

Multi-Modal & Multi-Dimensional and *Operando* Synchrotron Investigation of Energy-Storage Materials

Yu-chen Karen Chen-Wiegart, Ph.D.

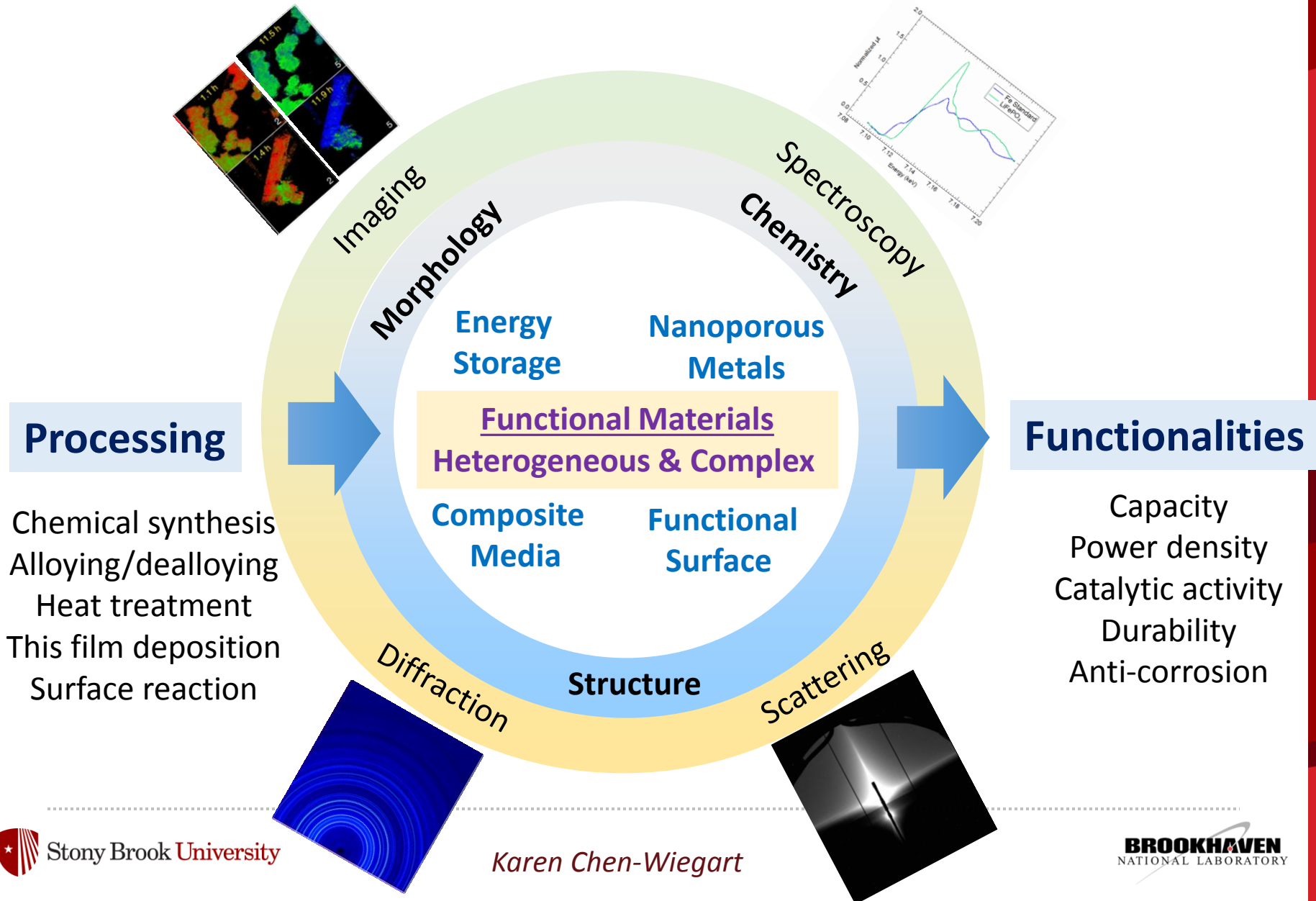
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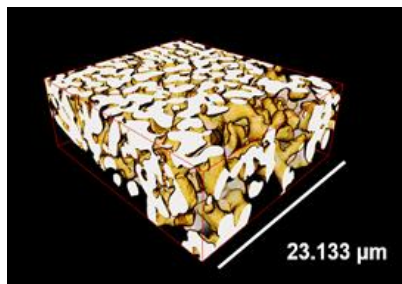
Joint Appointment (Lead on Task Force of Multi-modal Approach)

National Synchrotron Light Source – II, Brookhaven National Laboratory

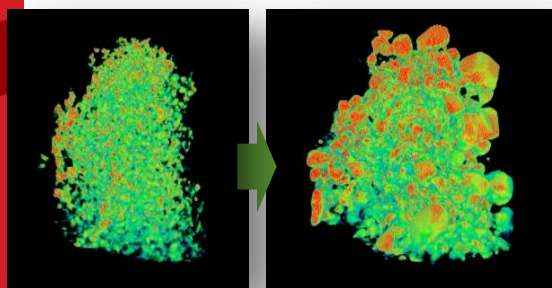
Multi-Modal Synchrotron Approach



Multi-Dimensional X-ray Imaging



In situ 3D morphology evolution



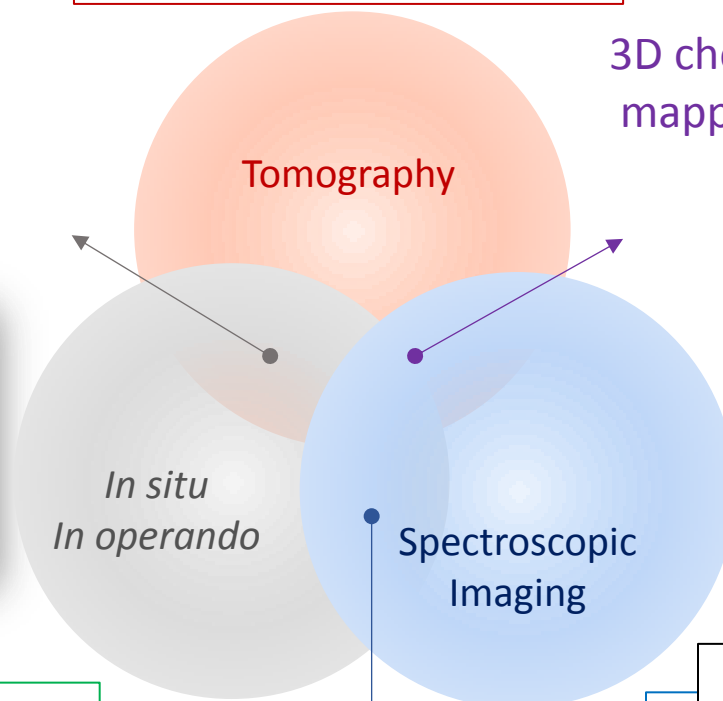
Solid oxide fuel cells

Need 2

Time resolved
with real Environment

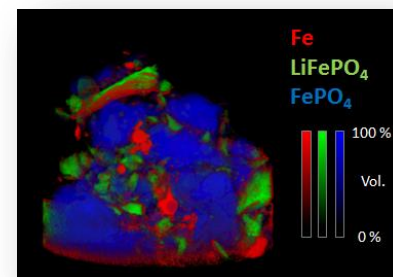
Need 1

3 Dimensional (Spatial)
Characterization

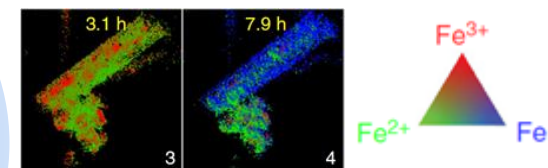


In situ
chemical evolution

3D chemical
mapping



Li-ion Battery



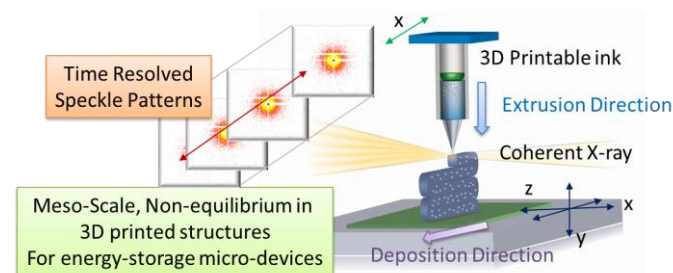
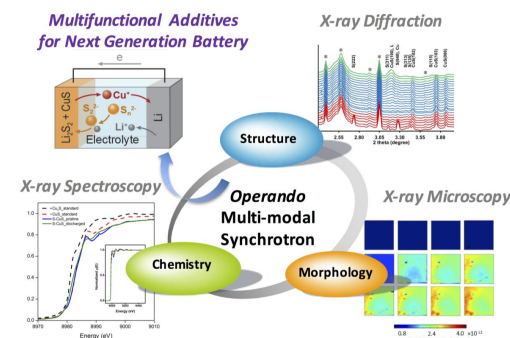
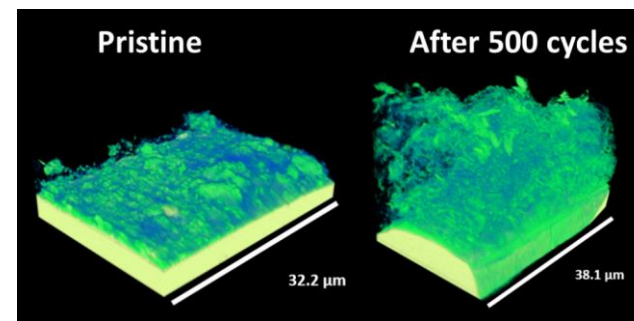
Li-ion Battery

