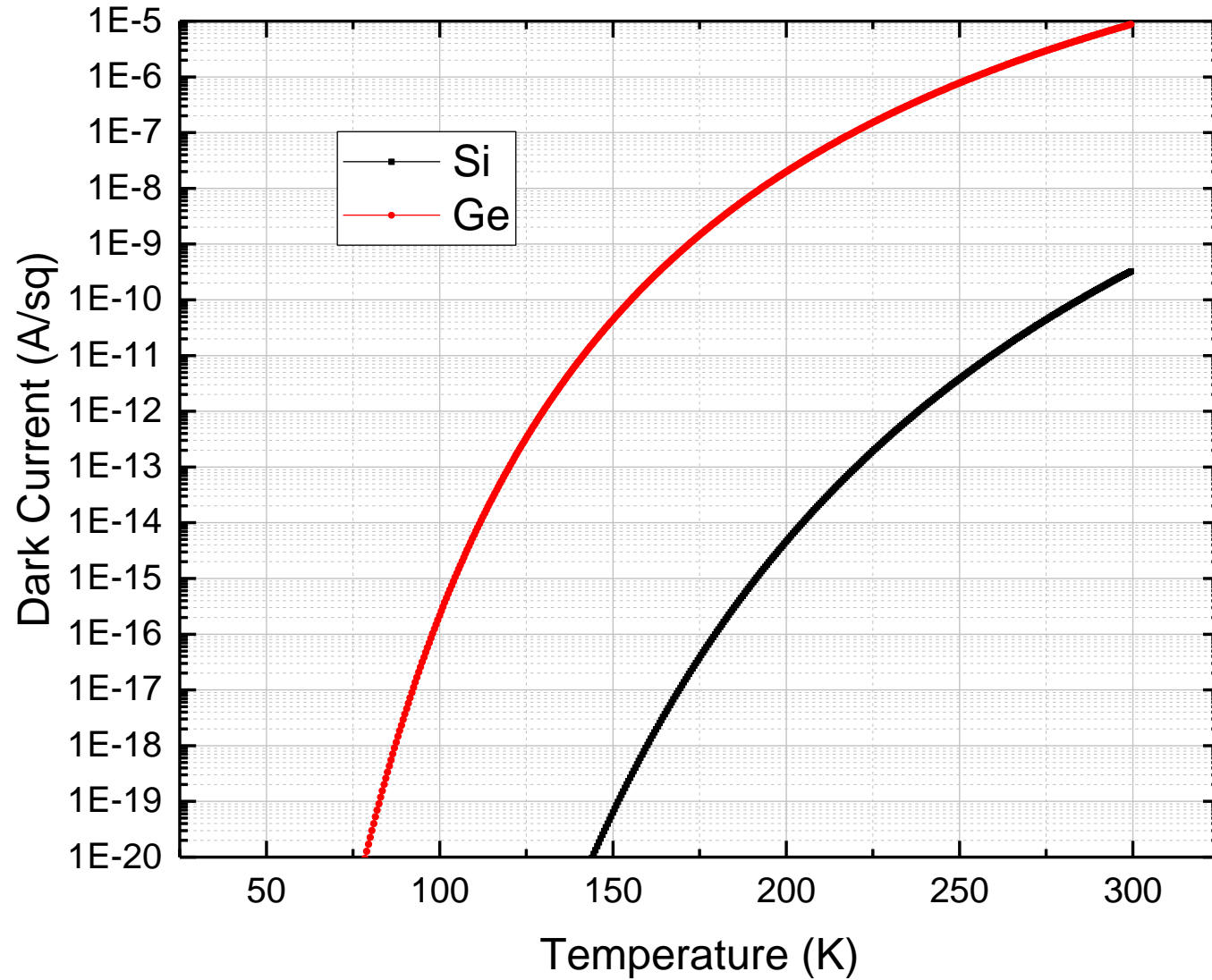
Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.		
Properties	Si	Ge
Energy gap (eV)	1.12	0.67
Intrinsic carrier concentration (cm <sup>-3</sup> )	1.45 × 10 <sup>10</sup>	2.4 × 10 <sup>13</sup>
Minority carrier lifetime (s)	2 × 10 <sup>-3</sup>	10 <sup>-3</sup>

	Si	Ge	GaAs
Holes (m <sub>p</sub> /m <sub>0</sub> )	0.69	0.28	0.57
Electron (m <sub>n</sub> /m <sub>0</sub> )	0.36	0.22	0.47
Energy gap (eV)	1.12	0.67	1.42
Intrinsic carrier concentration (cm <sup>-3</sup> )	1.45 × 10 <sup>10</sup>	2.4 × 10 <sup>13</sup>	2.2 × 10 <sup>19</sup>
Intrinsic carrier lifetime (s)	2 × 10 <sup>-3</sup>	10 <sup>-3</sup>	10 <sup>-7</sup>
Lattice constant (Å)	357	357	353
Melting point (°C)	1414	938	1238
Minority carrier lifetime (s)	2 × 10 <sup>-3</sup>	10 <sup>-3</sup>	10 <sup>-7</sup>
Mobility (cm <sup>2</sup> /V·s)			
Electron (μ <sub>n</sub> )	1,500	3900	8,500
Holes (μ <sub>p</sub> )	450	1900	450
Thermal diffusivity (cm <sup>2</sup> /s)	0.9	0.36	0.24
Thermal conductivity (W/cm·°C)	1.5	0.6	0.46

$$J_{leakage} \propto \frac{n_i}{\tau_g} \propto \frac{\exp(-E_g/2kT)}{\tau_g}$$

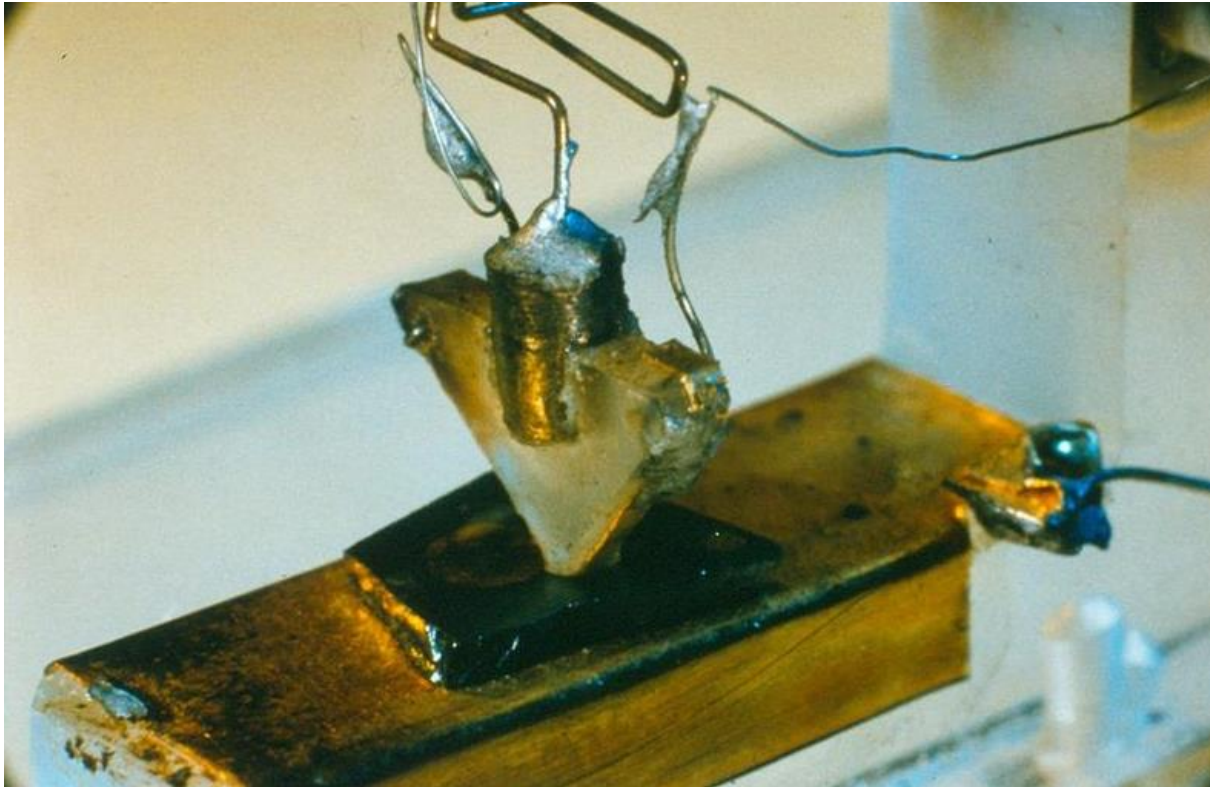
# Si Vs Ge

Ideal dark current for Si Vs Ge, assumption Si forced to  $\sim 1$  nA/Sq, and same  $\tau_g$

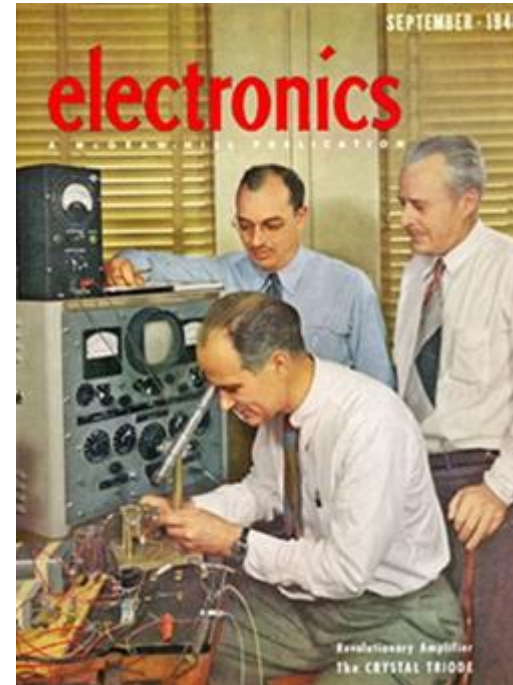


**Needs Cryo-cooling !**

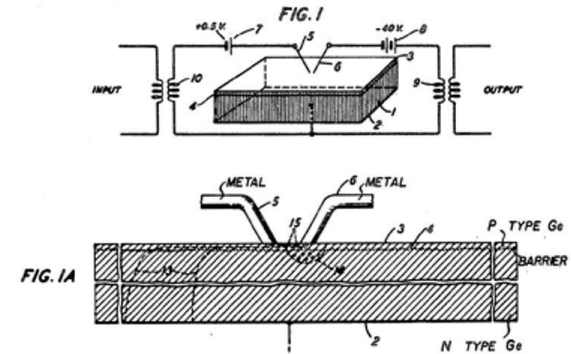
# Germanium: First transistor



First Transistor: Bardeen, Brattain, and Shockley, Bell Labs, 1947



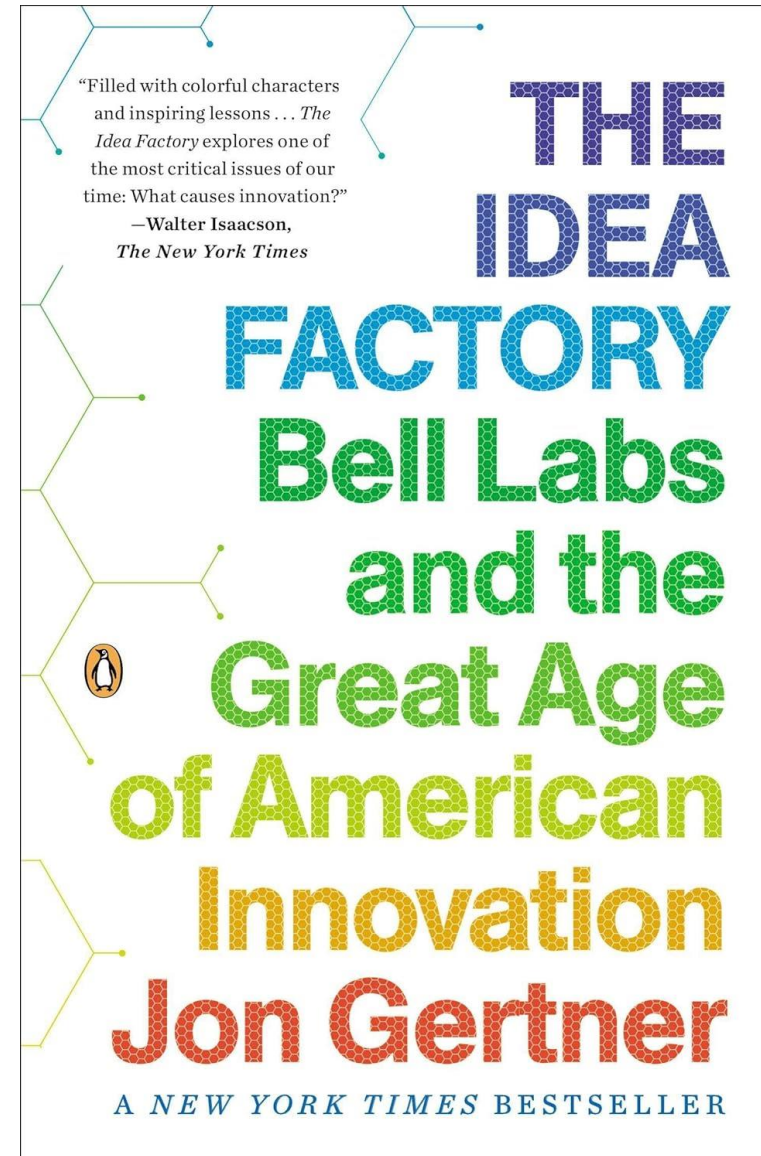
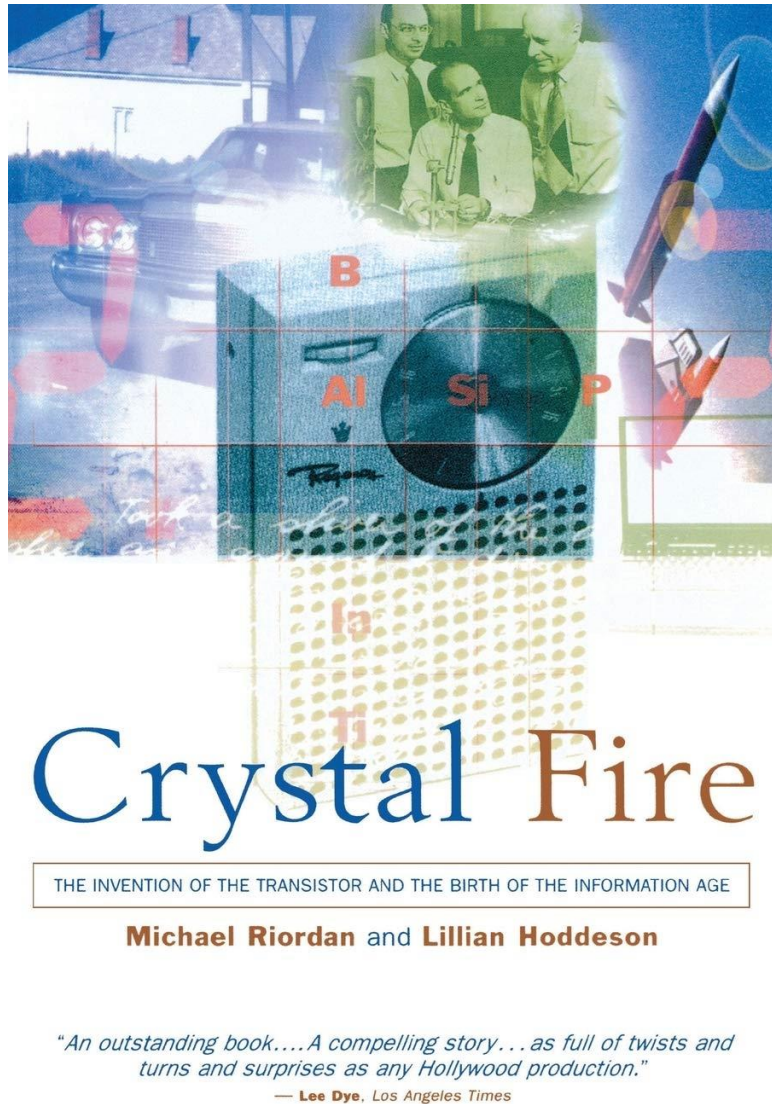
Oct. 3, 1950 J. BARDEEN ET AL 2,524,035  
THREE-ELECTRODE CIRCUIT ELEMENT UTILIZING SEMICONDUCTIVE MATERIALS  
Filed June 17, 1948 3 Sheets-Sheet 1



- Germanium was easy to purify because of lower melting point
- Not until 1954, Texas Instruments demonstrated the silicon grown junction transistor



# Brief history of transistor

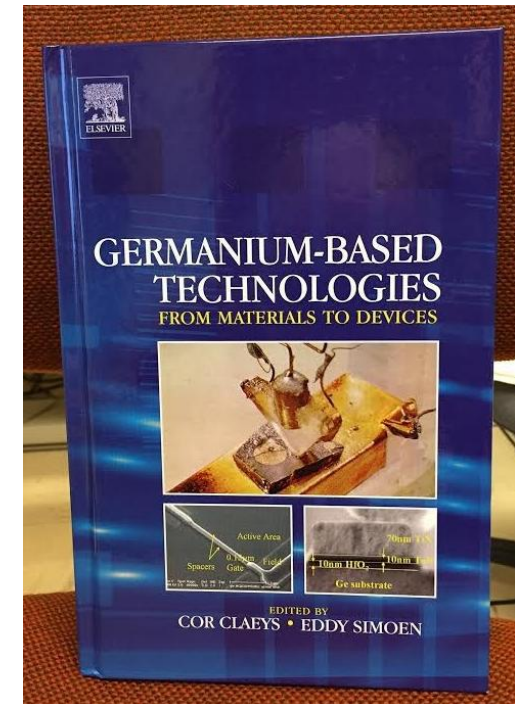
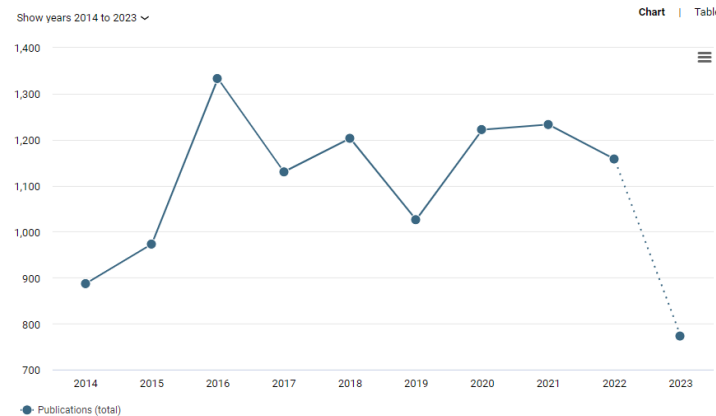


# Si Vs Ge

Table 1. Comparison of some basic properties of Si, Ge, and GaAs at 300 K.