Need 3

Elemental & Chemical
sensitivity

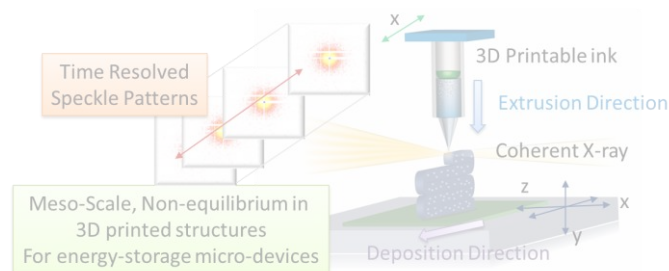
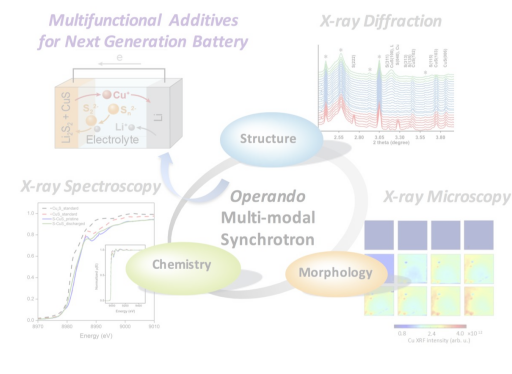
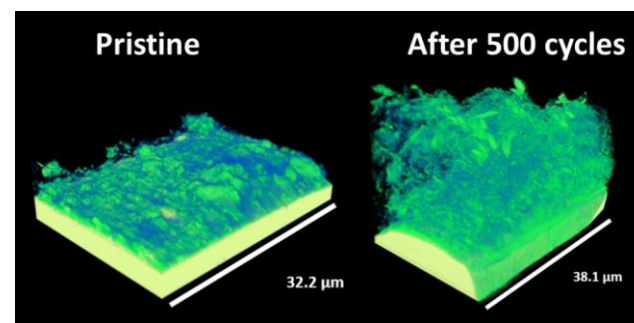
Outline

- Nanoporous-Si Anode
3D morphological Evolution
 - X-ray Nano-tomography
- CuS Additives in Li-S batteries
 - Multi-modal: XRF, XAS and XRD
- *In situ* study of 3D Printing for future energy storage
 - Coherent X-ray Scattering: XPCS

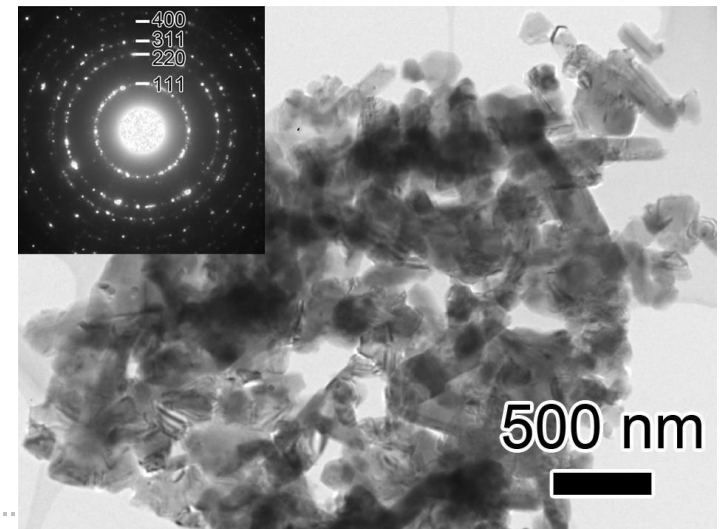
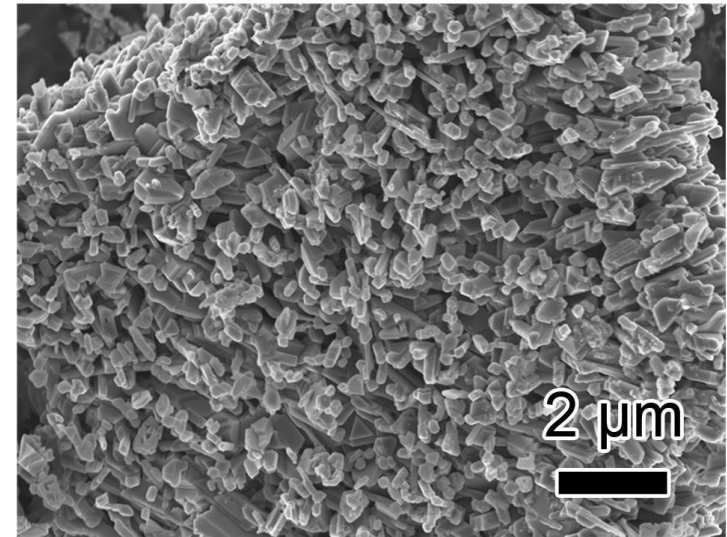
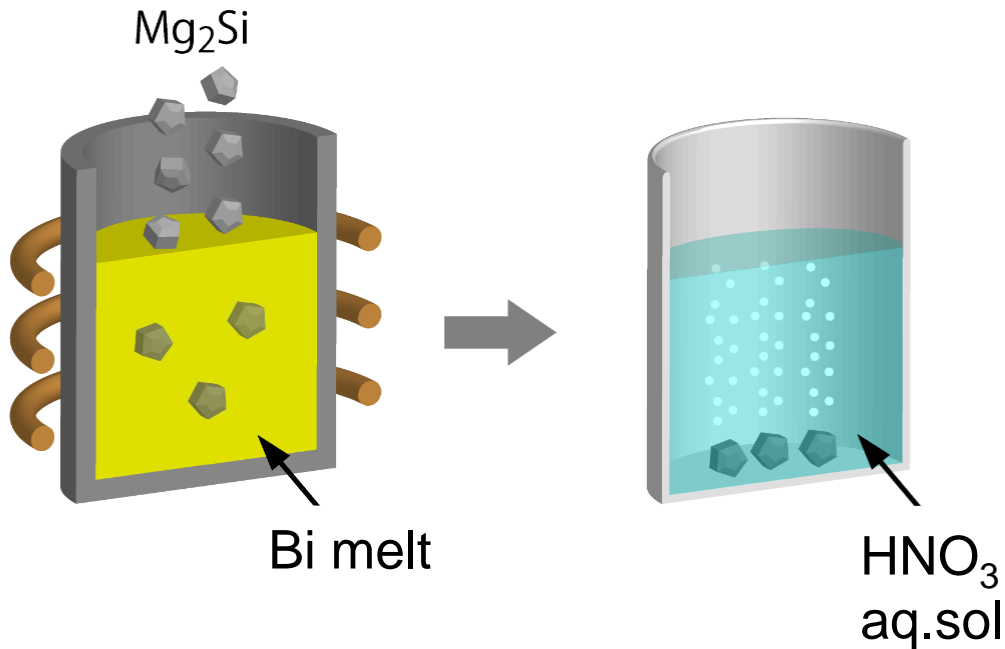


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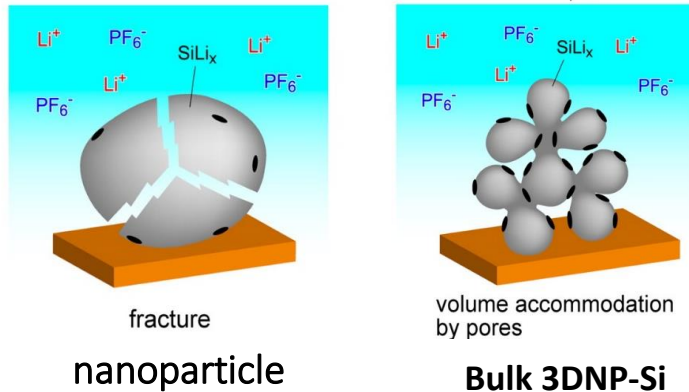


Nanoporous Si for Li-ion Battery Anode



- Immersing Mg₂Si in Bi melt
- Collecting the floating powders ($\rho_{\text{Bi}} \gg \rho_{\text{Si}}$)
- Etching Bi from the collected powders
- Filtering & drying

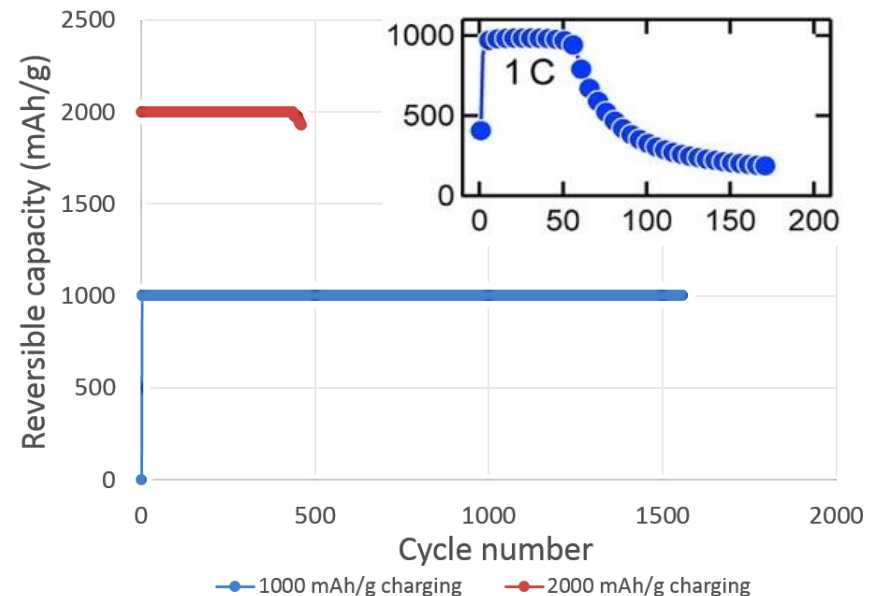
Challenges in nanoporous Si anode



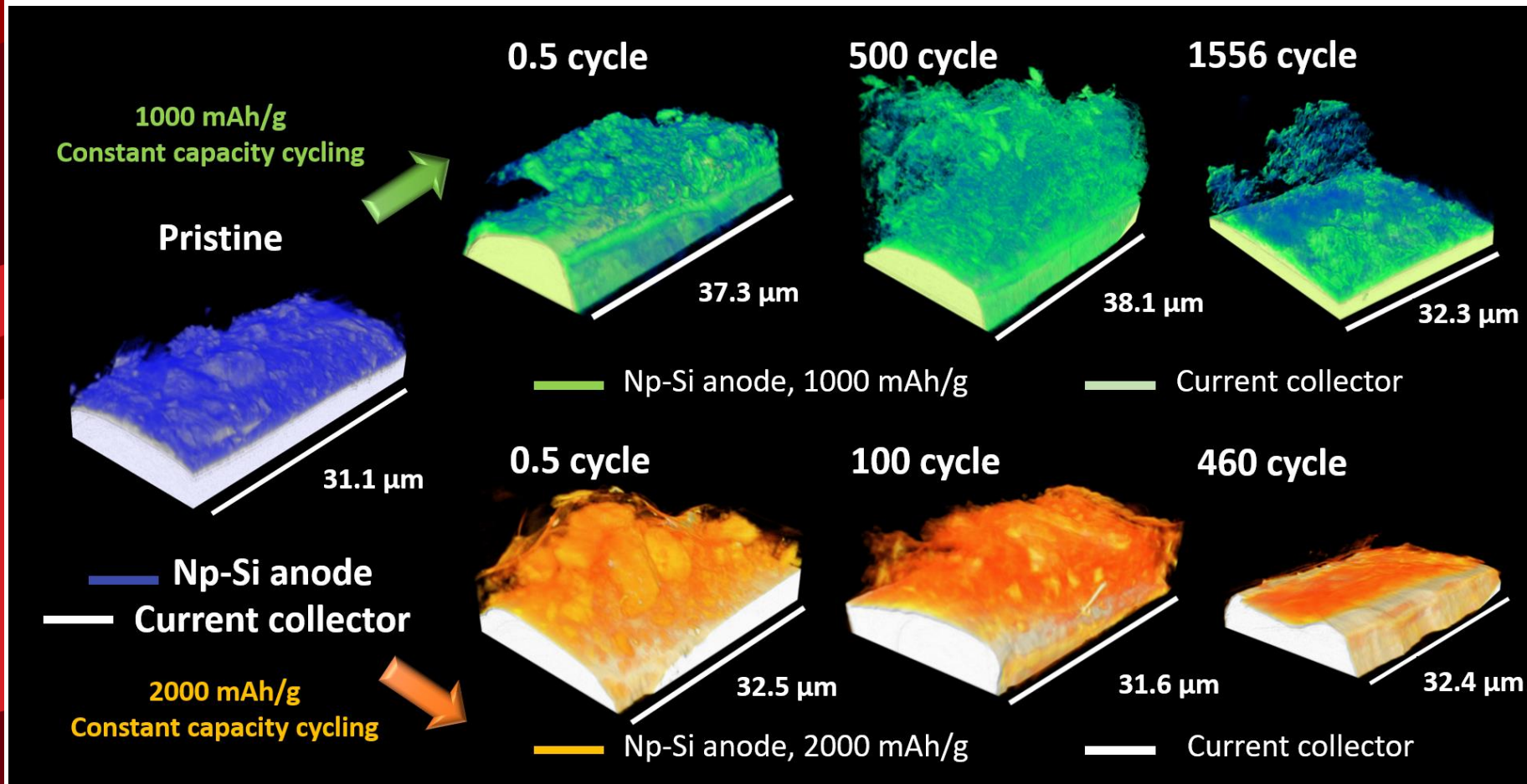
- Porosity: 60.4%
- Ideal volume accommodation limit: 253%
- Correspond lithiation capacity:
 - 2000 mAh/g

- Capacity-dependent behavior
 - 1000 mA/h-g: over 1500 cycles
 - 2000 mA/h-g: < 500 cycles

3D morphological change
v.s. different charging capacity?

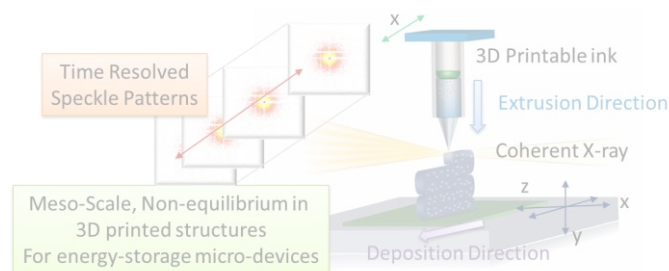
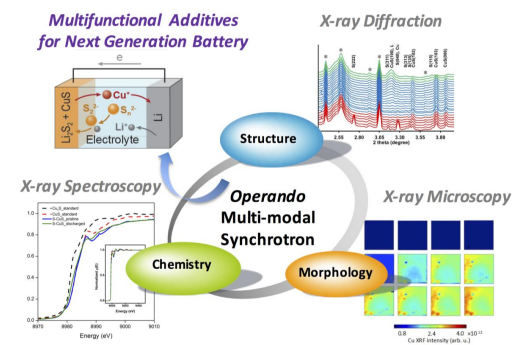
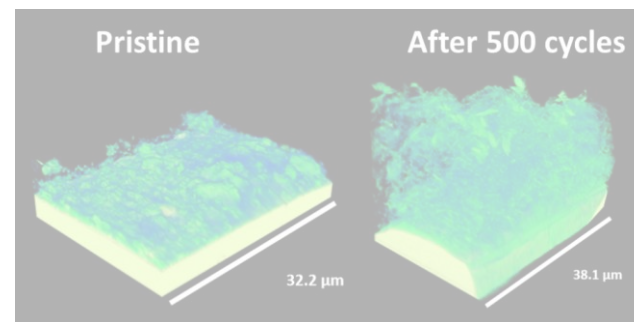


Capacity Dependent 3D Morphological Evolution of Nanoporous Si Anode



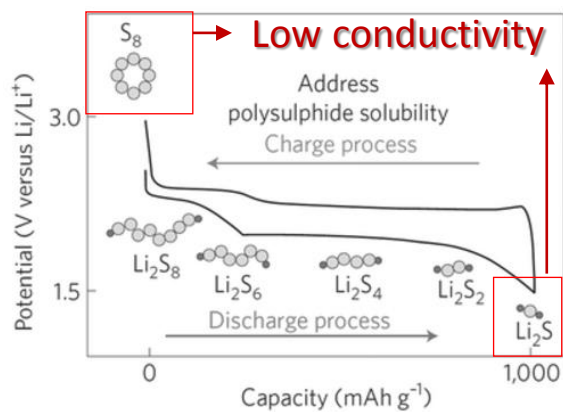
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Mechanistic Understanding of Li-S Batteries with Hybrid Electrodes: Multi-functional Additive

- **Li-S battery:** much higher energy density (6x by weight) & lower cost
- Hybrid electrodes utilizing metal sulfide additives: conductivity \uparrow & capacity +

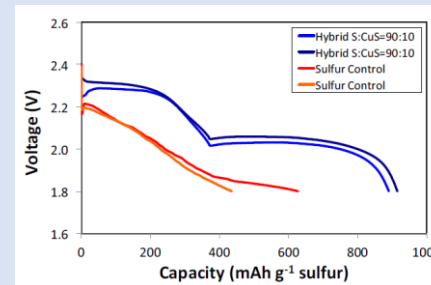


Capacity-Contributing
Conductive Additives

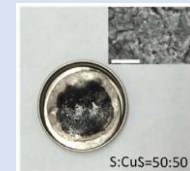
Metal Sulfide - Sulfur
Hybrid Electrode

CuS-S, FeS₂-S,
TiS₂-S, etc

Higher power capability
Capacity \uparrow @ high rate discharge



Poorly understood
Interactions
& Side-reactions



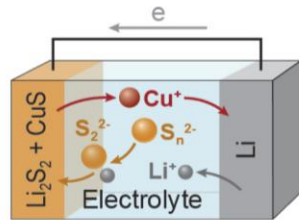
Dissolution/Re-deposition

- **Interaction between components (sulfur cathode, additives, Li anode, & electrolyte) at the system level & side-reactions :**
complex, specific to each type of additive with vastly different behavior, poorly understood
- **Understanding mechanisms** \rightarrow guide designing better additives \rightarrow Li-S performance \uparrow

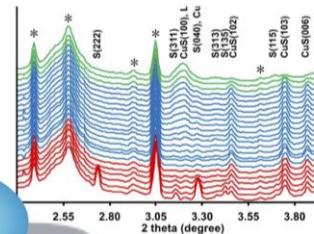
Conventional closed-cell electrochemical characterization
 \rightarrow average of different processes \rightarrow no longer sufficient