Properties	Si	Ge
Mobility ( $\text{cm}^2/\text{V}\cdot\text{s}$ )		
Electron ( $\mu_n$ )	1,500	3900
Holes ( $\mu_p$ )	450	1900

- Germanium has high carrier mobilities
- Ge has the largest hole mobility of any semiconductor
  - Renewed interest in Ge due to high-mobilities





# Ge Issues

- Technology not as mature as Si
- Key to devices (and detectors) are field oxides
- GeO<sub>2</sub> is unstable (water soluble) and thermally unstable at temperatures normally employed for device processing.
- High-K dielectrics?
- Capping of Ge/GeO<sub>2</sub> with other oxides

## Brattain and Bardeen first test

He spent Friday, December 12, poking around at the other four spots on the germanium, testing their resistance to current flow in either direction. Again, they all seemed to be making good contact with the surface. Where was the oxide layer? He gradually began to realize that he had inadvertently washed it off before depositing the gold. "The germanium oxide formed by an anodic process is soluble in water," he recalled, "and when I washed the glycol borate off, I washed the oxide film off!"

Brattain was pleased, too. He mentioned it to his old Whitman College buddy Walker Bleakney, who had known Bardeen at Princeton. "You'll find that Bardeen doesn't very often open his mouth to say anything," he warned Brattain. "But when he does, YOU LISTEN!"

## Stabilization of the GeO<sub>2</sub>/Ge Interface by Nitrogen Incorporation in a One-Step NO Thermal Oxynitridation

Gabriela Copetti,<sup>†</sup> Gabriel V. Soares,<sup>†</sup> and Cláudio Radtke<sup>\*,‡</sup>

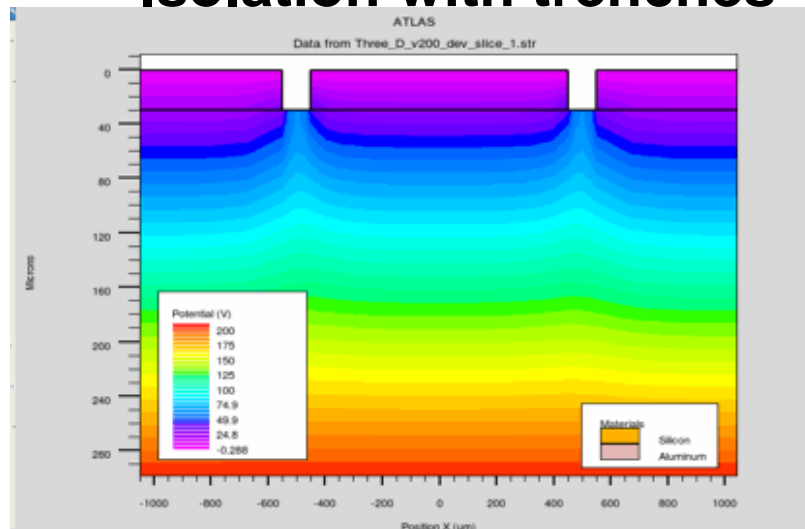
<sup>†</sup>Instituto de Física and <sup>‡</sup>Instituto de Química, UFRGS, 91509-900 Porto Alegre, Brazil

However, the lack of a stable passivation layer for Ge surface hinders the development of such technology. Unlike silicon dioxide (SiO<sub>2</sub>), germanium dioxide (GeO<sub>2</sub>) is water-soluble and also thermally unstable at temperatures usually employed during device processing.<sup>2</sup> This instability is due to the interfacial reaction  $\text{GeO}_2 + \text{Ge} \rightarrow 2\text{GeO}$  that occurs at temperatures greater than 400 °C. Oxygen vacancies generated at the GeO<sub>2</sub>/Ge interface diffuse through the oxide toward the surface, where they promote GeO desorption (as evidenced by thermal desorption spectroscopy<sup>3</sup>), leading to the deterioration of the device's electrical properties.<sup>4</sup> First-principles calculations predicted that

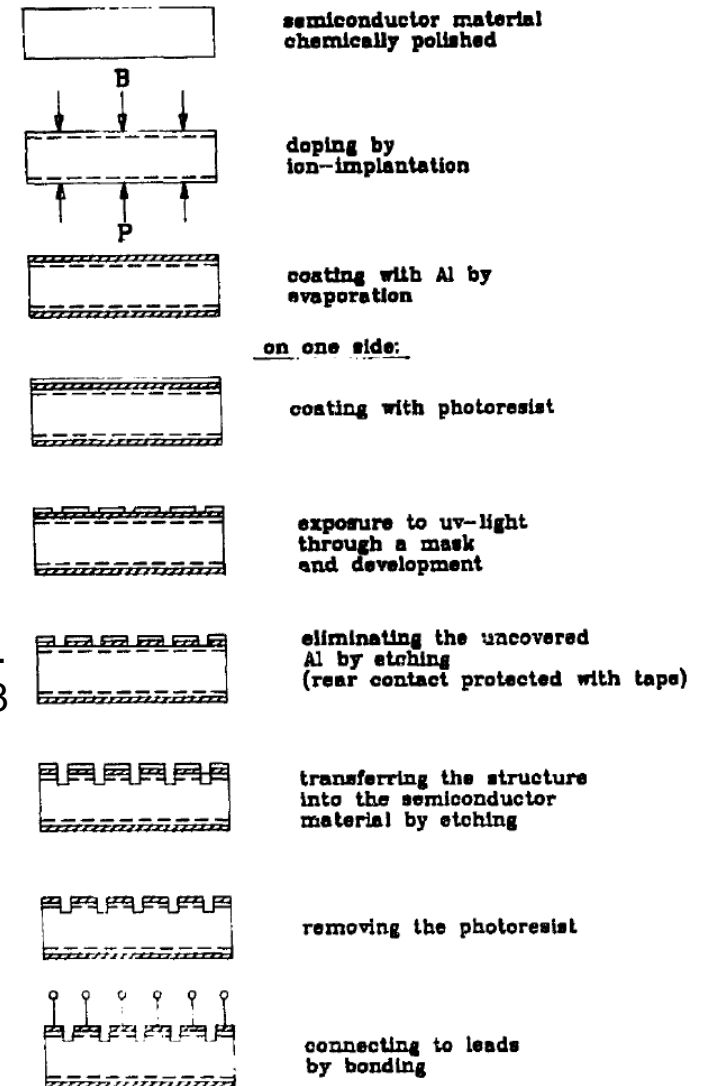
# Pixel Isolation with Trenches

- Sensors for radiation detections are in micron scale (not nm!)
- Simple devices can be fabricated with few masks
- Pixels isolation with physical trenches

## Isolation with trenches

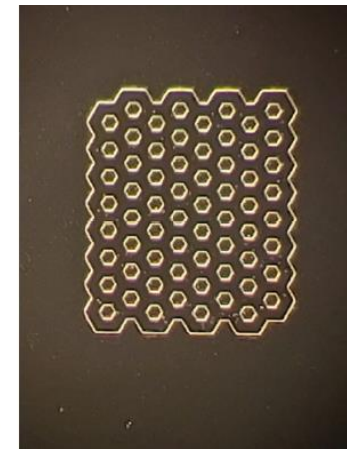
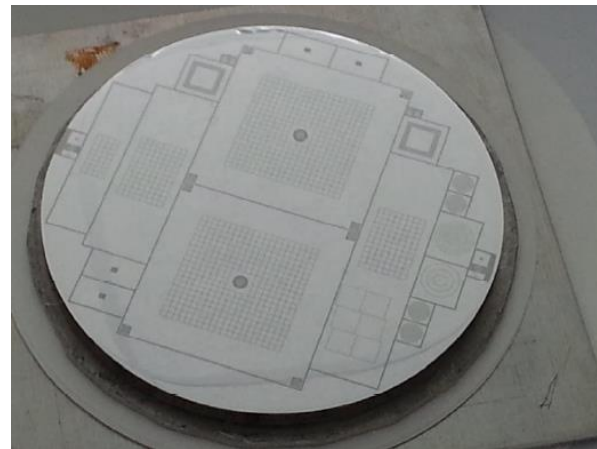
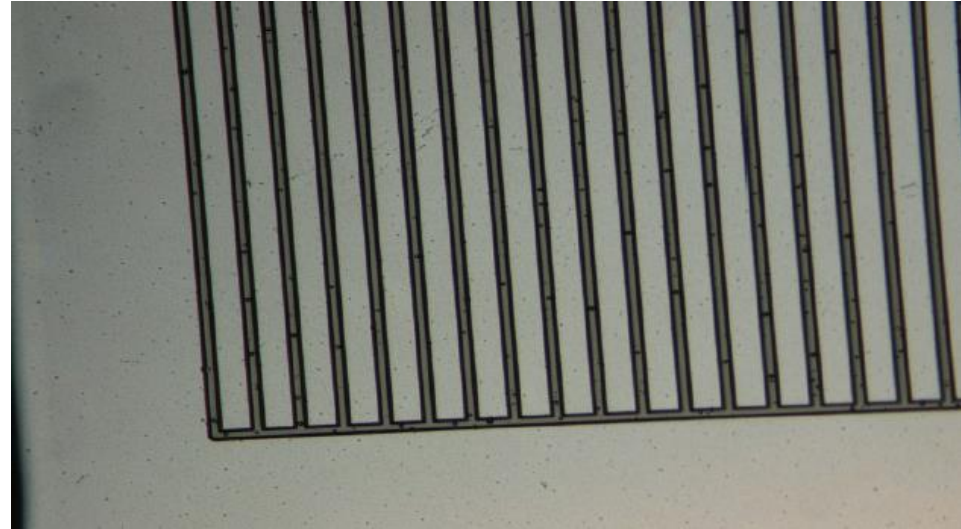


A.Hamacher et al. Nucl.Ins.  
Meth.Phys. Res. **A295**, 128  
(1990)



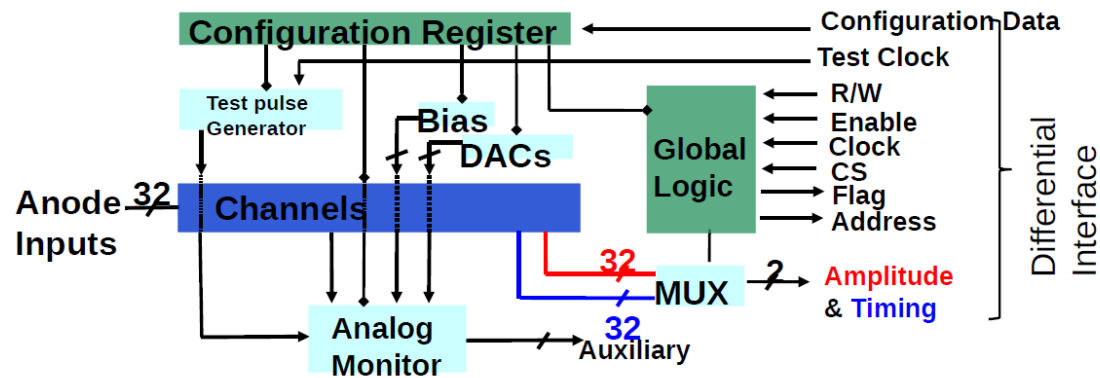
## Ge sensors with trenches

- P-type layer formed by ion implantation.
- N-type contact formed by diffusion.
- Implanted layer formed into pixels by etching trenches between them.
- Pixels become electrically isolated upon full depletion.



# ASIC: MARS, a new spectroscopy chip

- MARS has 32 low-noise channels, each with its own peak detector and timing circuitry
- Charge-sensitive preamplifier capable of reading electrons or holes, with 4 gain settings from 3600mV/fC to 600mV/fC (12.5keV – 75keV F.S.)
  - Shaper with 4 time constants from 0.25 us to 2 us.
  - Per-channel peak detector and timing generator. Timing generator can measure time-over-threshold (for pileup detection) or time-of-arrival (for detecting charge-shared events)
  - All external signals are fully differential to preserve the system's low noise

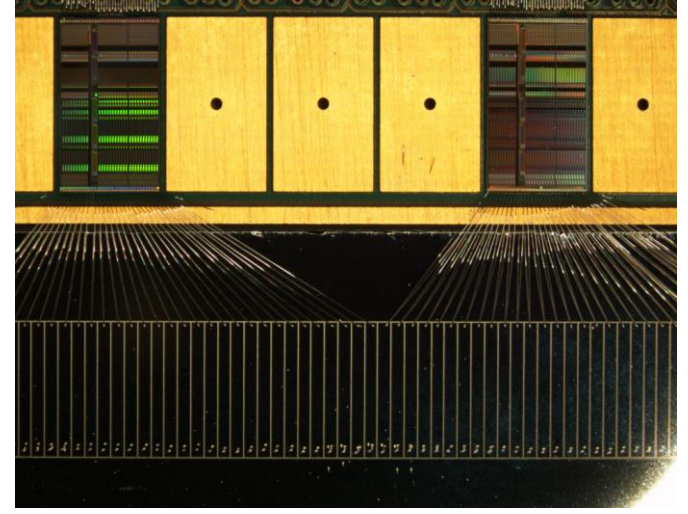
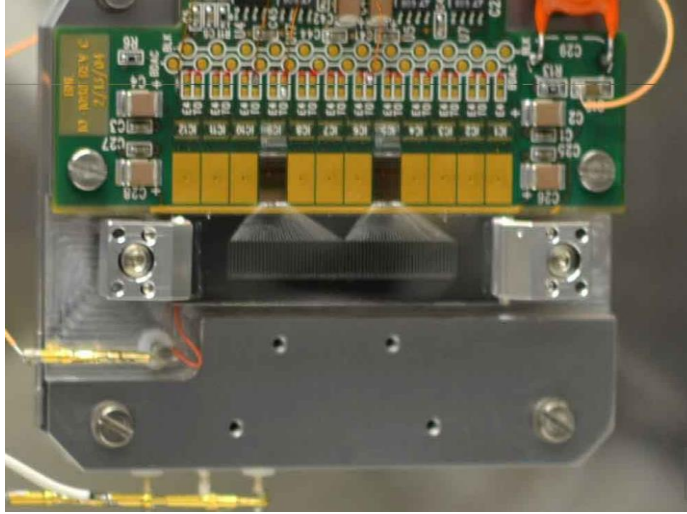
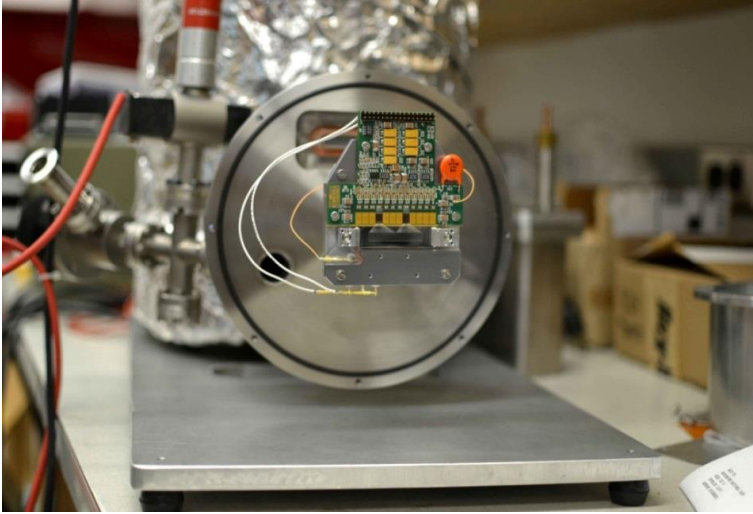


32 readout channels

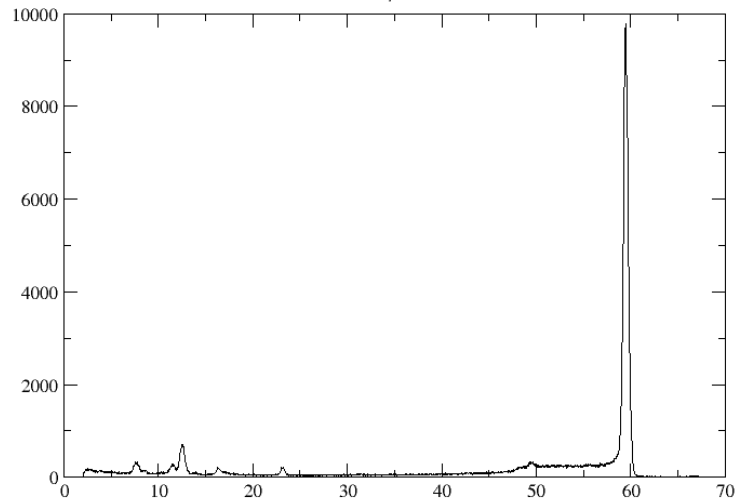
- Amplitude and timing measurement per channel
- LVDS interface, analog differential outputs
- Test pulse generator
- Multiplexed analog monitors