Operando Multi-modal Synchrotron Investigation for Structural and Chemical Evolution of CuS Additive in Li-S battery

**Multifunctional Additives
for Next Generation Battery**

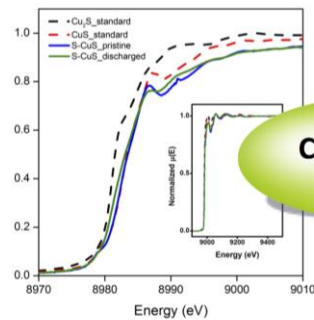


X-ray Diffraction



Structure

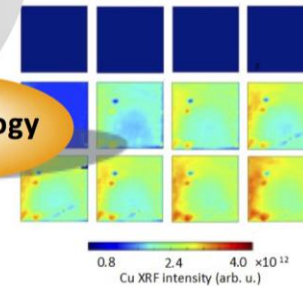
X-ray Spectroscopy



Chemistry

**Operando
Multi-modal
Synchrotron**

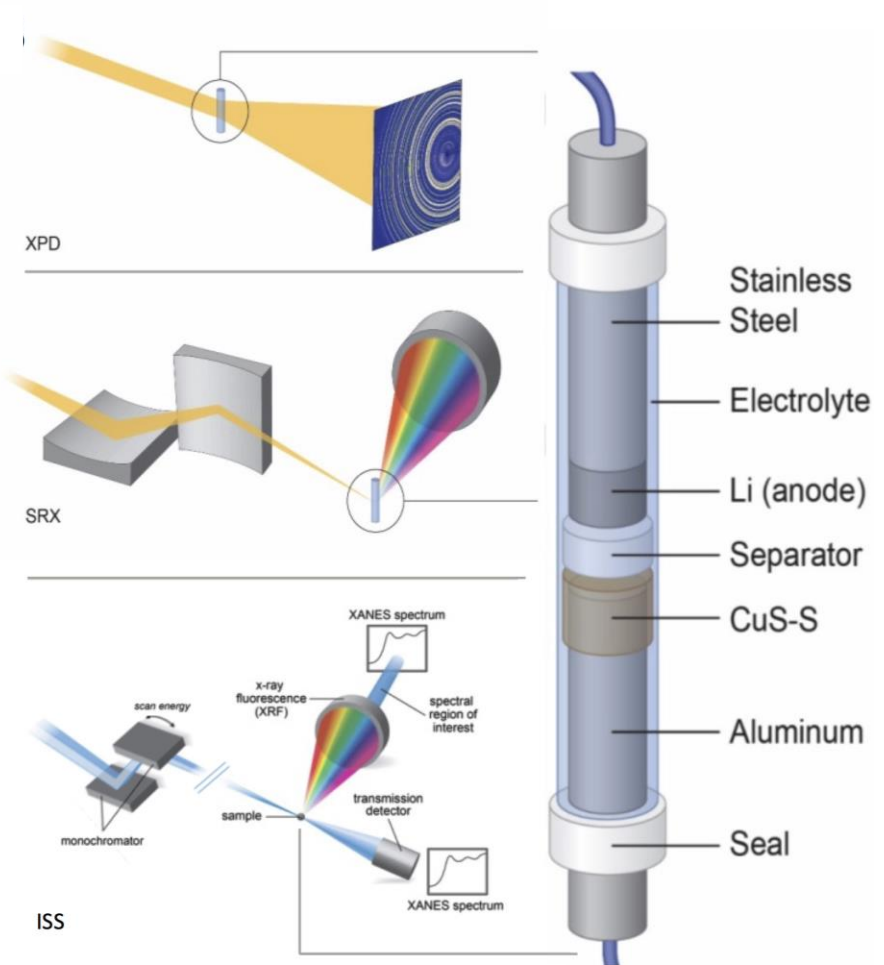
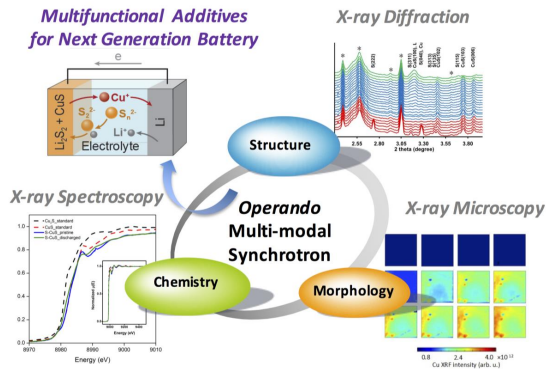
X-ray Microscopy



Morphology

Ke Sun, Chonghang Zhao, Cheng-Hung Lin, Eli Stavitski, Garth Williams, Jaiming Bai, Eric Dooryhee, Klaus Attenkofer, Juergen Thieme, Yu-chen Karen Chen-Wiegart, Hong Gan, *Scientific Reports* (2017)

Operando Multi-modal Synchrotron Investigation for Structural and Chemical Evolution of CuS Additive in Li-S battery



X-ray powder diffraction (XPD) 28-ID-2, NSLSII

- Wavelength: 0.1838 & 0.2362 Å
- Beam size: 0.5 mm²
- Technique: X-ray Powder Diffraction

Inner-Shell Spectroscopy (ISS) 8-ID, NSLSII

- Beam energy: Cu K-edge (8979 eV)
- Spot size: 0.8 mm × 0.3 mm (h × v)
- Technique: X-ray Absorption Spectroscopy

Sub-micron Resolution X-ray Spectroscopy (SRX), 5-ID, NSLSII

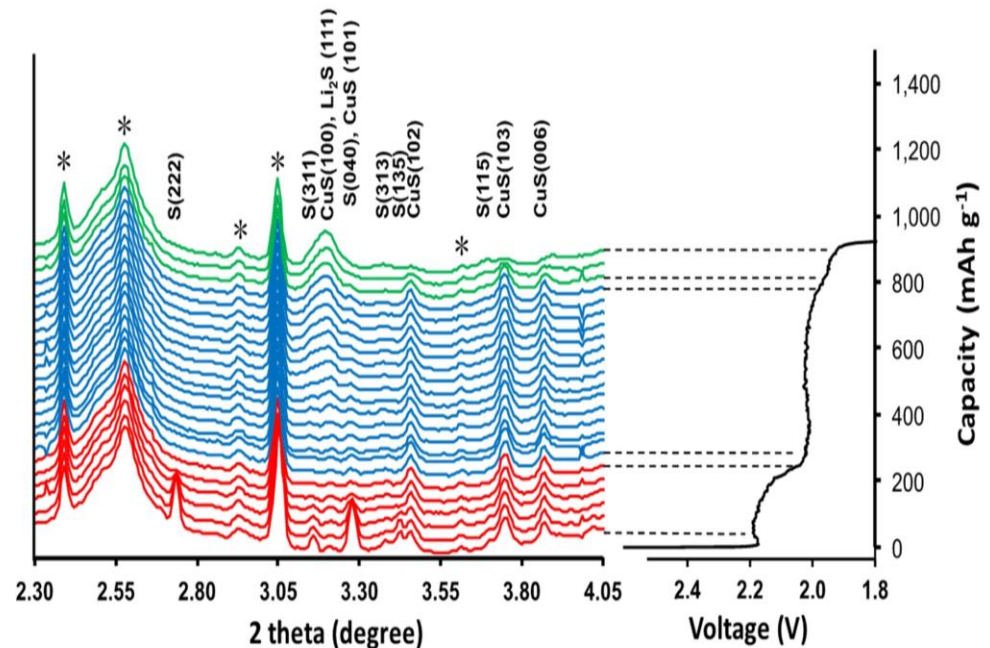
- Beam energy: 10 keV
- Spot size: ~ 1 μm
- Technique: X-ray Fluorescence Microscopy

Ke Sun et al., Scientific Reports (2017)

Operando X-ray Diffraction at XPD, NSLS-II

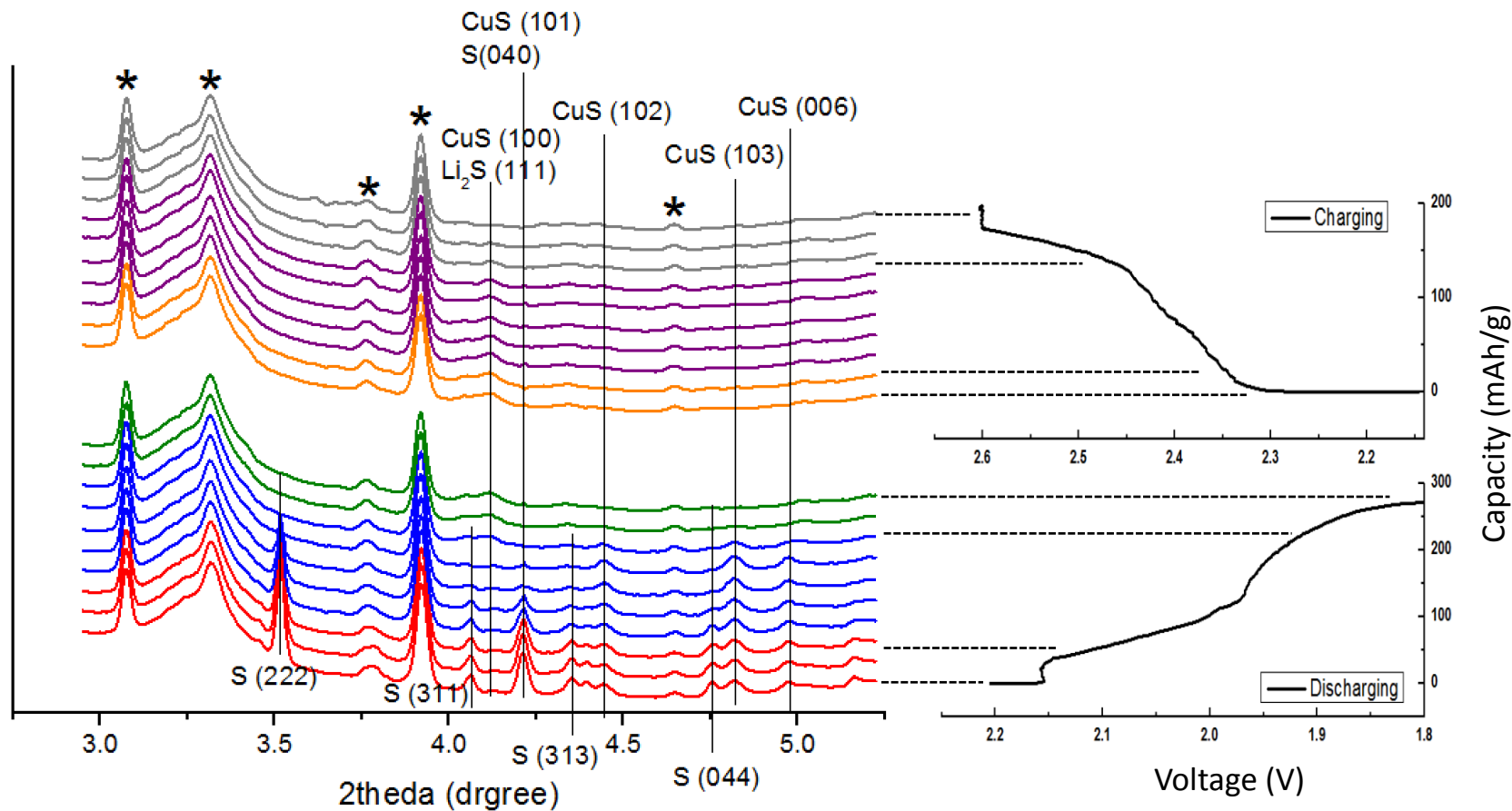
- Reactions during discharge:
 - $2\text{Li} + 2\text{CuS} \Rightarrow \text{Cu}_2\text{S} + \text{Li}_2\text{S}$ (~ 2.14 V with slow rate)
 - $2\text{Li}_2\text{S}_n \rightarrow \text{Li}_2\text{S}_2$ (~ 2 V, liquid-solid reduction)

- The only new peak emerging is $\text{Li}_2\text{S}(111)$
- No Cu_2S detected during the discharge in diffraction pattern may be due to amorphous phases
- Metal Cu is also not observed due to the cut-off voltage at 1.8 V.



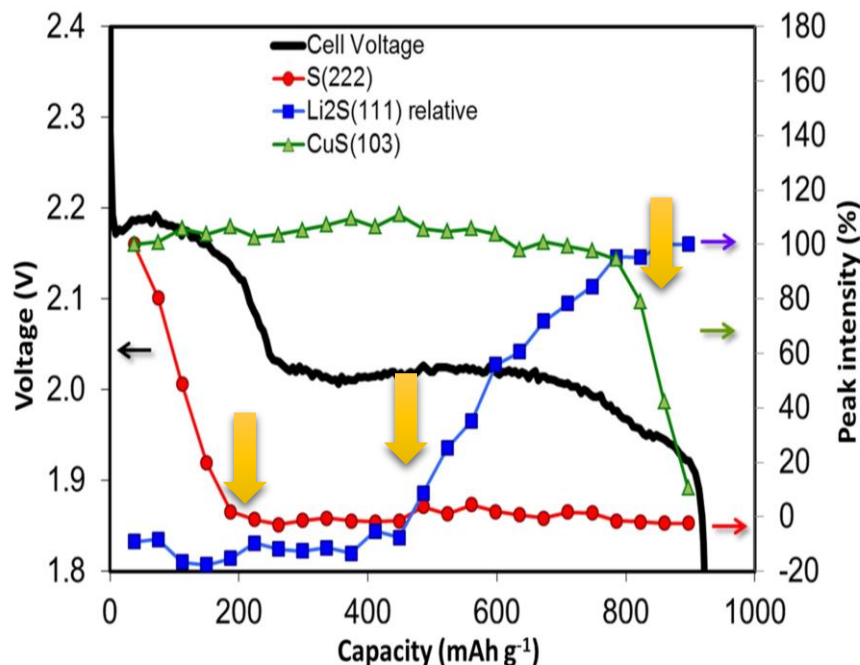
Ke Sun et al., Scientific Reports (2017)

Operando X-ray Diffraction at XPD, NSLS-II Lithiation and De-lithiation



Phase Evolution during Discharge by Operando X-ray Diffraction

The area of three strong peaks were integrated at each depth of discharge.



Ke Sun et al., Scientific Reports (2017)

1) S (222) decreases right after lithiation

→ crystalline sulfur converts into polysulfides → dissolve into the electrolyte

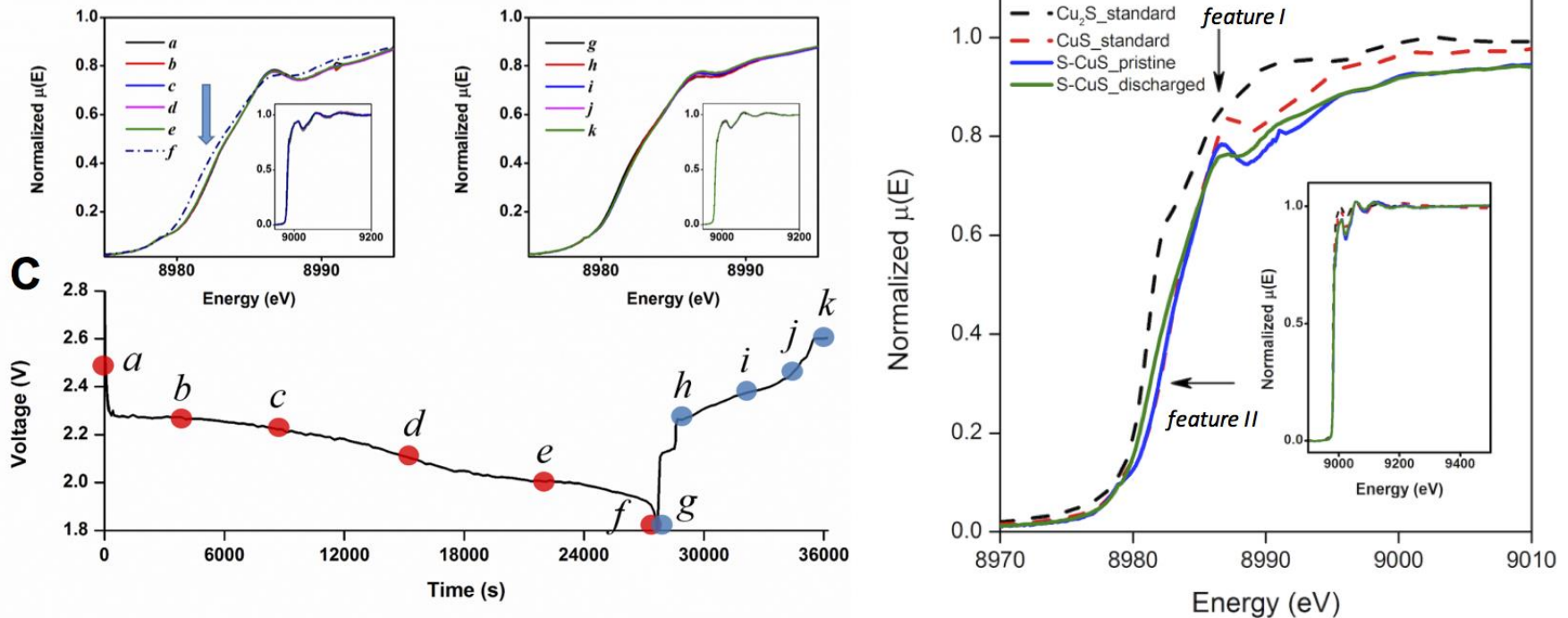
2) Li_2S (111) only appears at ~43% capacity

→ nucleation after the solid sulfur is fully converted into polysulfides and Li_2S_2 remains amorphous

3) CuS (103) intensity only decreases until the end of discharge

→ conversion of Li_2S_2 to Li_2S occurs ahead of CuS reduction

Operando X-ray Absorption Spectroscopy (XANES)