# Multi-element Germanium prototype: 64-strip



Am241 energy spectrum  
Strip #29



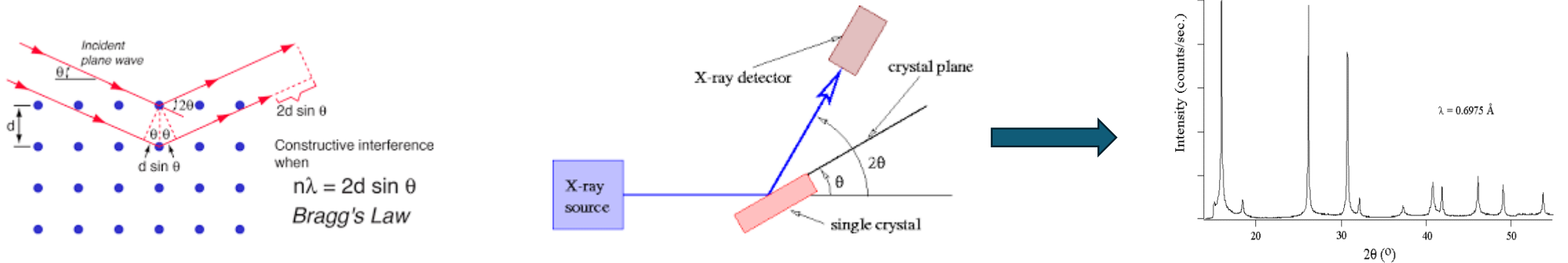
Energy resolution:  
~450eV @ 60keV



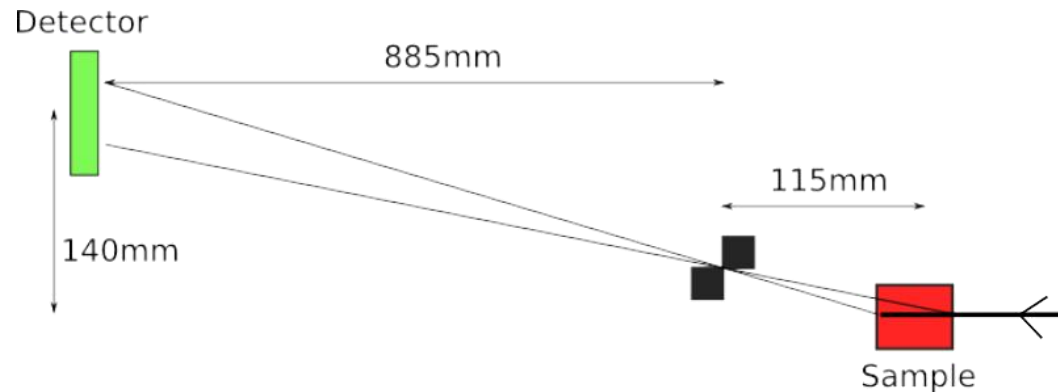


# Multi-element Germanium prototype: what do we do with it?

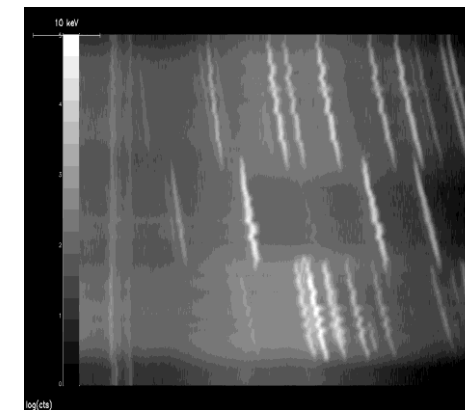
Recall: Bragg's law



## Energy Dispersive X-ray Diffraction with 64-Ge Strip Detector

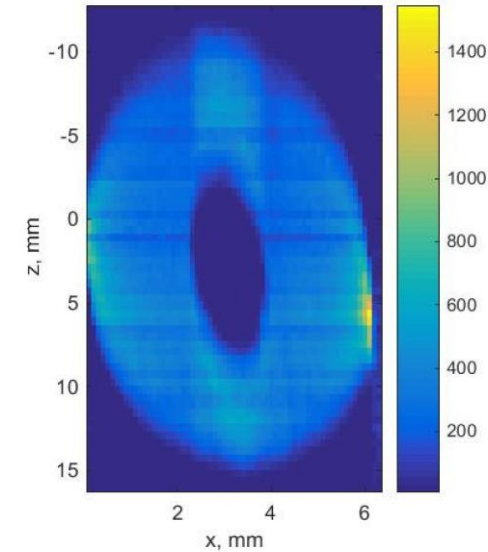
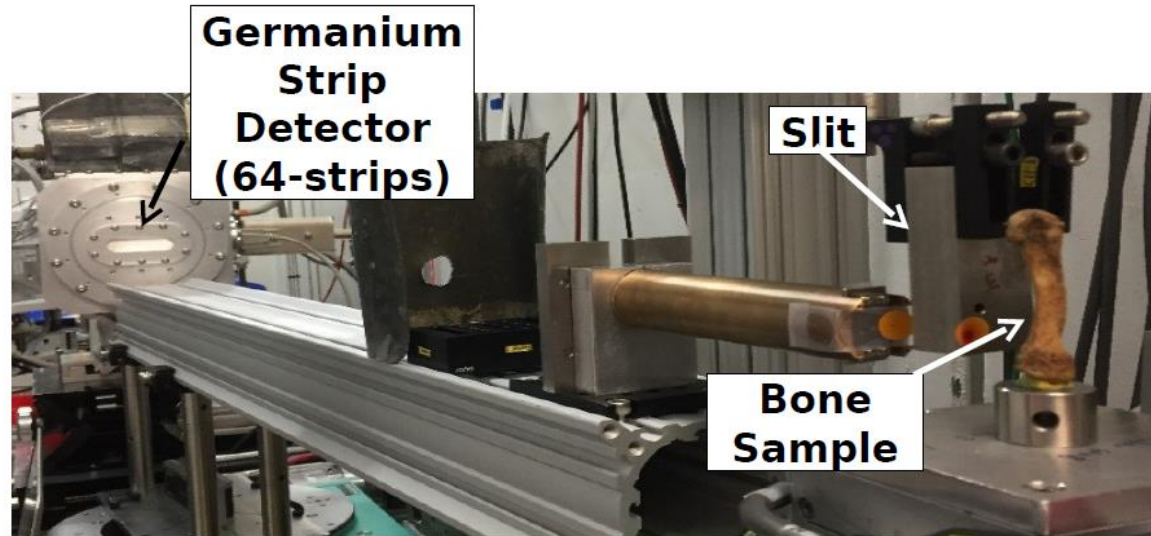


- 100um white beam enters from right
- Slit projects scattered beams onto strip detector
- Each detector element collects spectra (i.e. diffraction pattern) from one region of sample.



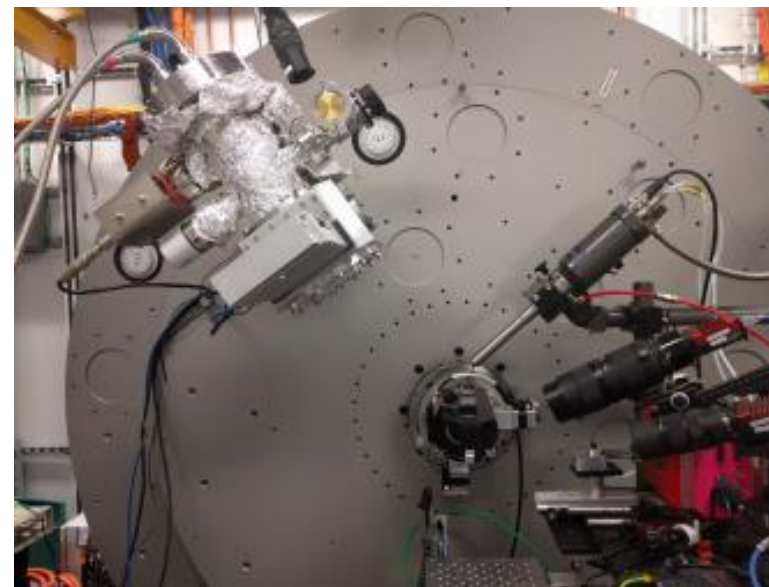
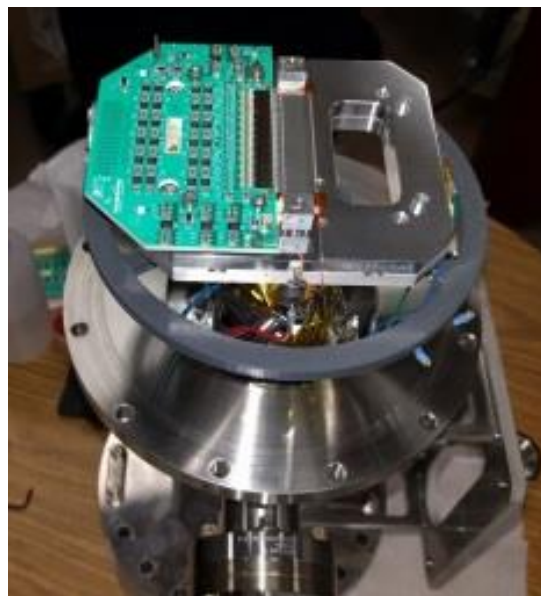
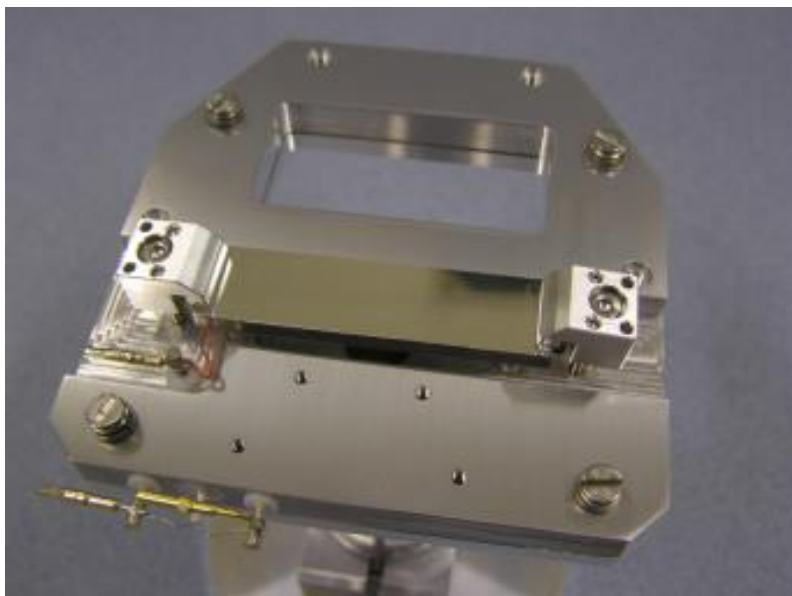
# Multi-element Germanium prototype: what do we do with it?

## Energy dispersive diffraction imaging



- The energy dispersive real space reconstruction of a hydroxyapatite (hAp) bone phantom is shown to the left.
- Several diffraction peaks were reconstructed and peak intensity from one is shown here (hAp 004 reflection).
- The shape including empty central region is correctly reconstructed.
- Peak centers from each phase present can also be reconstructed to yield strain information.

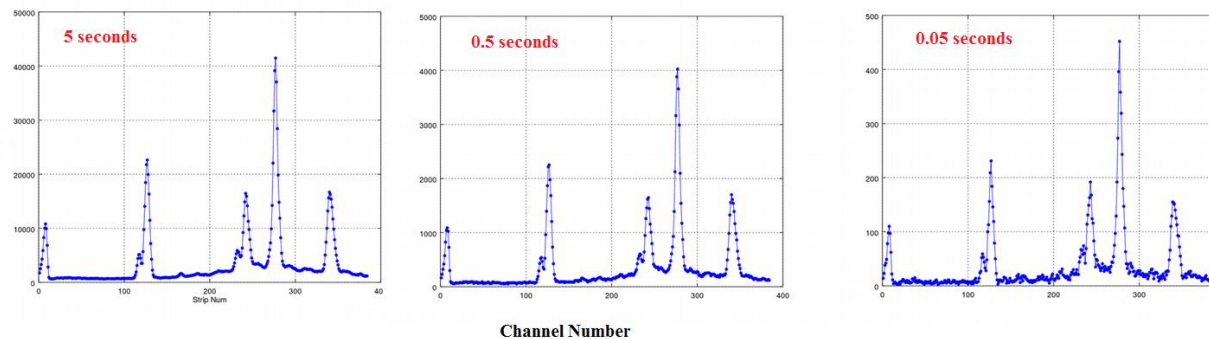
# 384-element detector: XPD beamline, NSLS II



For this application we built a 384-strip detector

- 384 strips, 0.125 mm x 8mm
- 3mm thick
- 12 ASICs with full parallel readout

Time-stamping events enables dynamic measurements

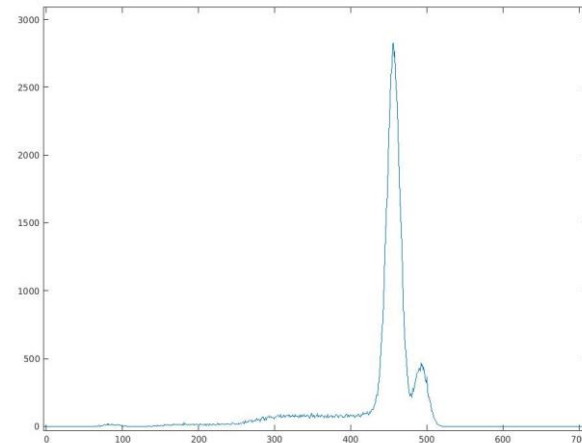
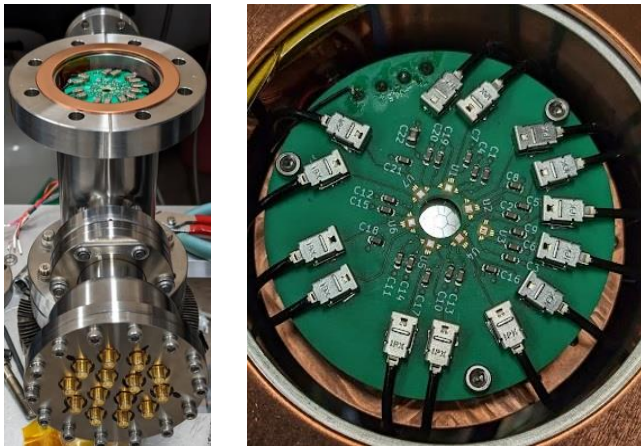


# **New Germanium detectors at BNL: Present**



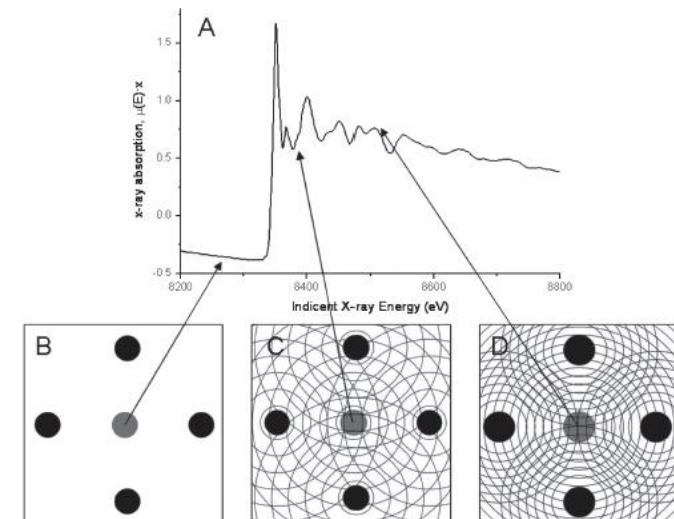
# 7 Channel Ge pixel with low-noise readout

- Energy resolving Silicon drift sensor is the detector of choice for XAFS beamline. However, are very inefficient above 15 KeV
- High-purity Germanium (HPGe) on the other hand offers excellent energy resolution and detection efficiency for high-energy X-ray.
- Problem of Si escape peak being close to the absorption edge can be resolved with Ge ( $K_{\alpha} \sim 9.8$  KeV).



Energy resolution  $\sim 260$  eV  
Cube PRE\_39  $\sim 20$  e ENC@  $1 \mu$ s

## Extended X-ray Absorption Fine Structure (EXAFS)

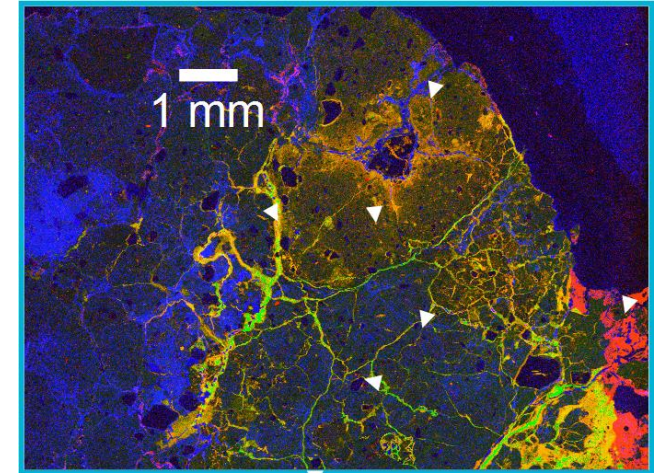
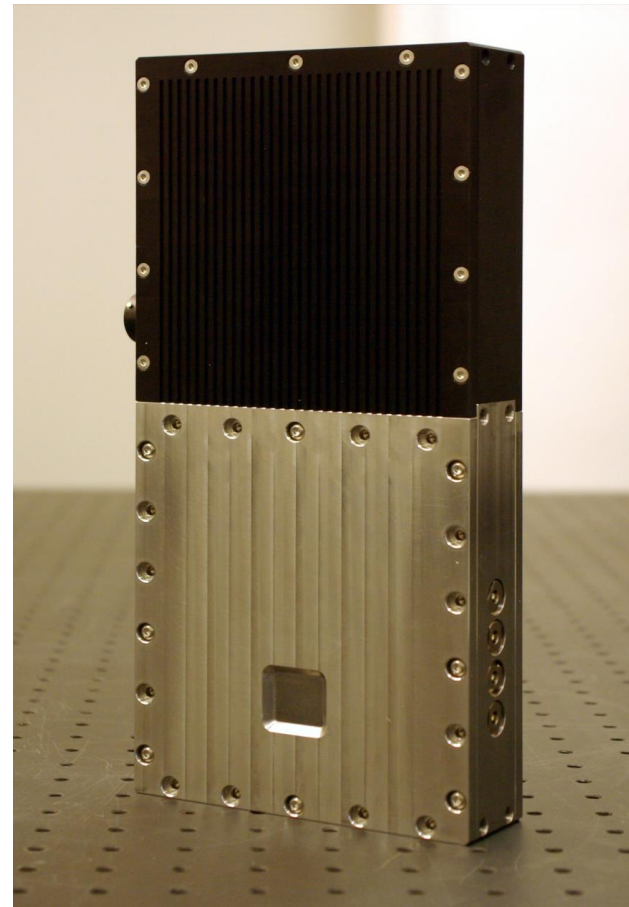
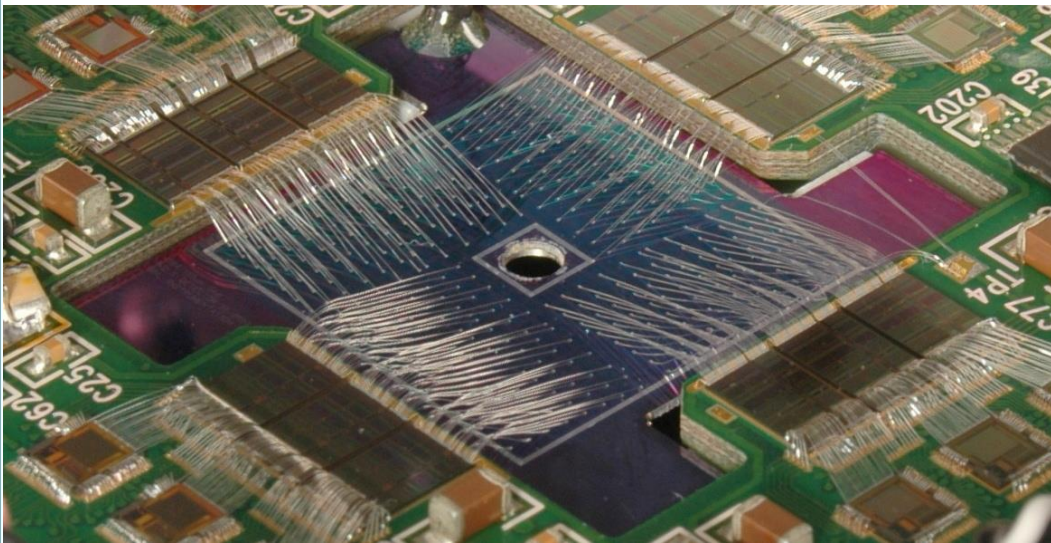


S. D. Shelly et al. in Methods of Soil Analysis: Part 5  
(2008) Soil Science Society of America

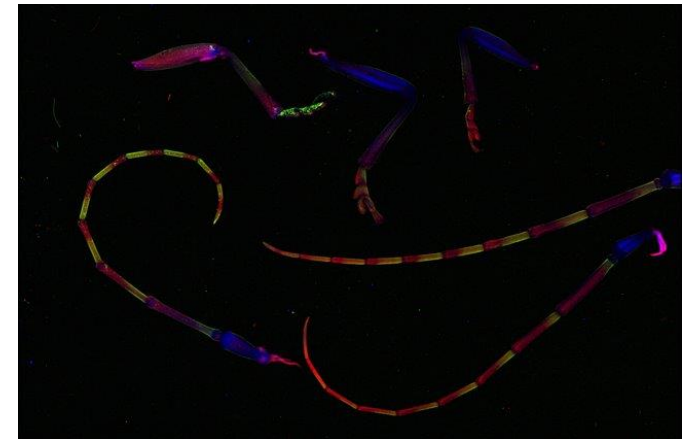


# Fluorescence Imaging with MAIA Detector

Traditionally done with Silicon drift array. Scan over larger area is time consuming  
MAIA (384 p-i-n diodes) on the fly scanning: Faster scan time.  
Detector limits the energy that can be scanned (less than 20 KeV)  
Rely on *L* lines for high-Z materials



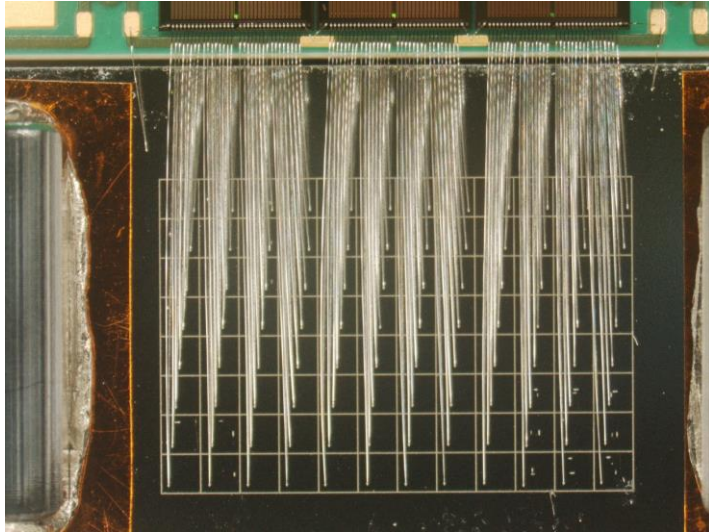
Rob Hough, James Cleverley, CSIRO, private communication



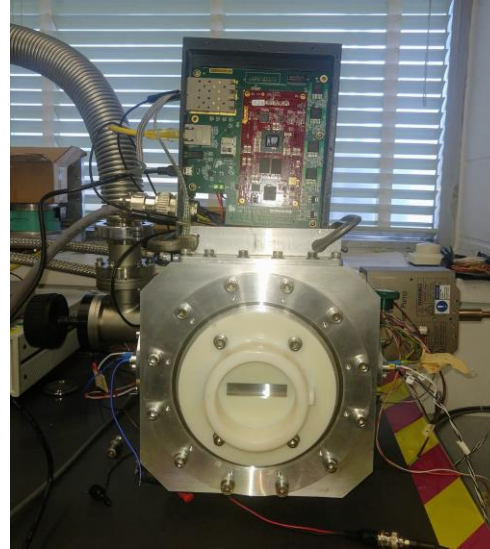


# 96- pixel Germanium detector (Ge version of Maia: Gaia)

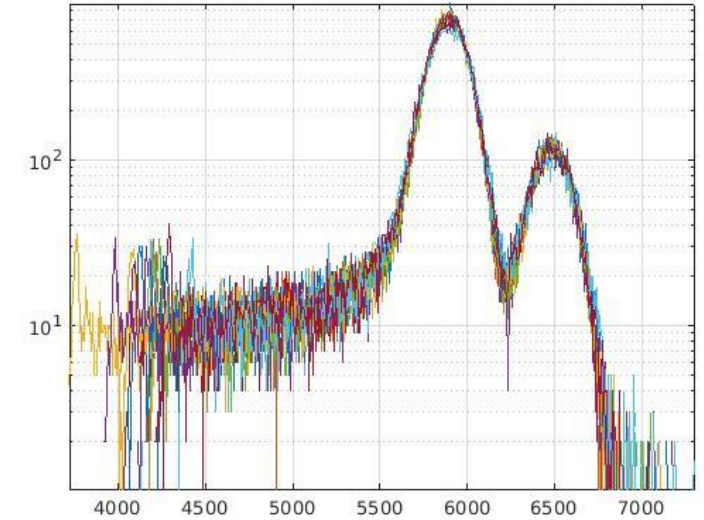
96-germanium pixel sensor wire bonded to HE-MARS



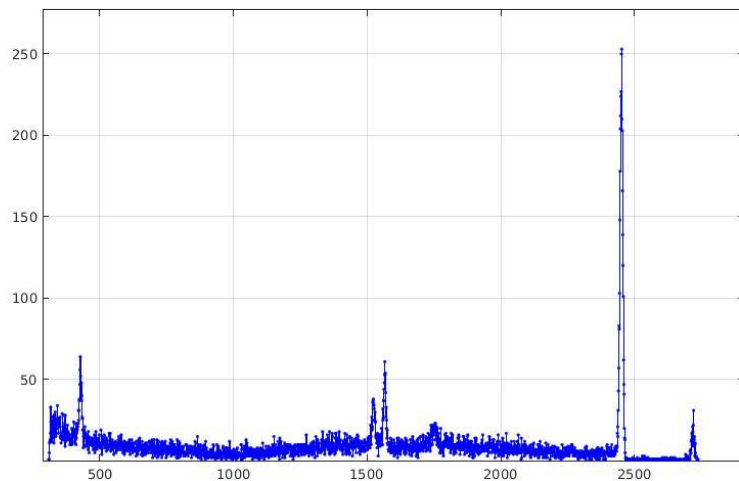
Complete detector system with closed cycle cryostat



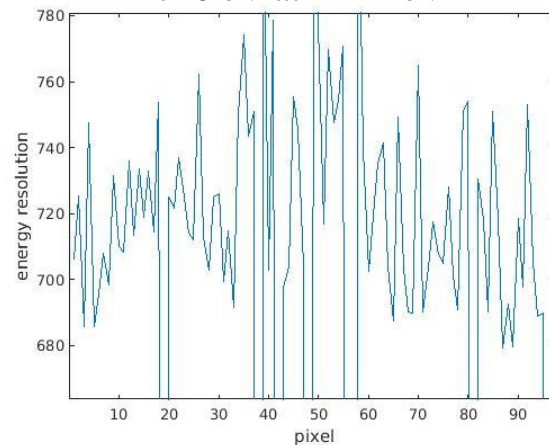
Response from Fe55  
Resolution ~280 eV



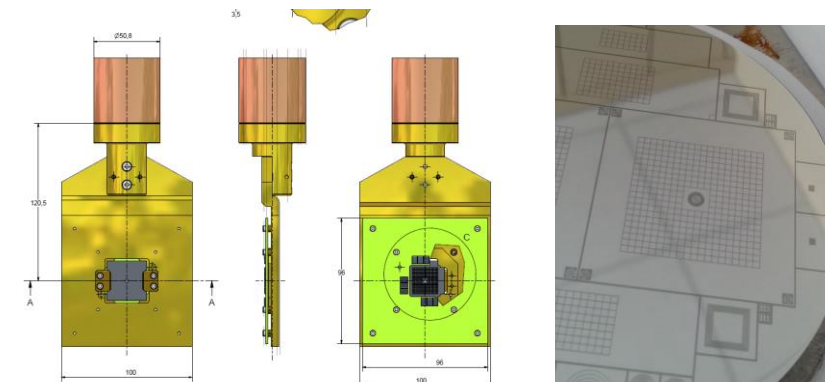
Single pixel response from Co57



Resolution from Co 57  
~720 eV at 122 KeV

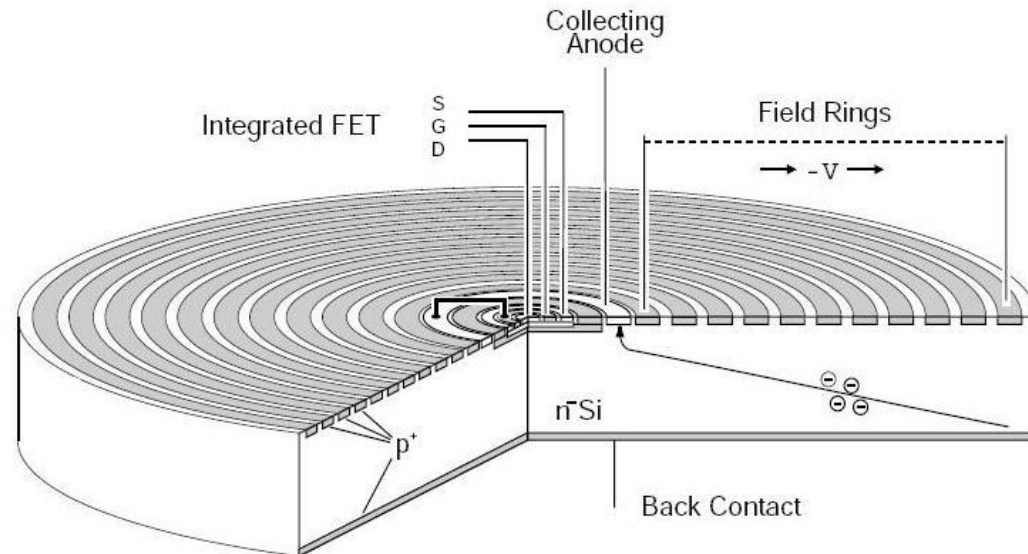


Plan for 384-Germanium Maia detector  
Left: drawing of the sensor mount and cold head  
Right: 384-Gaia sensors



# Germanium Drift Detector (GDD)

- Silicon drift detector offers the best energy resolution. Invented by Gatti and Rehak at BNL.
- Uses long lifetime and high mobility of charges in depleted silicon to reduce capacitance of collection electrode.
- Planar Germanium version of drift detector not made yet. Fabrication challenges



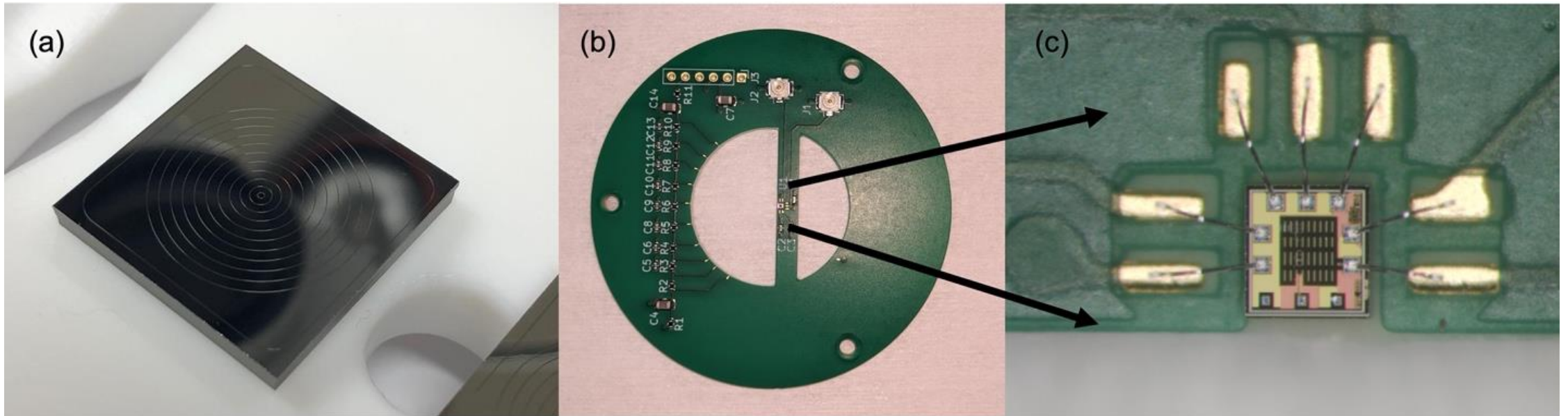
# Germanium Drift Detector (GDD)

Cryostat and sensor cooling design currently underway

Read out with Cube pre-amp.

Sensor characterization would be done with cryogenic probe station

**Stay tuned for more results!**

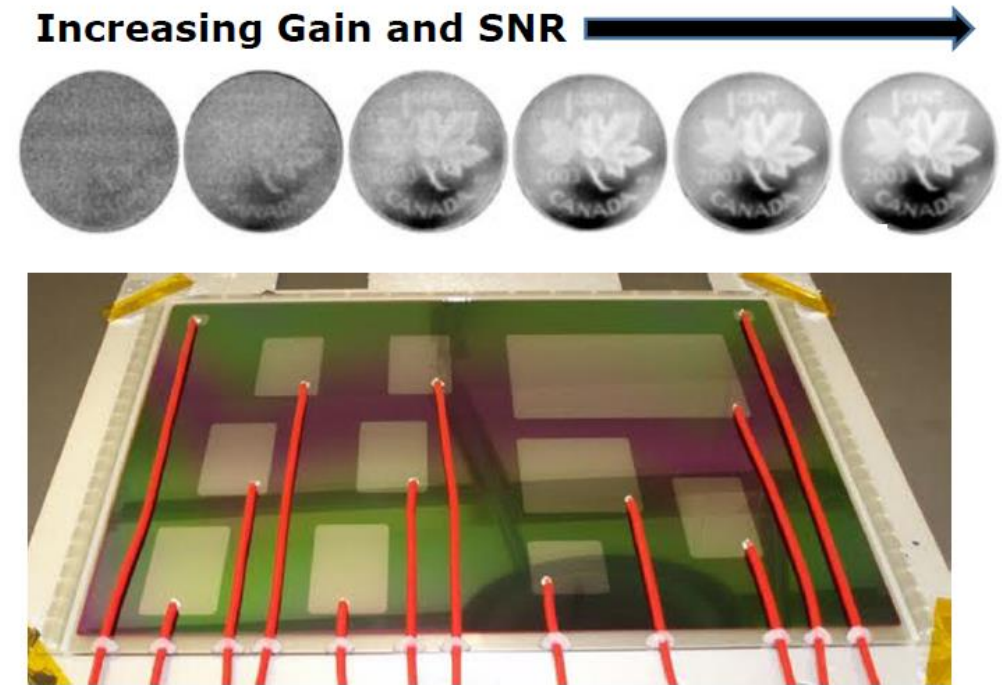


# Amorphous Selenium for Medical Imaging



# Amorphous Selenium

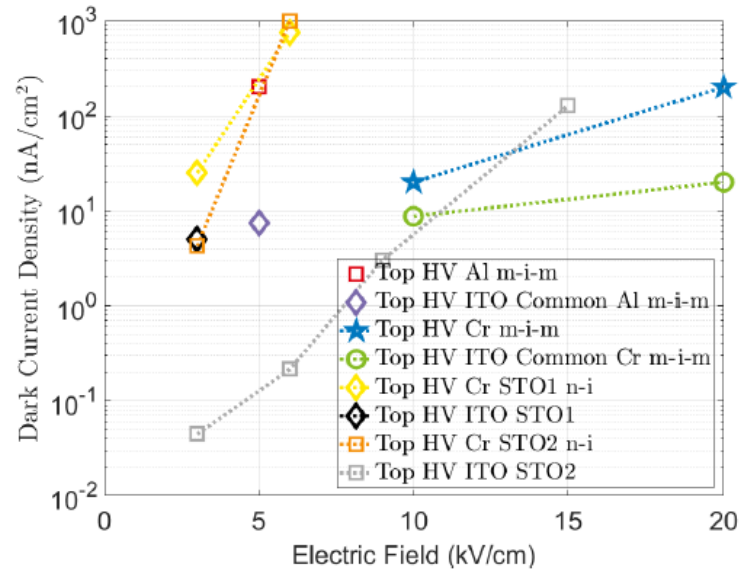
- The most prevalent commercial detector for low-light detection with high dynamic-range and linear-mode operation is the vacuum photo multiplier tube (PMT).
- Bulky and fragile, have poor quantum efficiency in the visible spectrum, are insensitive to infrared light and highly sensitive to magnetic fields.
- Amorphous Selenium (*a*-Se) is emerging as a viable solid-state imager with avalanche gain for low light and low-dose radiation detection applications.



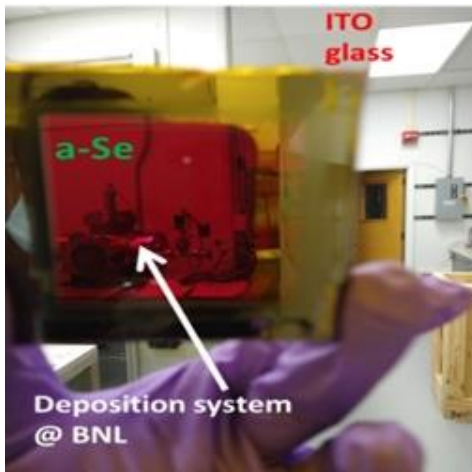
**Figure:** Si TFT array covered with a continuous layer of avalanche *a*-Se detector (24 cm x 30 cm). **1b**

**1b. J. Scheuermann et al. *In: Med. Phys.* (2018).**

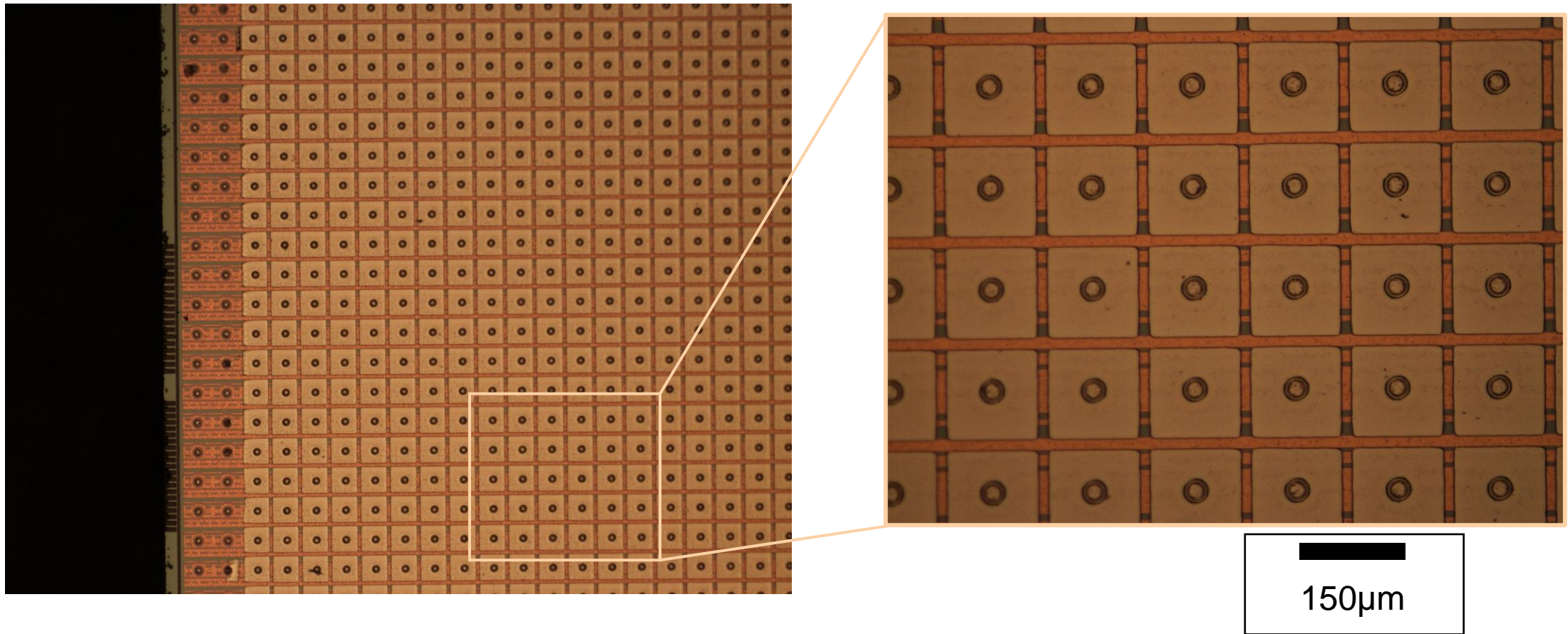
# a-Se: Advantages & Challenges



- Amorphous Selenium is commercially used as a direct conversion X-ray and Gamma ray detector for digital radiography and mammography.
- It can be deposited uniformly over a large area at substantially lower costs as compared to crystalline semiconductors.
- High spatial resolution (below 10 microns) can be achieved for hard X-rays.
- Carrier mobility is poor
- $W$  is high & field dependent