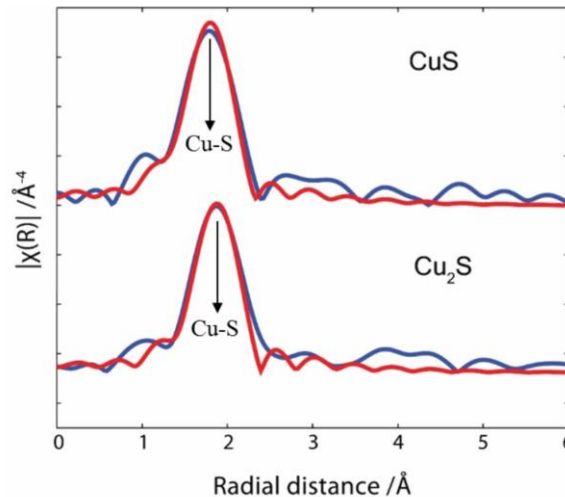
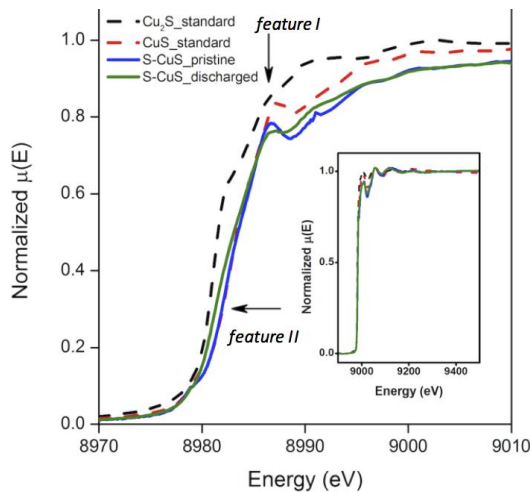


- Series of *operando* Cu K-edge XANES spectra during cycling:
 - No XANES spectral evolution until end of discharge (point f)
 - Consistent with Operando XRD
- At the end of discharge (point f): two changes in the XANES spectrum
 - *feature I*, which is characteristic for CuS, in Fig. D becomes less prominent
 - *feature II*, the edge jump shifts to lower energies

Ke Sun et al., Scientific Reports (2017)

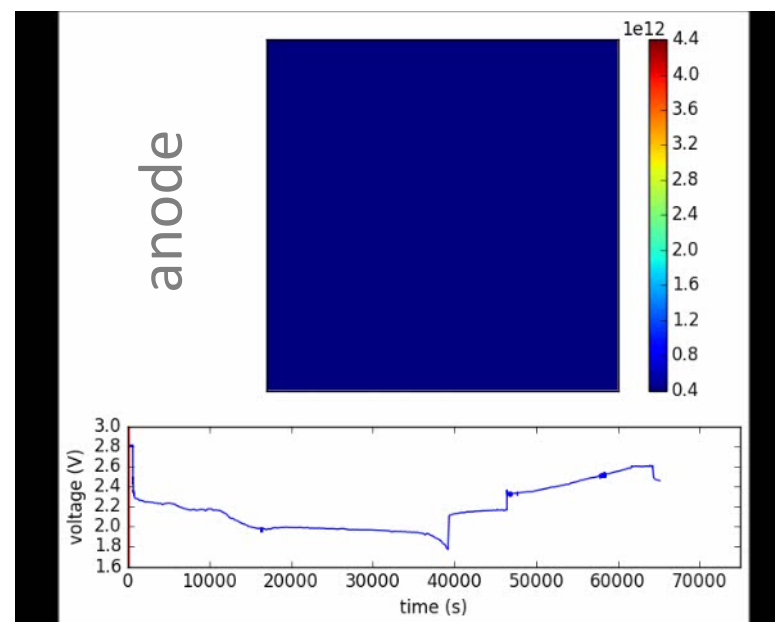
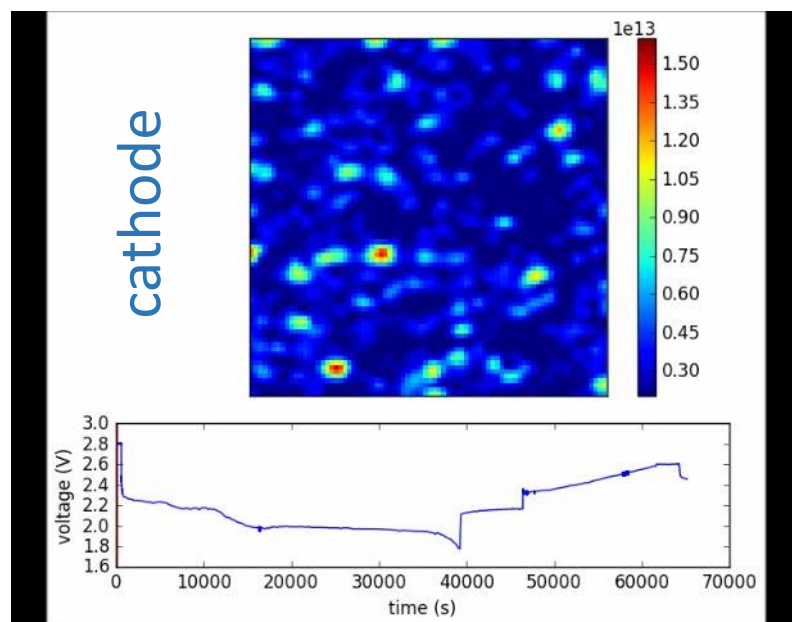
Operando X-ray Absorption Spectroscopy (EXAFS): Structural analysis after full discharge,

- the Cu-S coordination number is reduced from 2.5 to 2.1
→ An elongation of the average bond length from 2.27Å to 2.29Å.
- The average composition of the discharged material $\sim \text{Cu}_{1.3}\text{S}$.



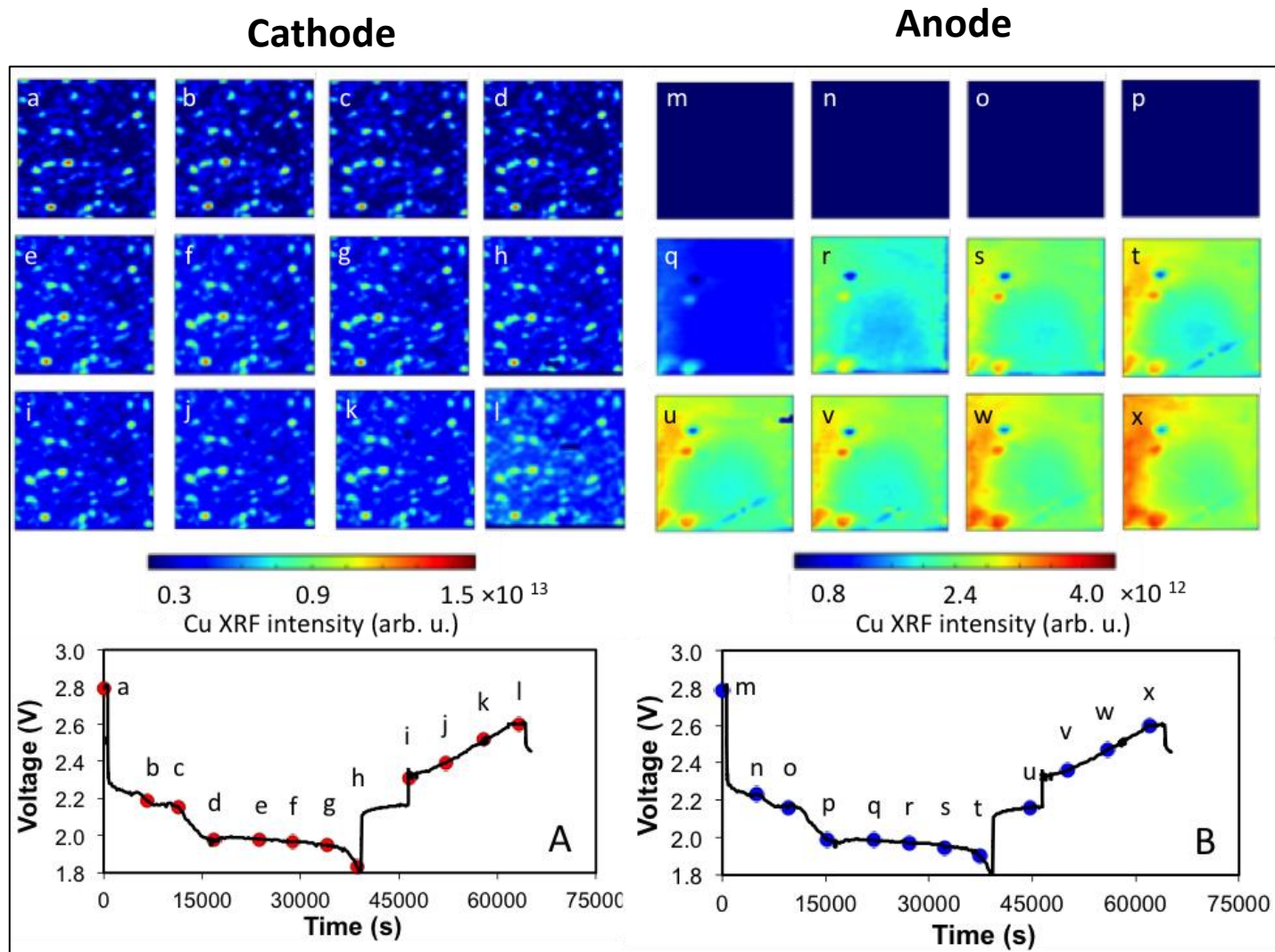
Sample	Cu-S coordination number	Cu-S distance, \AA	Debye-Waller factor
CuS (standard)	2.5 ± 0.2	2.272 ± 0.002	0.01 ± 0.001
Cu ₂ S (standard)	1.5 ± 0.3	2.315 ± 0.02	0.02 ± 0.003
Discharged (point f)	2.1 ± 0.2	2.289 ± 0.02	0.01 ± 0.003

Operando XRF studies on Cu dissolution/re-deposition in Li-S battery with CuS additives