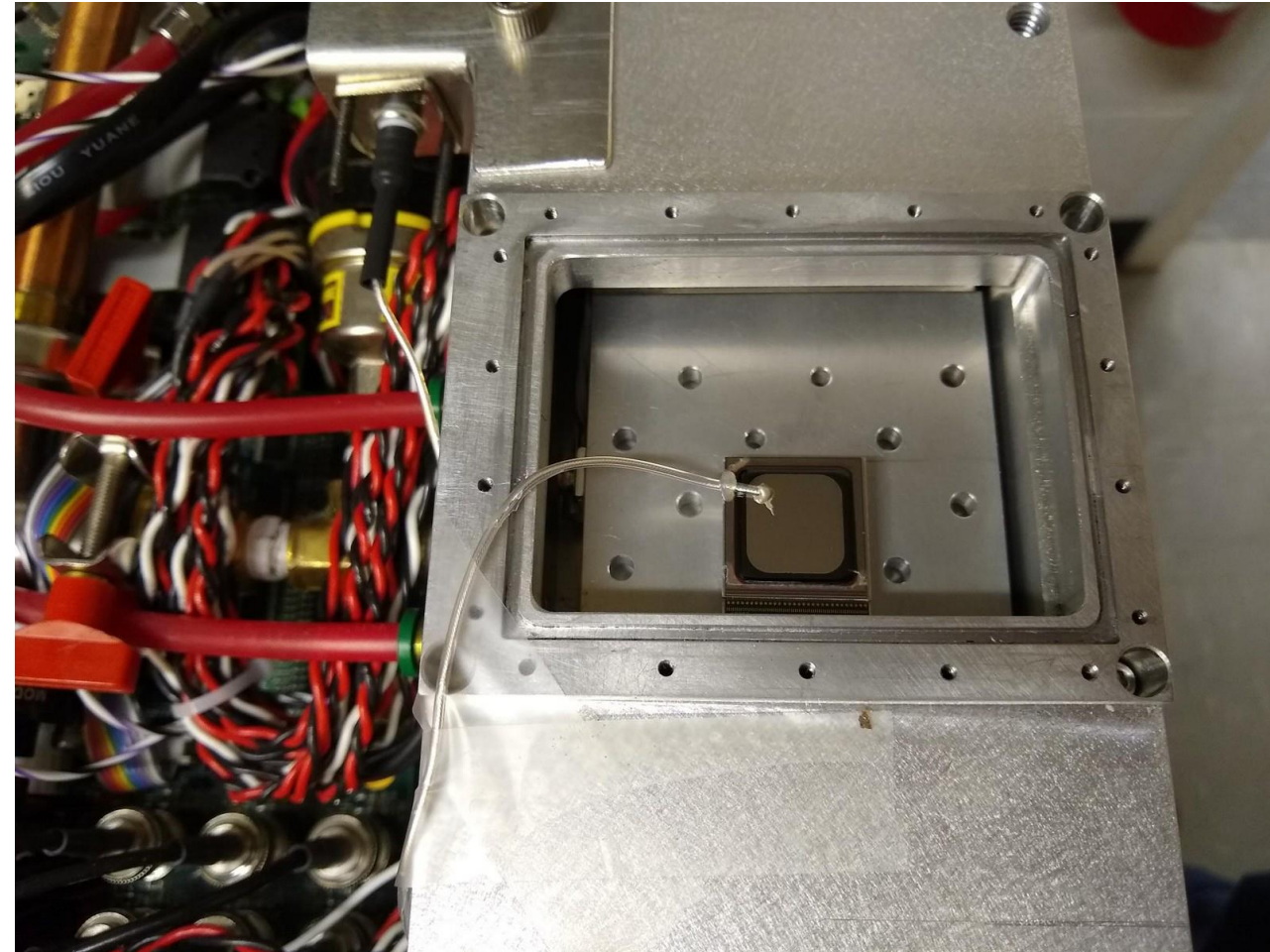
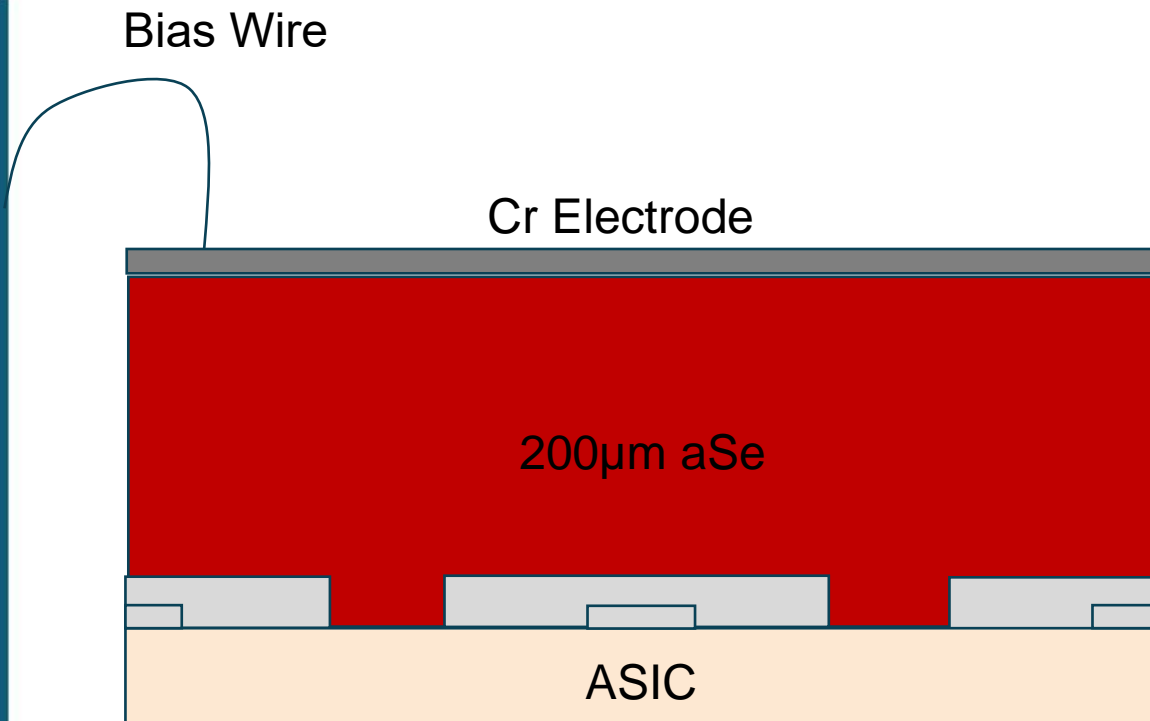
# Fabrication- Pad Extension



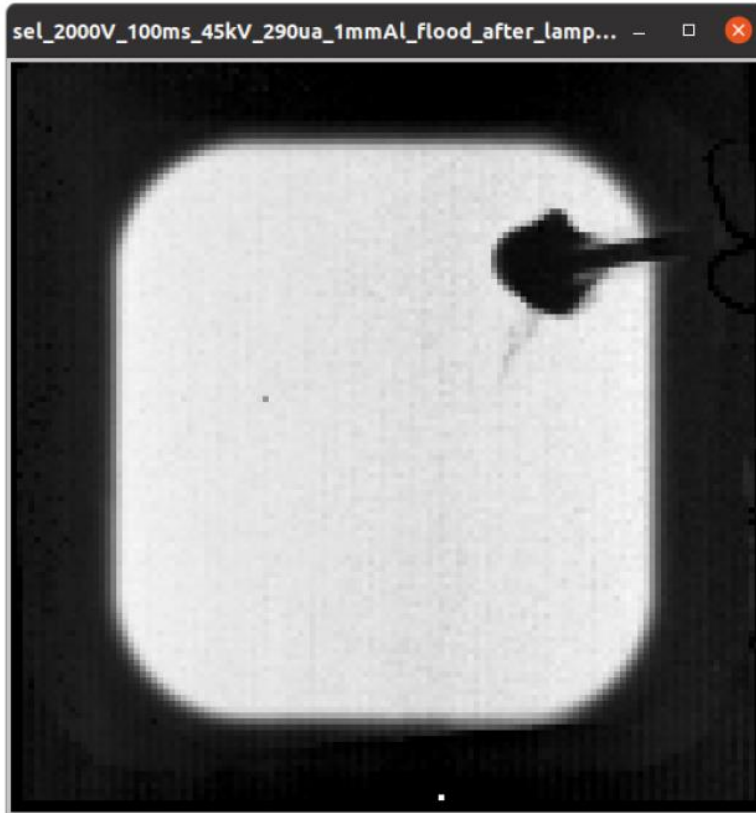
After Liftoff



# Fabrication- rest of the detector



# Initial Tests



1.6% rms  
variation

Ag x-ray tube, 45  
kV bias

Average of 100  
frames, 100ms  
exposure each

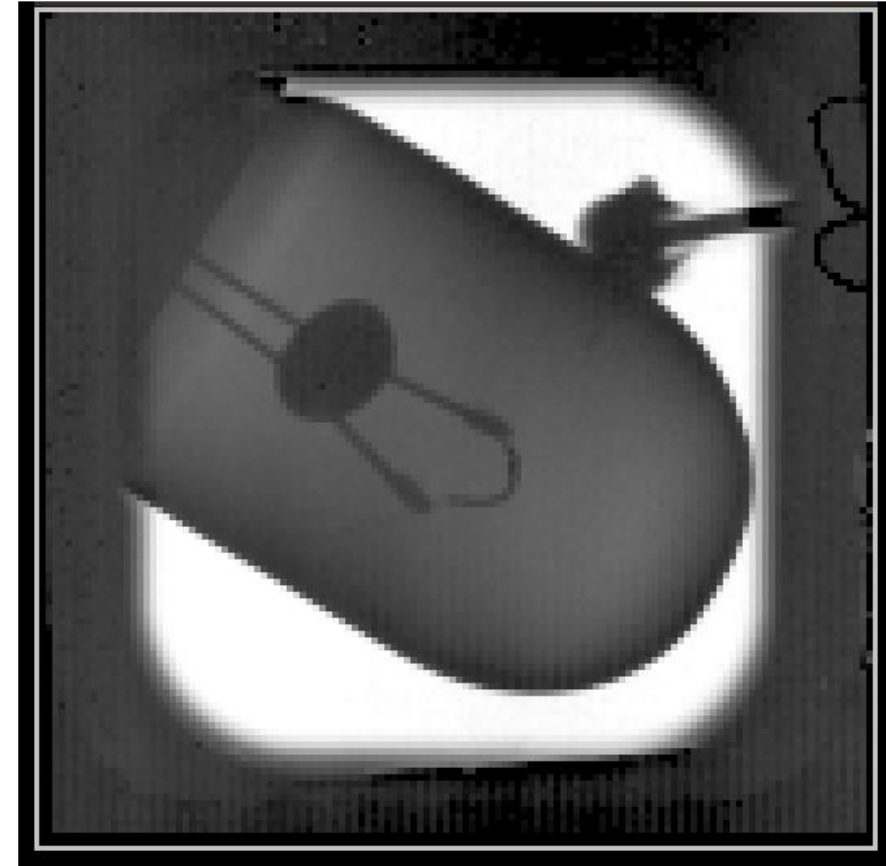
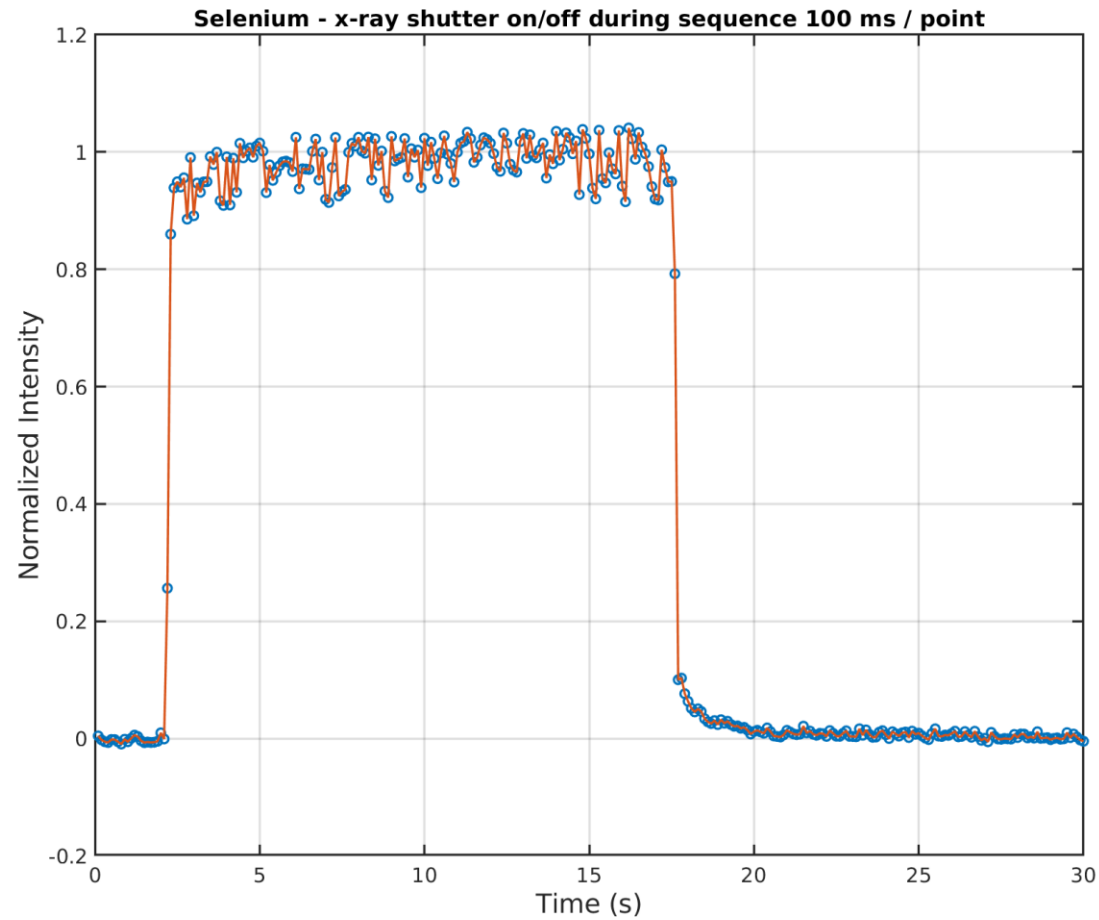


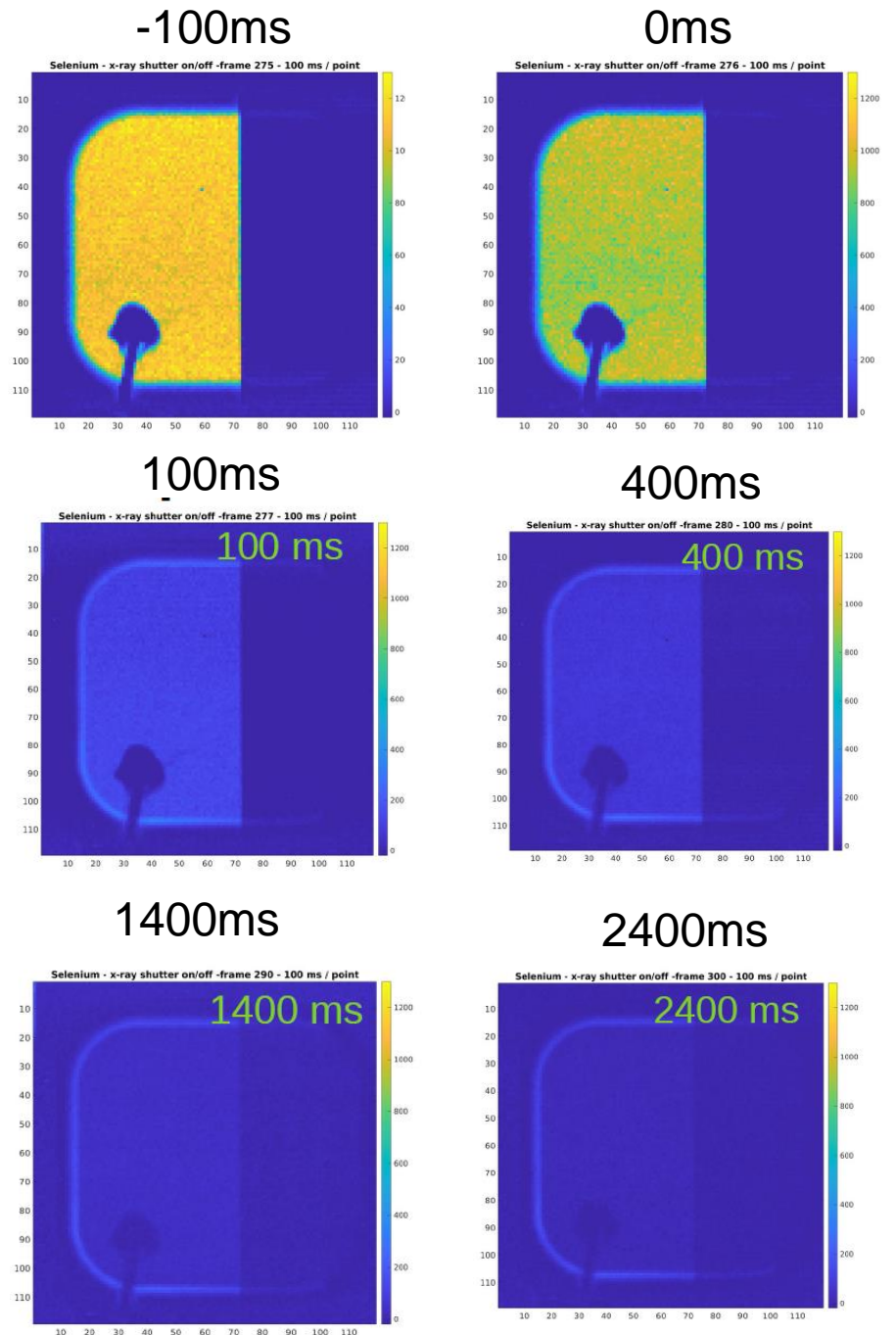
Image of lightbulb,  
raw data (no flat  
field)



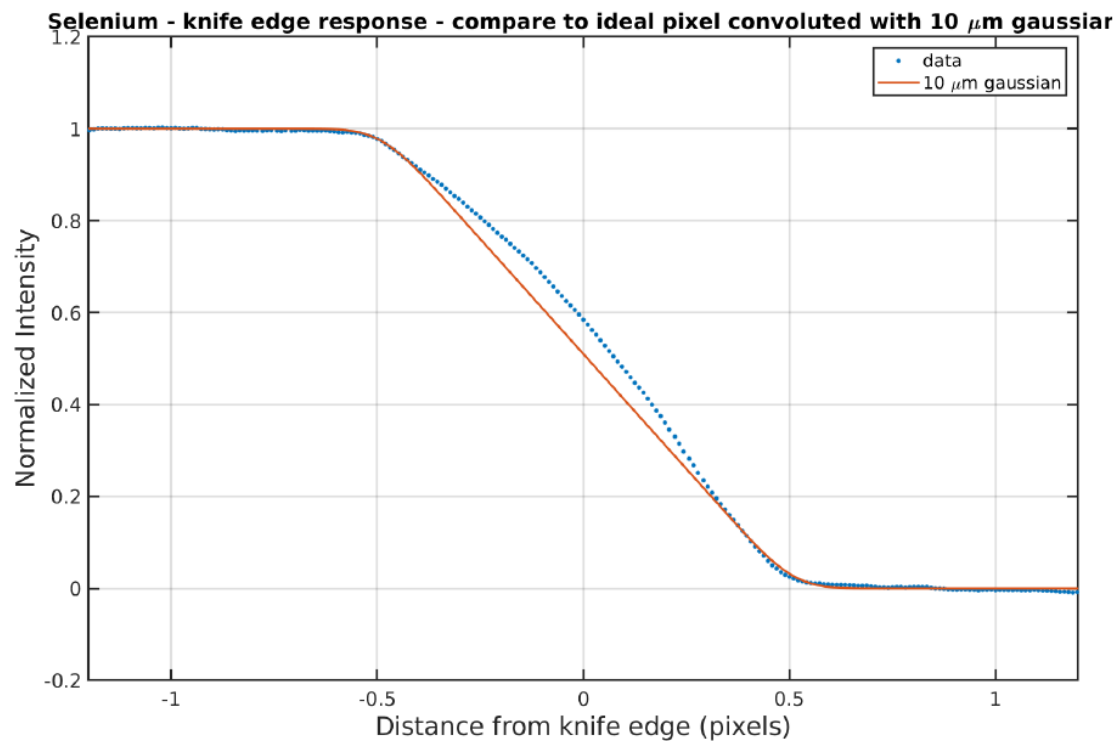
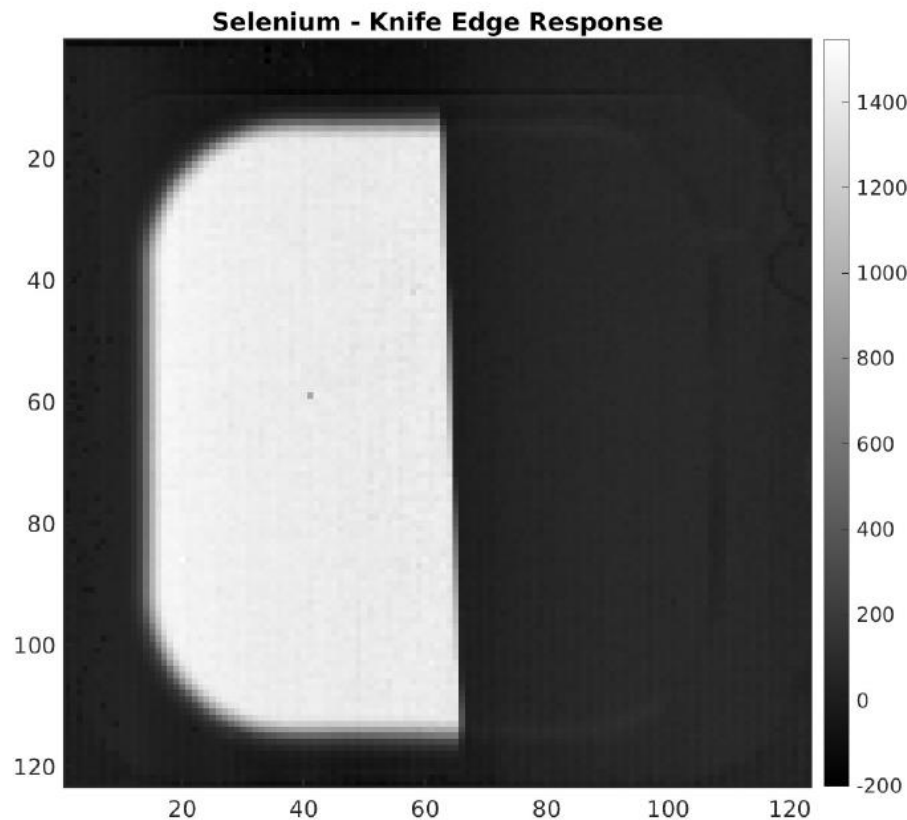
# Dynamic Response



Static knife edge, shutter opened and closed

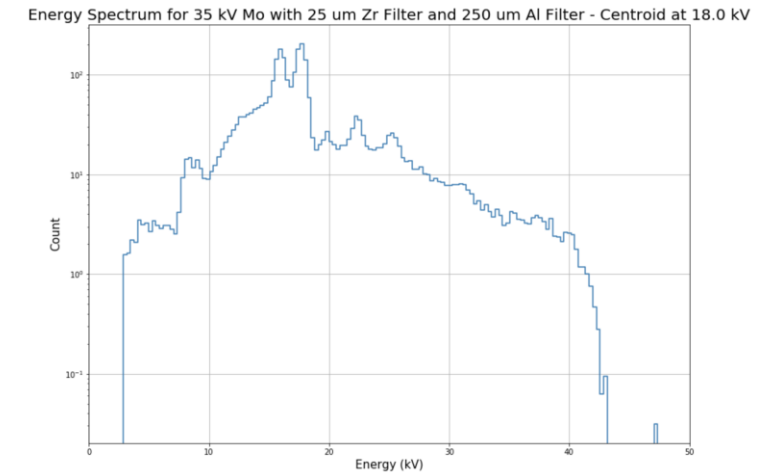
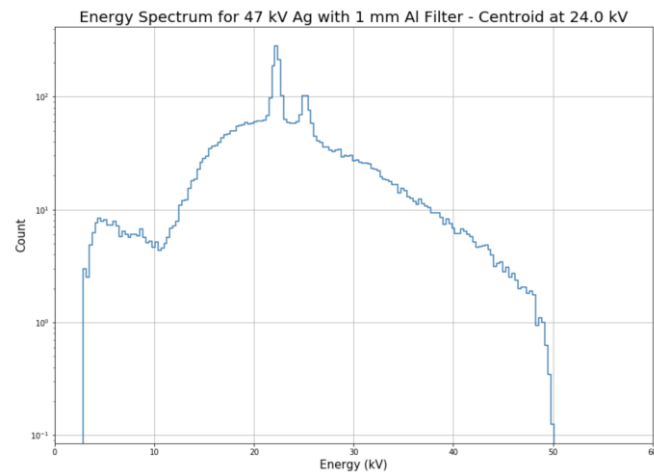
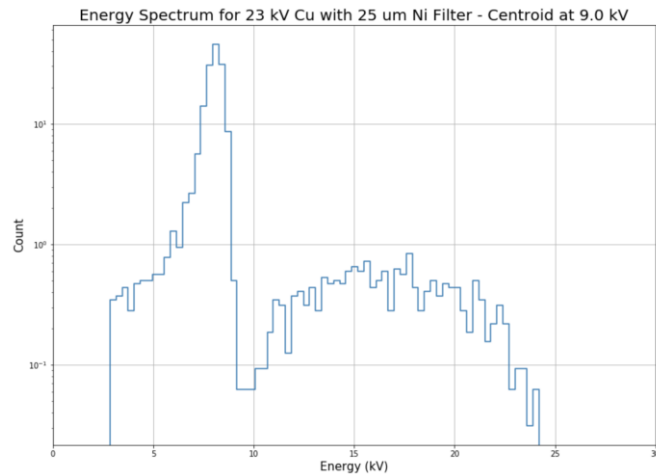
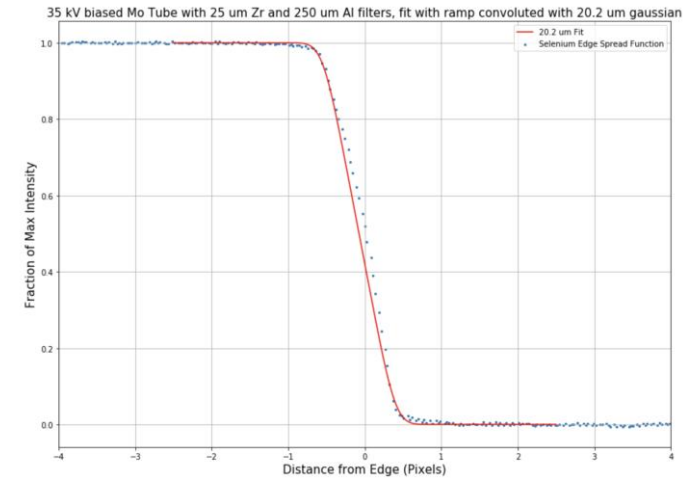
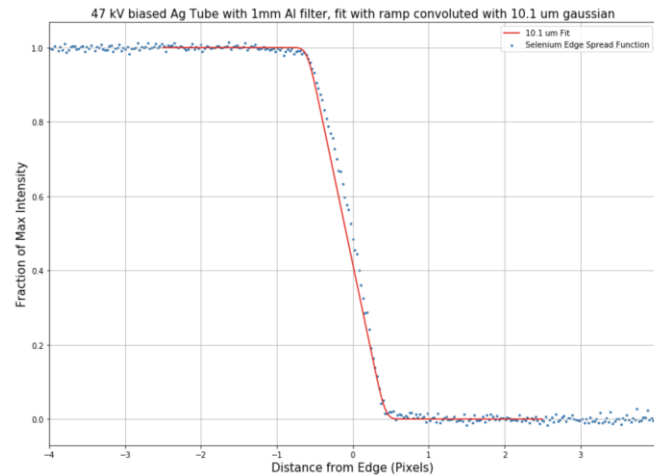
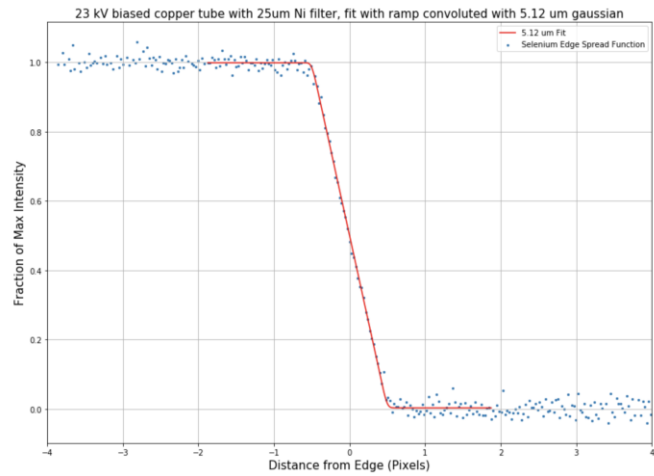


# Knife Edge at an Angle



Deviation from ideal response- more intense than expected

# MTF for different energies



# Simulation of intrinsic spatial resolution

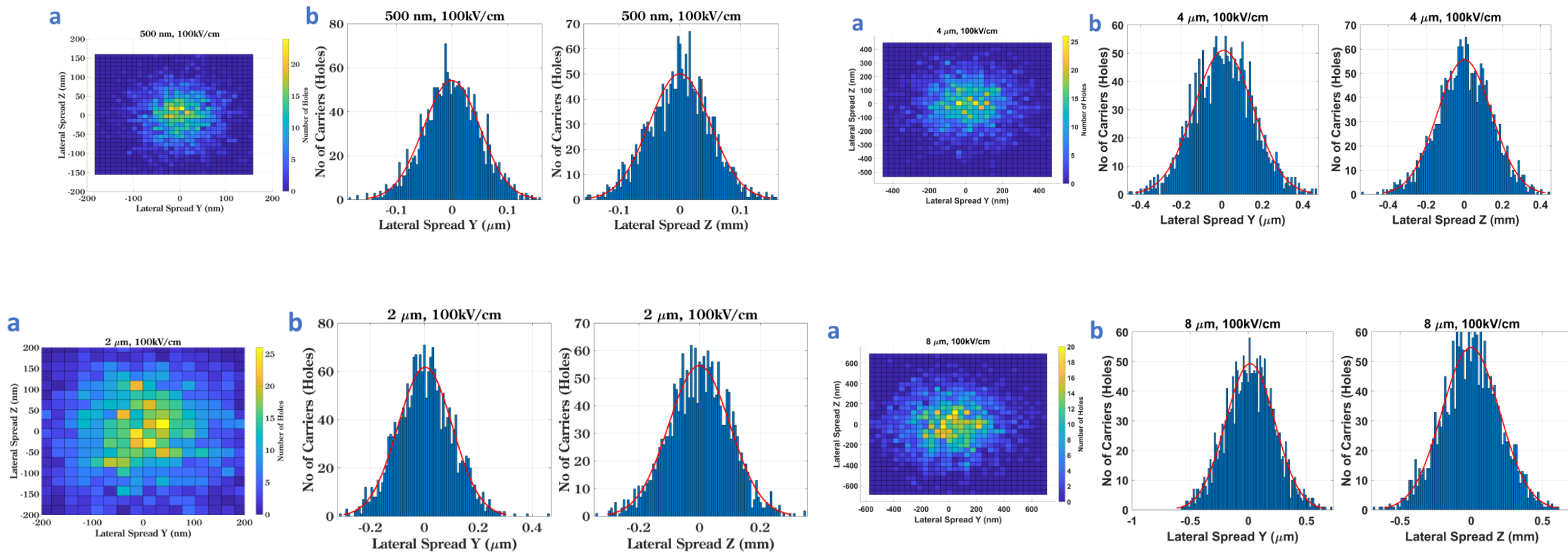
## Semiclassical Monte Carlo Boltzmann Transport equation

$$\frac{\partial f_{\mathbf{k}}(\mathbf{r})}{\partial t} + \frac{\mathbf{F}}{\hbar} \cdot \nabla_{\mathbf{k}} f_{\mathbf{k}}(\mathbf{r}) + \frac{1}{\hbar} \nabla_{\mathbf{k}} E_{\mathbf{k}} \cdot \nabla_{\mathbf{r}} f_{\mathbf{k}}(\mathbf{r}) = \left. \frac{\partial f_{\mathbf{k}}(\mathbf{r})}{\partial t} \right|_{\text{coll}}$$

- Transport of holes interrupted by acoustic, polar and non-polar phonons, disorder, and dipole scattering
- DFT used to calculate energy dependent phonon scattering, dipole and disorder scattering
- Transport model validated with experimental data i.e time-of-flight mobility and impact ionization gain

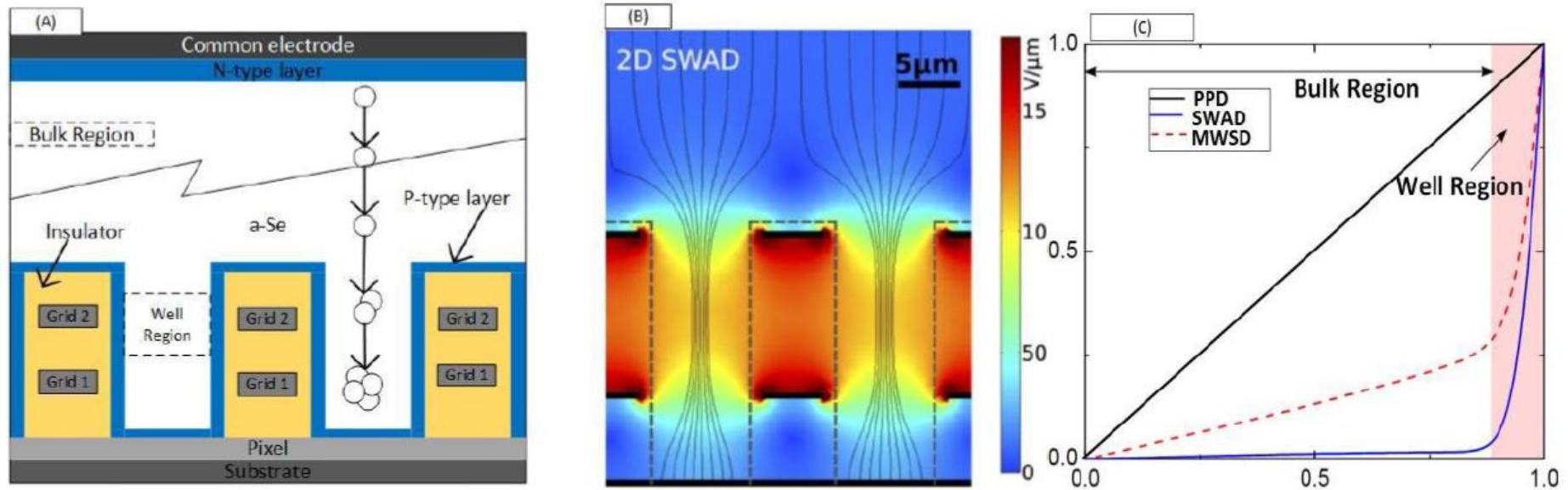


# Simulation of intrinsic spatial resolution

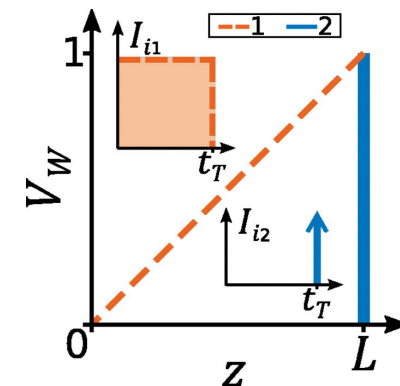


Intrinsic MTF for 200-micron Se was about 2.5 microns

# field-Shaping multi-Well Avalanche Detector



- Can address the problem of low carrier mobility and low charge conversion gain by having a Frisch Grid built in the sensor layer.
- Uniform gain, independent of depth of interaction.
- Reduced charge collection time.
- Leakage in the well has to be minimized. Leakage limits gain.
- Non-trivial fabrication process.