X-ray Spectroscopic Imaging: morphology & chemistry

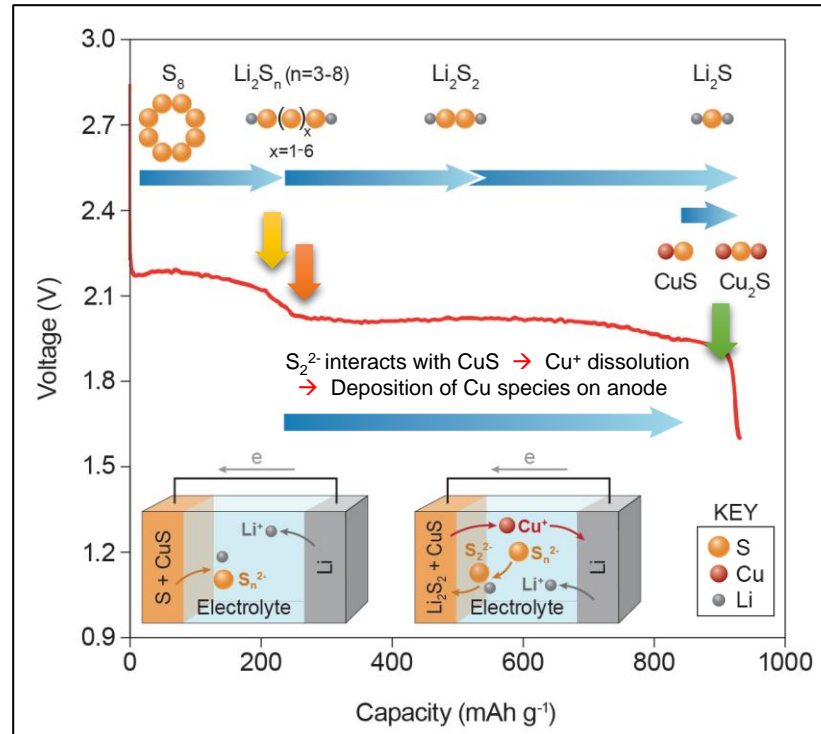
Operando XRF studies on Cu dissolution/re-deposition in Li-S battery with CuS additives



Mechanistic Understanding on CuS Additives

Interaction in Li-S battery

The crystalline sulfur cathode is completely consumed during the initial 21% of cell discharge, converted to high order polysulfide (Li_2S_n , $n = 4$ to 8).



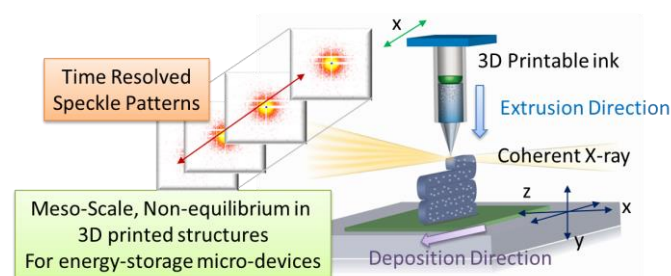
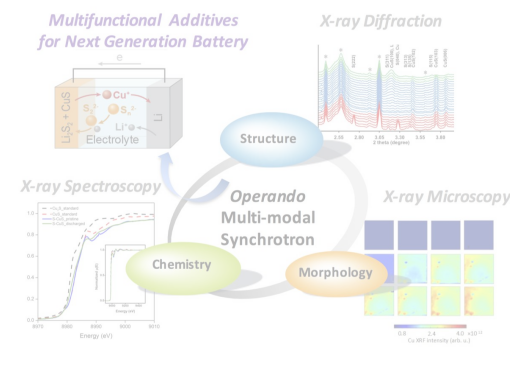
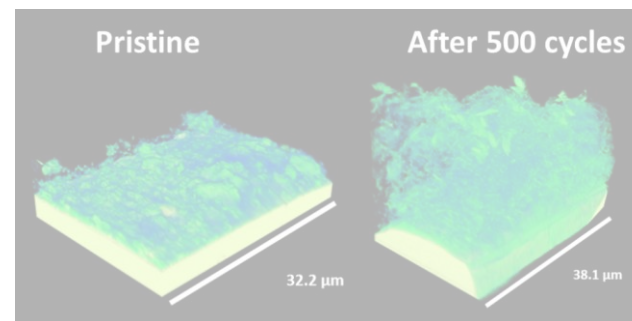
The lithiation of CuS starts at the very end of discharge to form amorphous Cu_2S at $\sim 1.95\text{V}$. During this process, CuS interacts strongly with soluble low order polysulfide species, either super saturated S_2^{2-} or S^{2-} . The dissolved Cu-ions then migrate from the cathode side to the anode side through the electrolyte, changing SEI layer and leading to capacity fade.

Starting from $\sim 25\%$ to $\sim 43\%$ cell discharge (beginning of the plateau discharge region), the conversion of polysulfide to amorphous phase Li_2S_2 happened, followed by conversion of Li_2S_2 into Li_2S by additional lithiation up to $\sim 85\%$ cell discharge.

Ke Sun et al., Scientific Reports (2017)

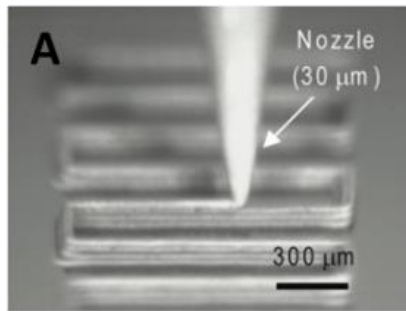
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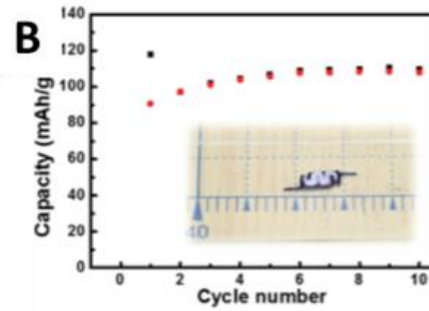


Meso-Scale Dynamics of 3D-Printing by *In Situ* Multi-modal Synchrotron Approach

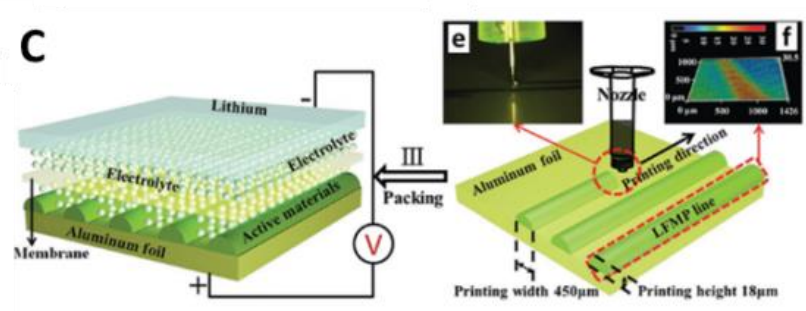
3D printing energy-storage micro-devices, or micro-batteries



Sun et al., *Adv. Materials* **2013**

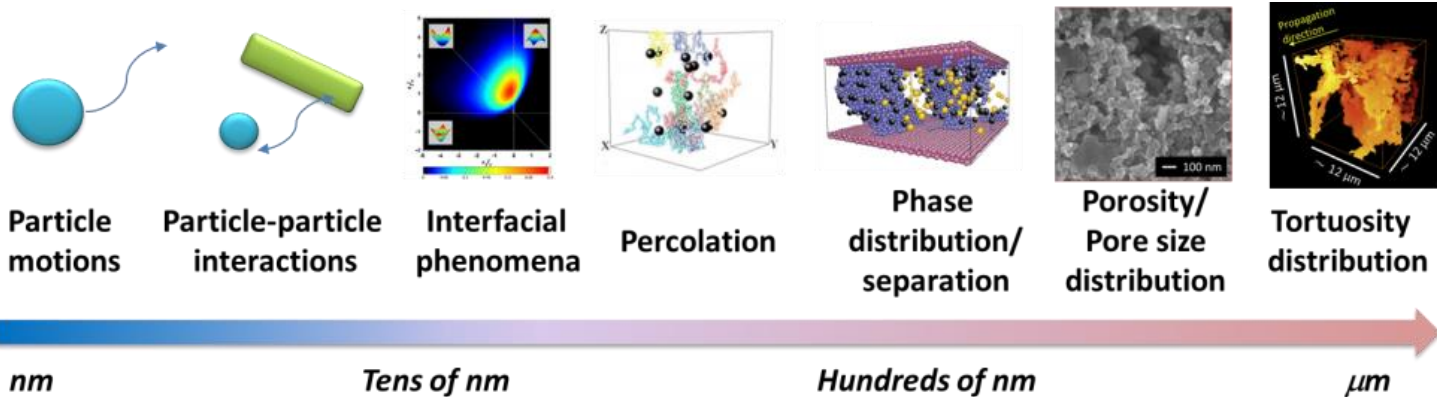


Fu et al., *Adv. Materials* **2016**



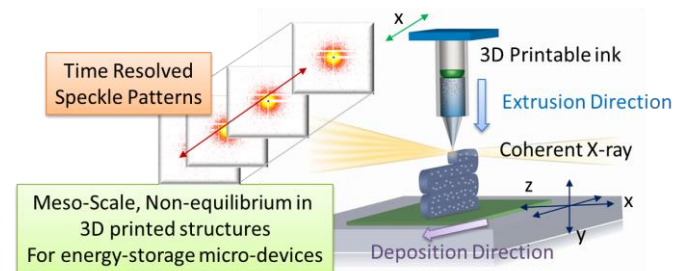
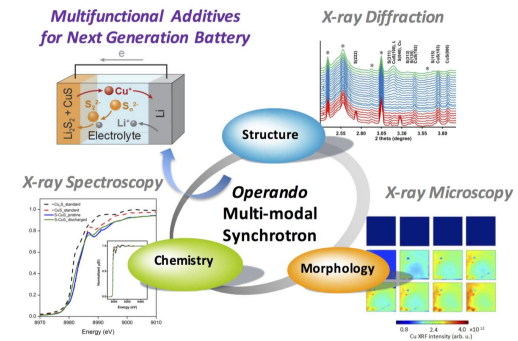
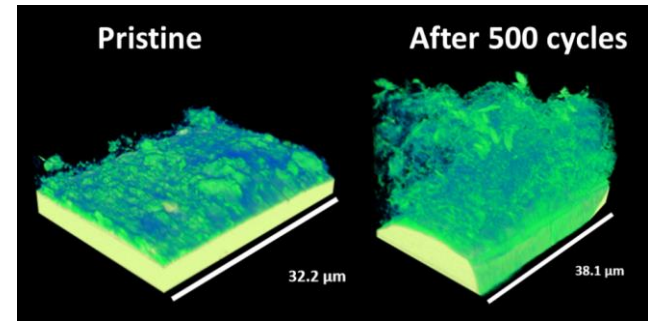
Hu et al., *Adv. Energy Mat.* **2016**