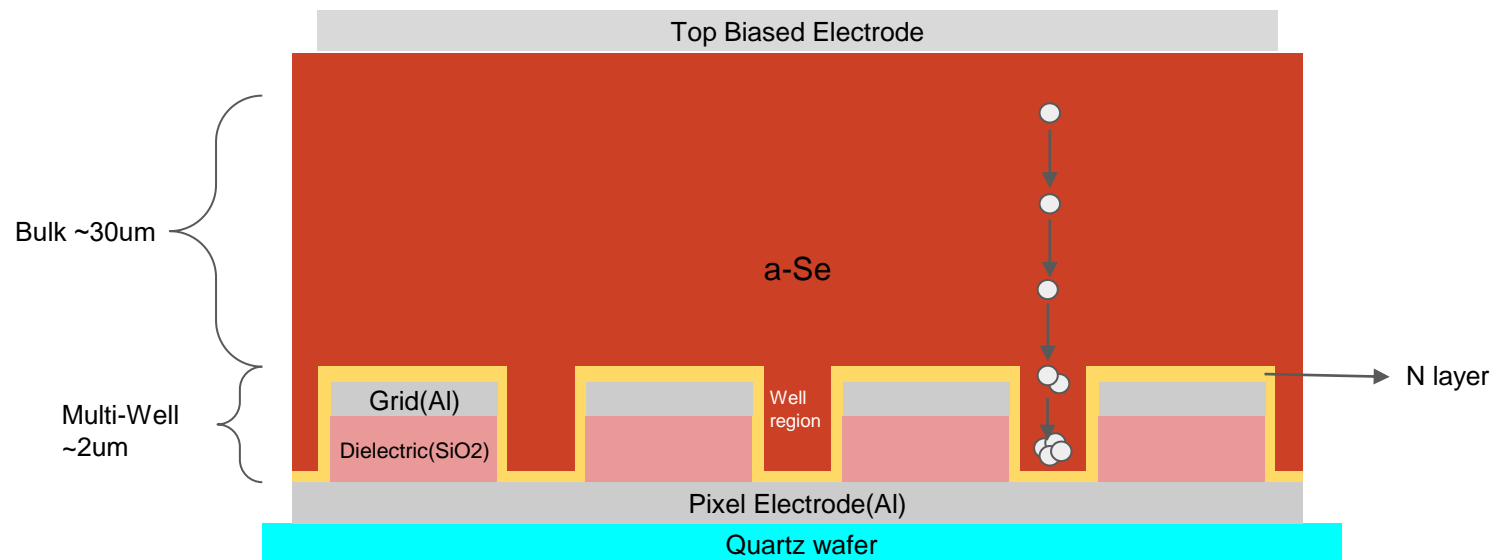


Appl. Phys. Lett. **101**, 213503 (2012)

# Future research in a-Se - SWAD with high-k hole blocking layer

- Fabrication of field-Shaping multi-Well Avalanche Detector (SWAD): Improving time resolution and uniformity in gain

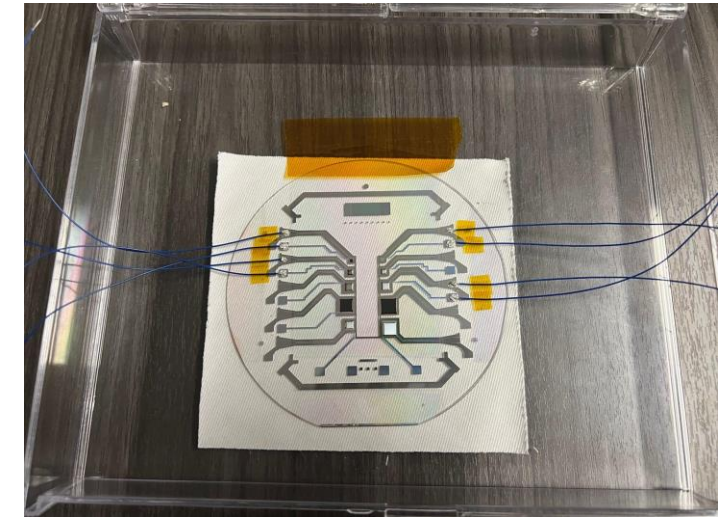
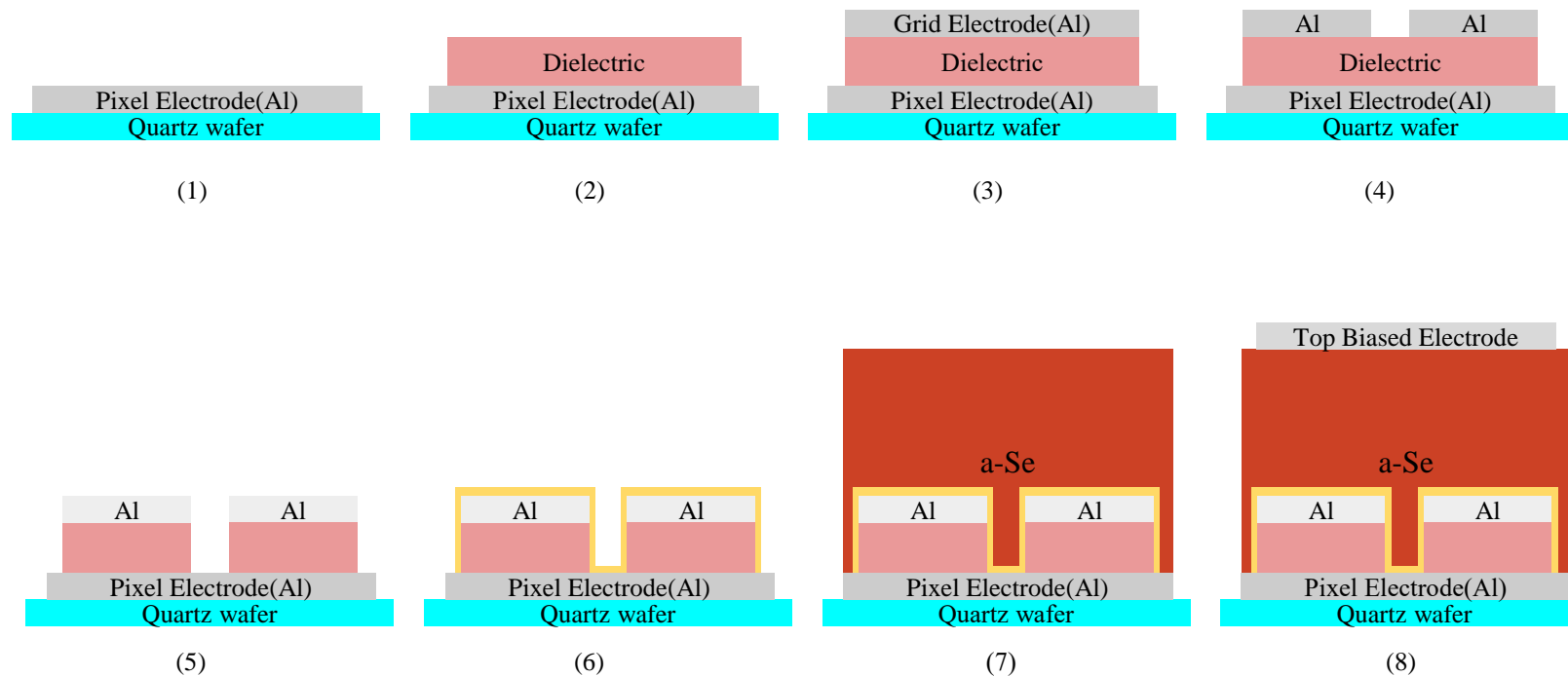
The schematic diagram of SWAD design



# SWAD with hole blocking layer

## Fabrication

The illustration of fabrication process steps to fabricate SWAD multiwell structure, utilizing photolithography, Reactive Ion Etching(RIE), and material deposition techniques

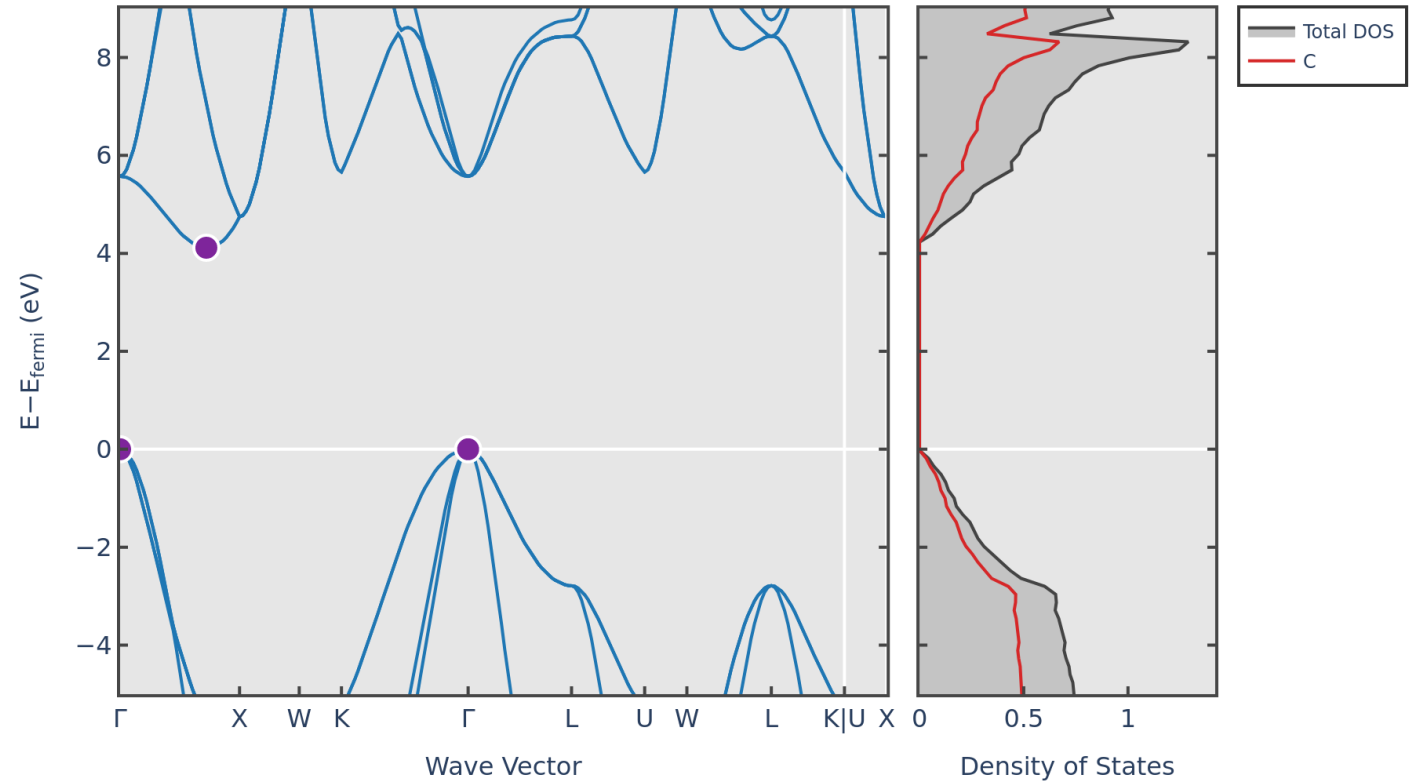
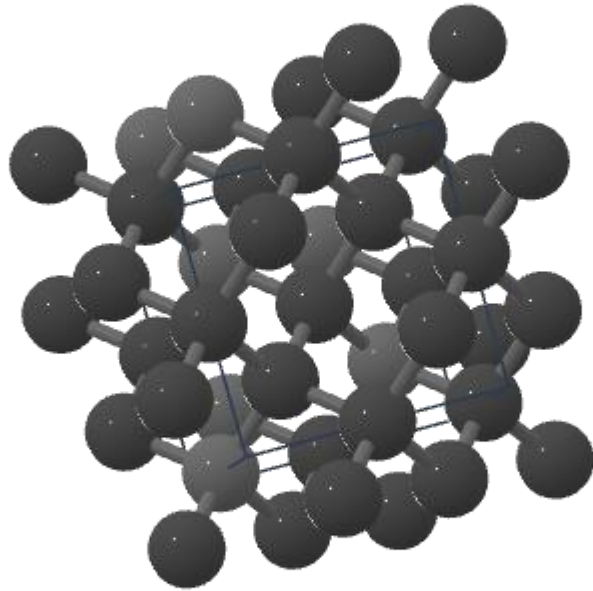




# Large band gap semiconductor

# Diamond

Cubic  $F\bar{d}3m$  space group  
Band Gap: 4.11 eV



# Why diamond

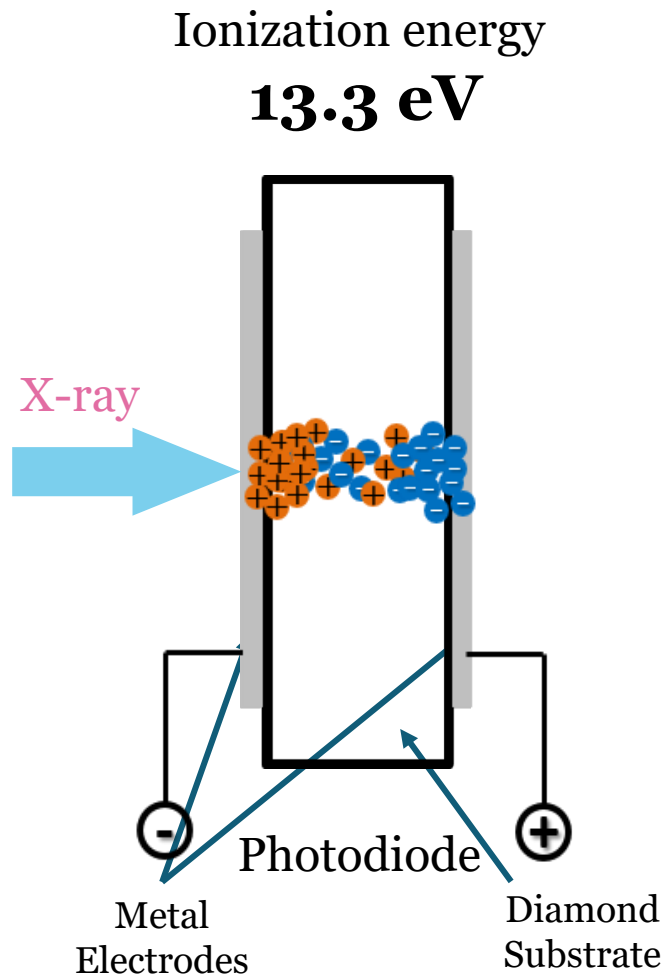
## Benefits of diamond

- Large band-gap: low leakage, high resistivity
- Radiation hard
  - Damage appears as reduction of signal
  - In silicon, there is potential for both signal reduction and increased leakage
- Higher saturation velocity
  - Diamond:  $2.7 \times 10^7$  cm/s (e)
  - Silicon:  $1 \times 10^7$  cm/s (e)
  - Transit time for a 50  $\mu\text{m}$ -thick device: 185 ps (diamond), 500 ps (silicon)

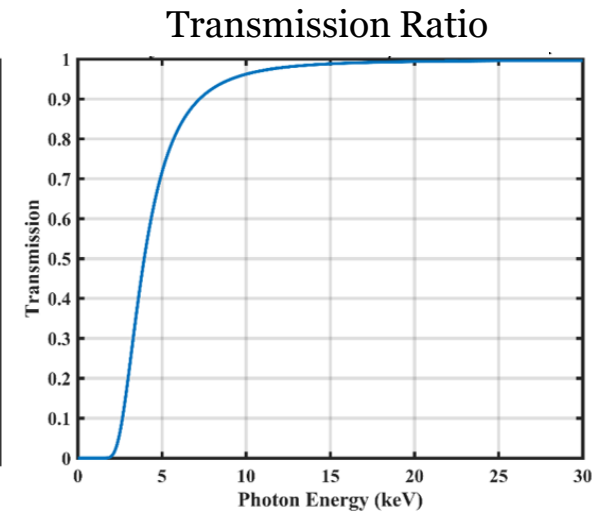
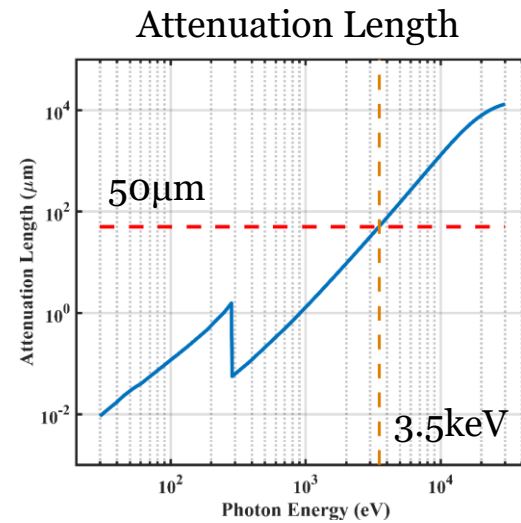
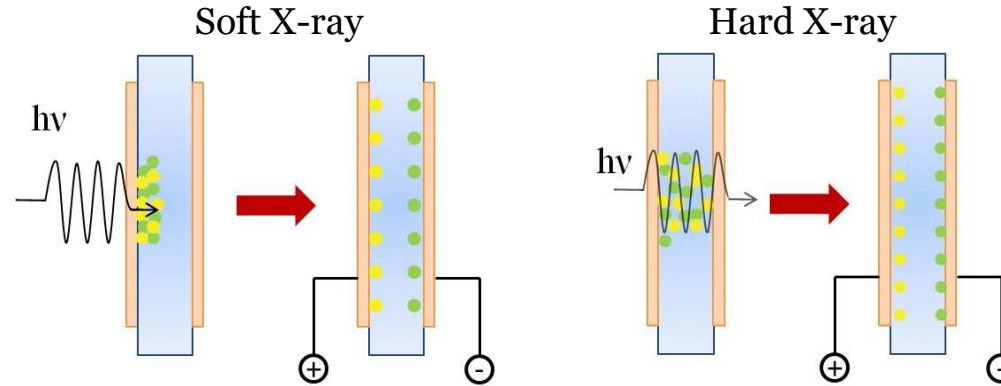
## Challenges of diamond

- Cost/availability
- Size
  - Single crystal material is limited to 4.5 mm  $\times$  4.5 mm
  - Polycrystalline diamond available 1 cm  $\times$  1 cm
  - Requires a tiling solution which complicates assembly
- Higher  $W$  than silicon – about 1/3 lower signal

# Diamond - Detector

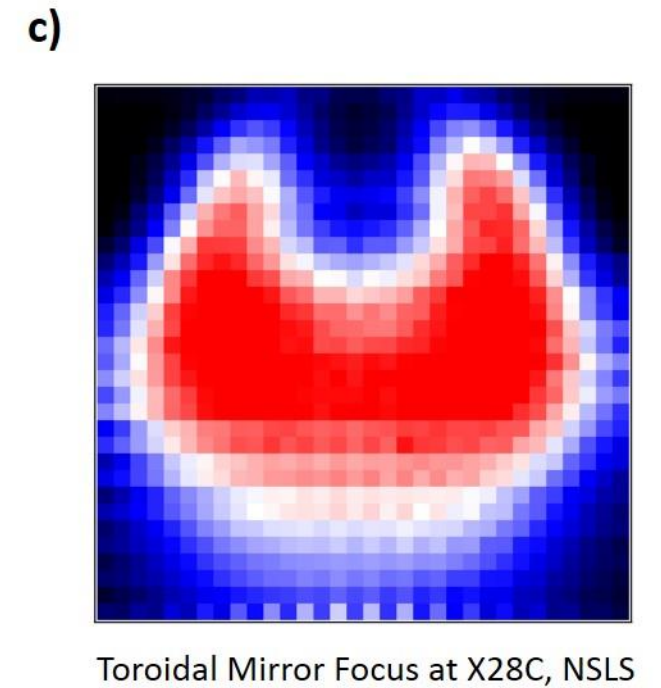
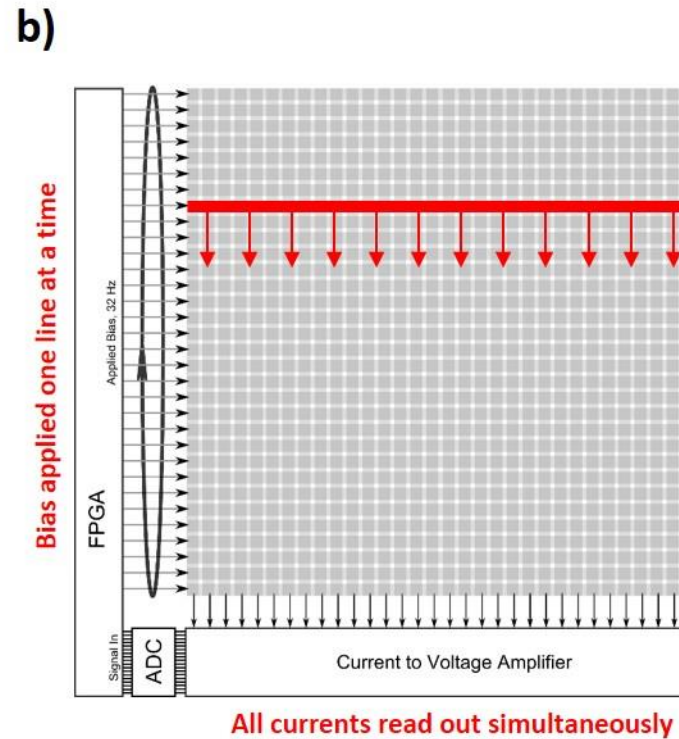
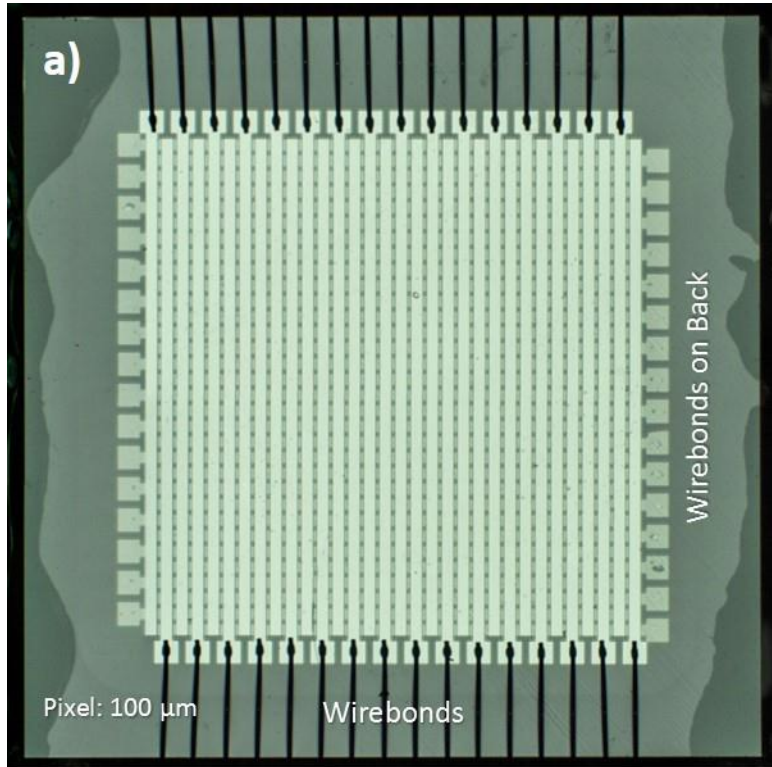


## Two Scenarios





# Diamond Imaging Detectors

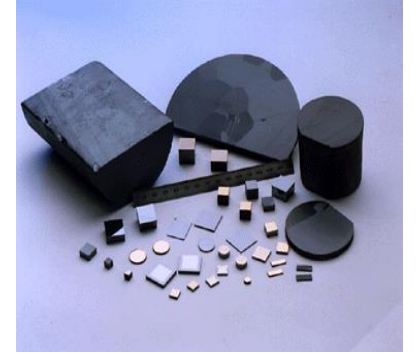


Diamond imaging detector. (a) A diamond sensor with 20 nm-thick platinum strips forming 100  $\mu\text{m}$  virtual pixels. (b) Readout scheme where the bias is applied to individual strips to read out the one line at a time. A complete image is readout out at  $\sim 32$  Hz. (c) image of white x-ray beam after a toroidal focusing mirror.

# CdZnTe for Gamma Rays

# Advantages of Cadmium Zinc Telluride (CZT) Material

- High average Z
- High density ( $\text{g/cm}^3$ )
- Wide bandgap  $\rightarrow$  high-resistivity
- $\rightarrow$  can operate at room temperature;  
(best results usually obtained at  $\sim 5\text{-}10\text{ C}$ )
- Provide high energy resolution (sub keV in X-ray region, and  $\sim 1\%$  at 662 keV in gamma-ray range) and sub-millimeter spatial resolution for imaging devices
- Device fabrication utilizes technologies developed in semiconductor industry, which make CZT detectors less expensive and more robust
- CZT is very attractive material for gamma-ray detectors; Its feasibility has been demonstrated in many applications
- However, crystal defects, which exist even in the best quality CZT crystals, limit availability and increase costs of CZT detectors; the main challenge facing today detector developers is how to overcome these problems



# Why do we need to detect gamma rays?

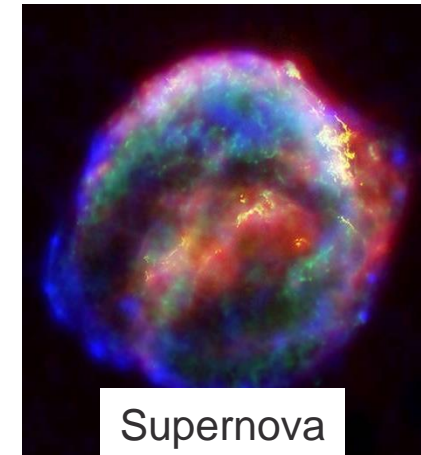
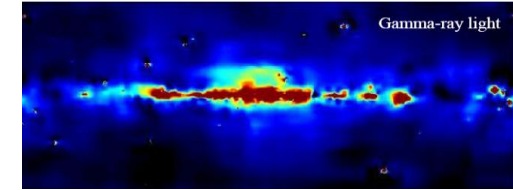
Gamma rays emitted by radioactive nuclei and unavoidably present in our environment

Provide unique signatures of isotopes and nuclear reactions taking place at nature

Gamma rays are emitted at discrete energies seen as peaks in their energy spectra

Because the natural widths of the peaks are very narrow they are called gamma-ray lines

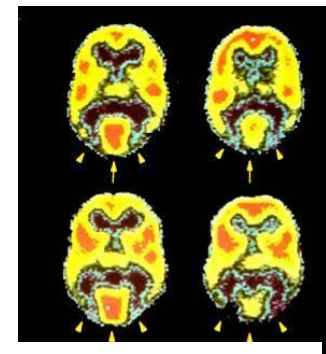
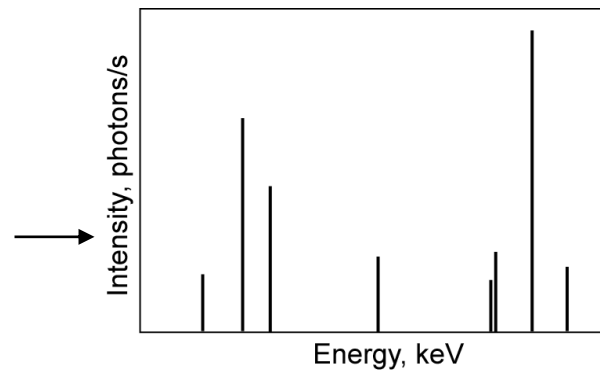
Energy spectra can be used like ID cards to identify particular nuclei



Supernova

Example of ideal gamma-ray energy spectrum

Lines indicate intensities and energies of gamma-rays



PET scan image



Nuclear reactor



# CZT growth techniques

**There are two major producers of commercial detector-grade CZT material:  
eV Products (USA) and Redlen (Canada)**

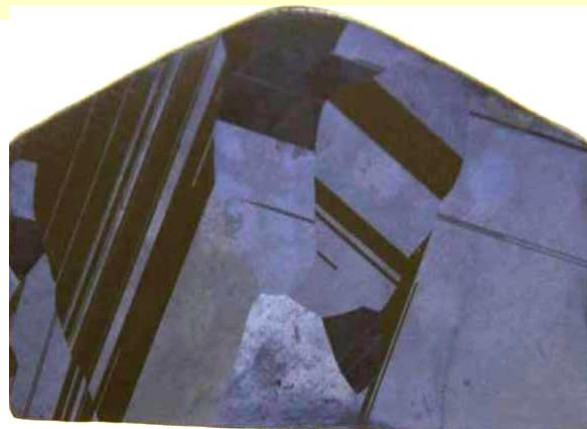
**They can grow relatively big single crystals, up to 10 of cc**

**For a while eV-Products and Redlen used different growth methods:**

## High-Pressure Bridgeman (eV)

~1100 C,  
Use high-pressure nitrogen gas to  
suppress formation of point defects  
Best quality CZT small-size crystals  
Very non-uniform on a large scale  
(extended defects)

Wafer cut  
from a 7-inch  
ingot



## Traveling Heater Method (Redlen)

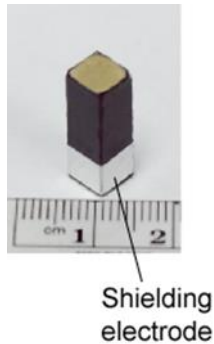
~600 C  
Very uniform  
Quality to satisfy requirements for  
making detectors  
Lower cost



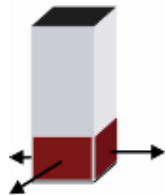
3-inch  
wafer

# Novel CZT instruments based on arrays of position-sensitive virtual Frisch-grid detectors

Solid shielding electrode (conventional)



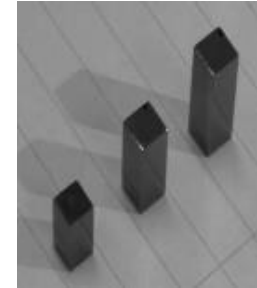
Position-sensitive virtual Frisch-grid detector



- To overcome the high cost and low availability of big CZT crystals we propose arrays of small cross-section, up to  $7 \times 7 \text{ mm}^2$ , but long, up to 5 cm, detectors
- Such crystals have much higher production-yield and lower; typically cut from the crystals rejected by other costumers
- 4 position-sensing pads (non-contacting electrodes) attached to the side surfaces near the anode; the pads signals provide X-Y coordinates
- The cathode signals are used to measure drift times to evaluate Z coordinates, like in TPC;
- C/A ratio gives independent estimates for Z
- Virtually grounded pads produce the Frisch-grid effect (as if a real grid were placed inside the detector)

This design (drift cell) was originally proposed for noble gas detectors by V. Dmitrenko et al. This idea was later applied to CZT detectors by G. Montemont (LETI) and D. McGregor (Kansas State University)

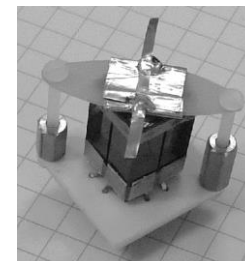
CZT bars



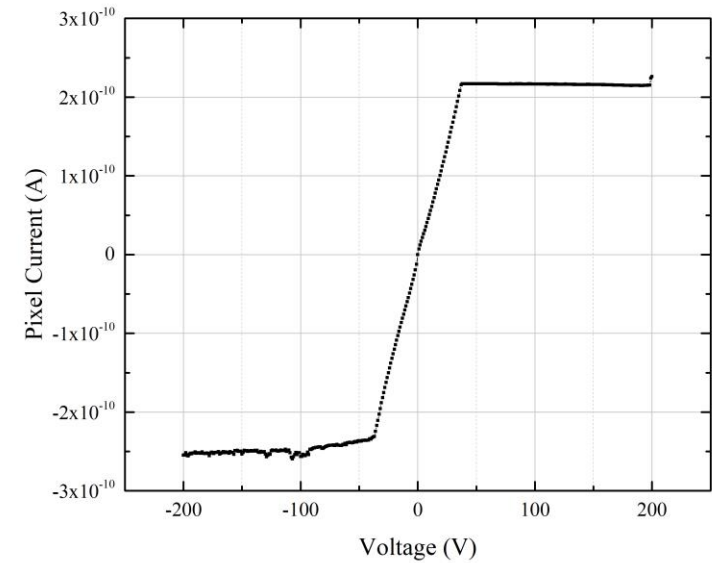
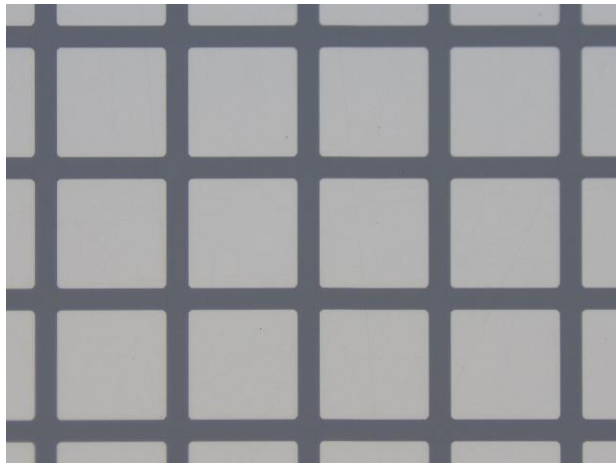
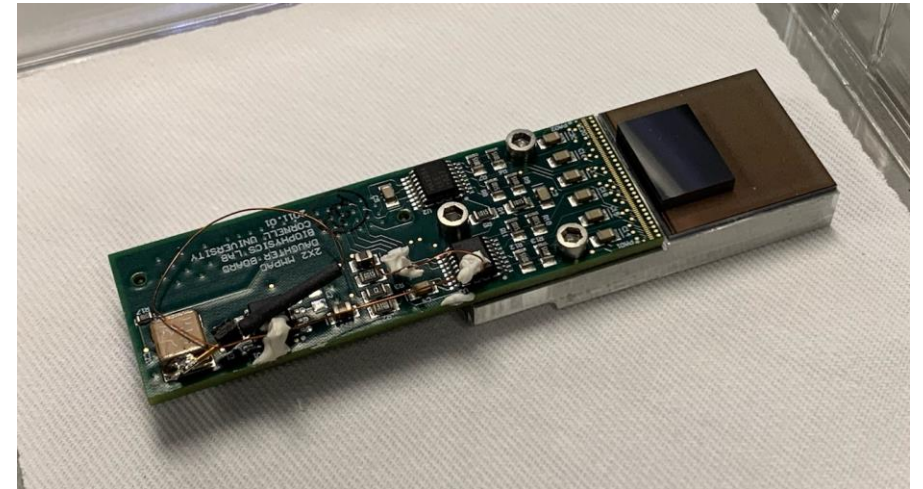
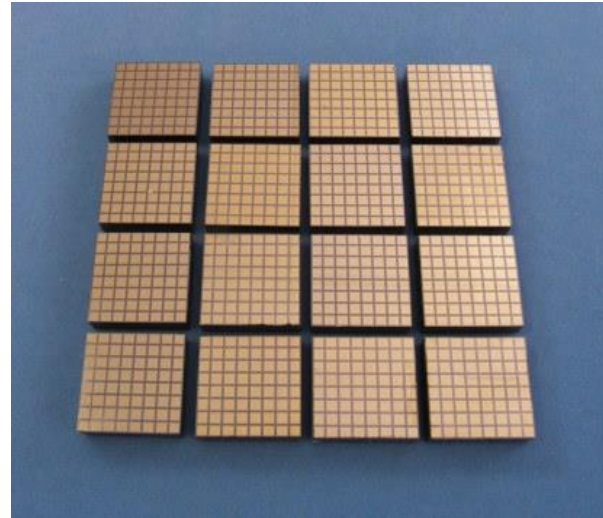
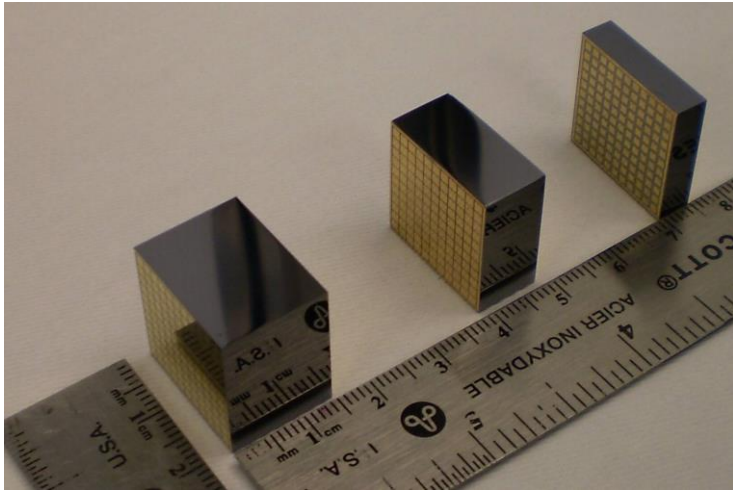
Cathode



Anode



# CZT (Redlen) sensors bonded to readout chip



# Acknowledgements

NSLS II Detector Group: D.P. Siddons, A.J. Kuczewski, S. LaMarra, and J. Li.

Instrumentation Division: G. Gaicomini, W. Chen, G. Carini, A. Bolotnikov, E. Muller, G. Deptuch, R. Angona, D. Elliott, D. Pinelli, J. Pinz, A. Verderosa, F. Capocasa, G. Pinaroli, B. Weldon, I. Harding

FZ-Julich : T. Krings

Cornell: S. Gruner, M. Tate, J. Thom-Levy

Stony Brook U/ Weill Cornell: A. Goldan, Z. Han, A. Mukherjee

High-Z Collaboration: Cornell, APS, SLAC, Northwestern