Meso-structures determine critical properties in a 3D printed structure

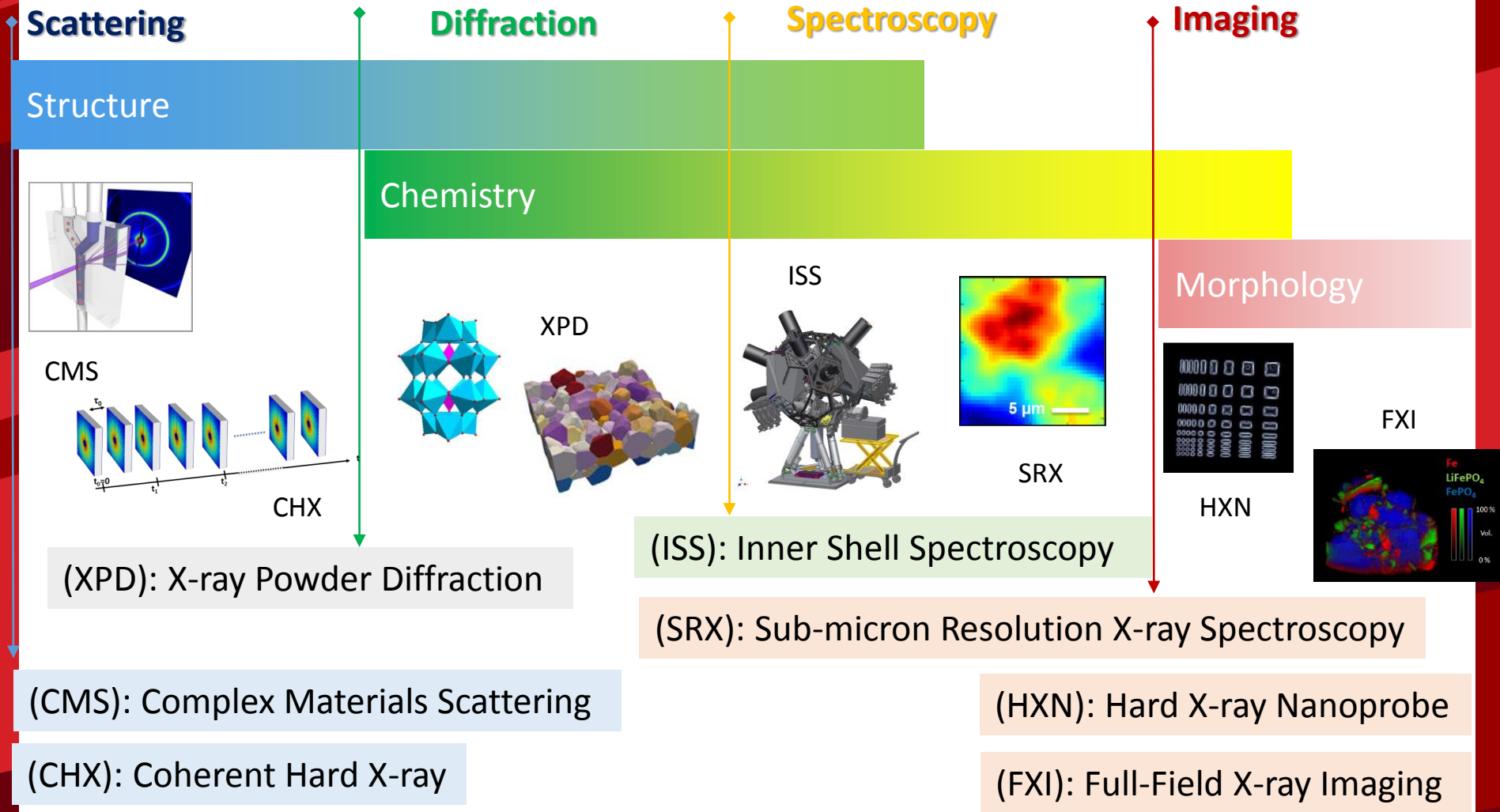


Summary

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Suite of beamlines with complementary techniques – Enabling time-resolved, *operando*, multi-modal and multi-dimensional studies



The Multi-Modal Issues Task Force @ NSLS-II

to identify the issues involved in utilizing techniques across multiple beamlines at NSLS-II, and in combining synchrotron techniques with other techniques, such as the electron-based imaging methods at CFN.

Potential research schemes and impact

Supporting facilities
& functions

User access

Multi-modal
Approach

Beamline hardware

Software

Goal

Logistics

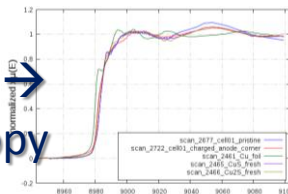
Techniques

Morphology
→ imaging



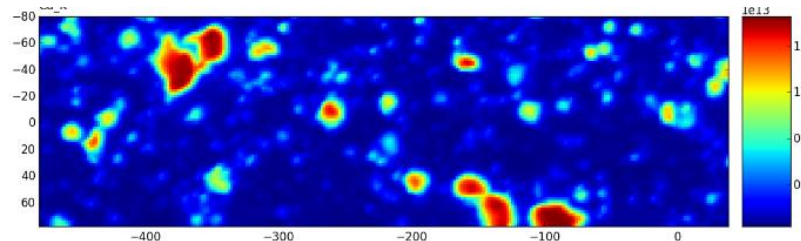
Structure
→ diffraction
& scattering

Chemistry →
spectroscopy

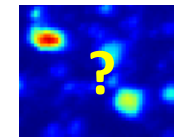


Complex

Heterogeneous

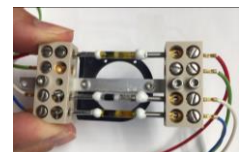


Li-S battery with CuS additives



Particles
(material)

Clusters
(Electrodes)



Interfaces & Devices
(System)

Acknowledgements



NSLS-II, BNL

Juergen Thieme, Garth Williams (SRX)
Jianming Bai, Eric Dooryhee (XPD)
Eli Stavitski, Klaus Attenkofer (ISS)
James Biancarosa & Michael Maklary
DAMA: Li Li, Tom Caswell, Dan Allan

Sustainable Energy, BNL: Hong Gan, Ke Sun ***Center for Functional Nanomaterials, BNL***

Eric Stach, Fernando Camino, Kim Kisslinger, Gwen Wright,
Dmytro Nykypanchuk

Beamline: Vincent De Andrade **APS-TXM**

Technical support: Alex Deriy
HPCAT (Glove box): Jessie Smith, Curtis Benson
TXM Optics: Michael Wojcik

Materials Science and Chemical Engineering Stony Brook University

<https://you.stonybrook.edu/chenwiegart/>



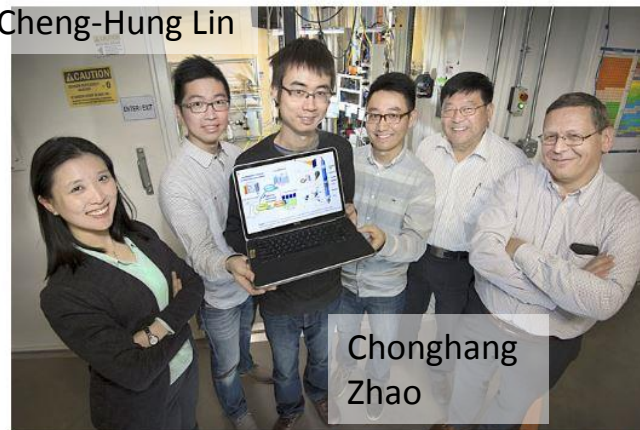
By Laura Migdolichian



Multi-Modal *Operando* X-Ray Study Yields New Insights on Lithium-Sulfur Batteries

May 2, 2018

Cheng-Hung Lin



Chonghang
Zhao

Karen Chen-Wiegart, Cheng-Hung Lin, Ke Sun, Chonghang Zhao, Hong Gan, Klaus Attenkofer (left to right) used the Inner Shell Spectroscopy (ISS) beamline to study the charge and discharge process of a Lithium-sulfur battery. [ENLARGE](#)

Thank You!

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